



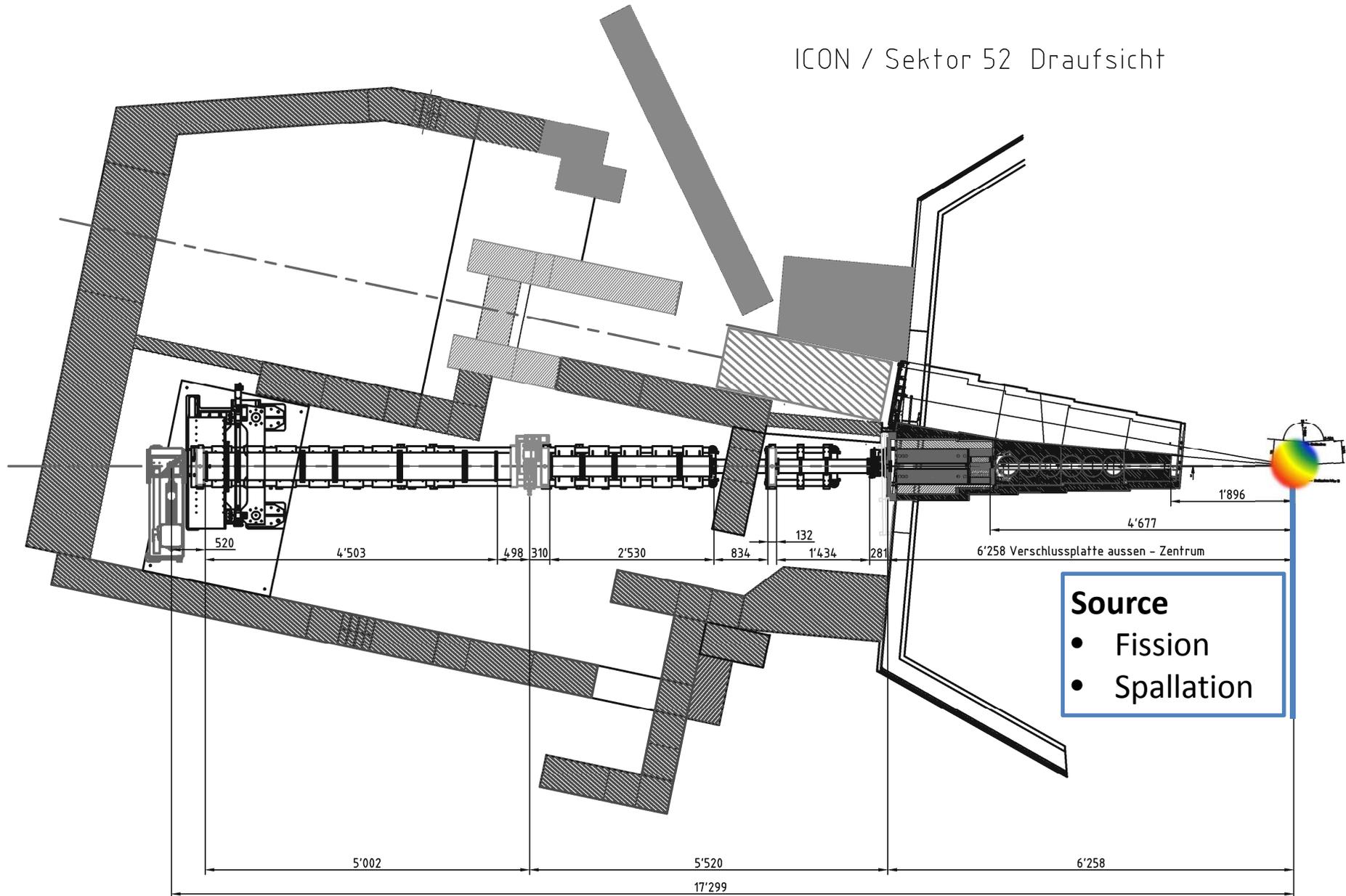
Wir schaffen Wissen – heute für morgen

Steven Peetermans

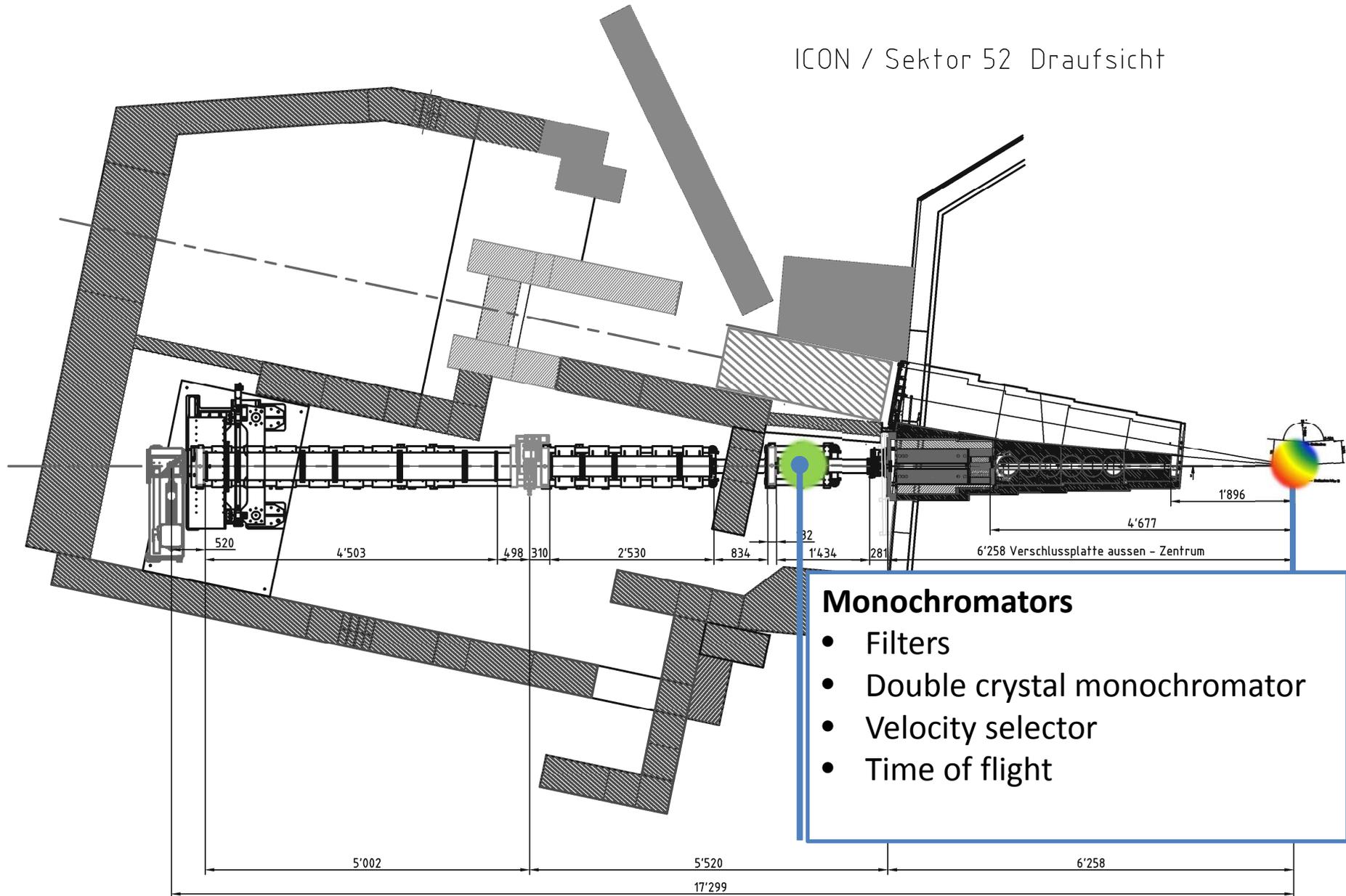
*Neutron Imaging & Activation Group, Paul Scherrer Institut, Switzerland  
now at AV Controlatom, Belgium*

**Energy-selective neutron imaging**

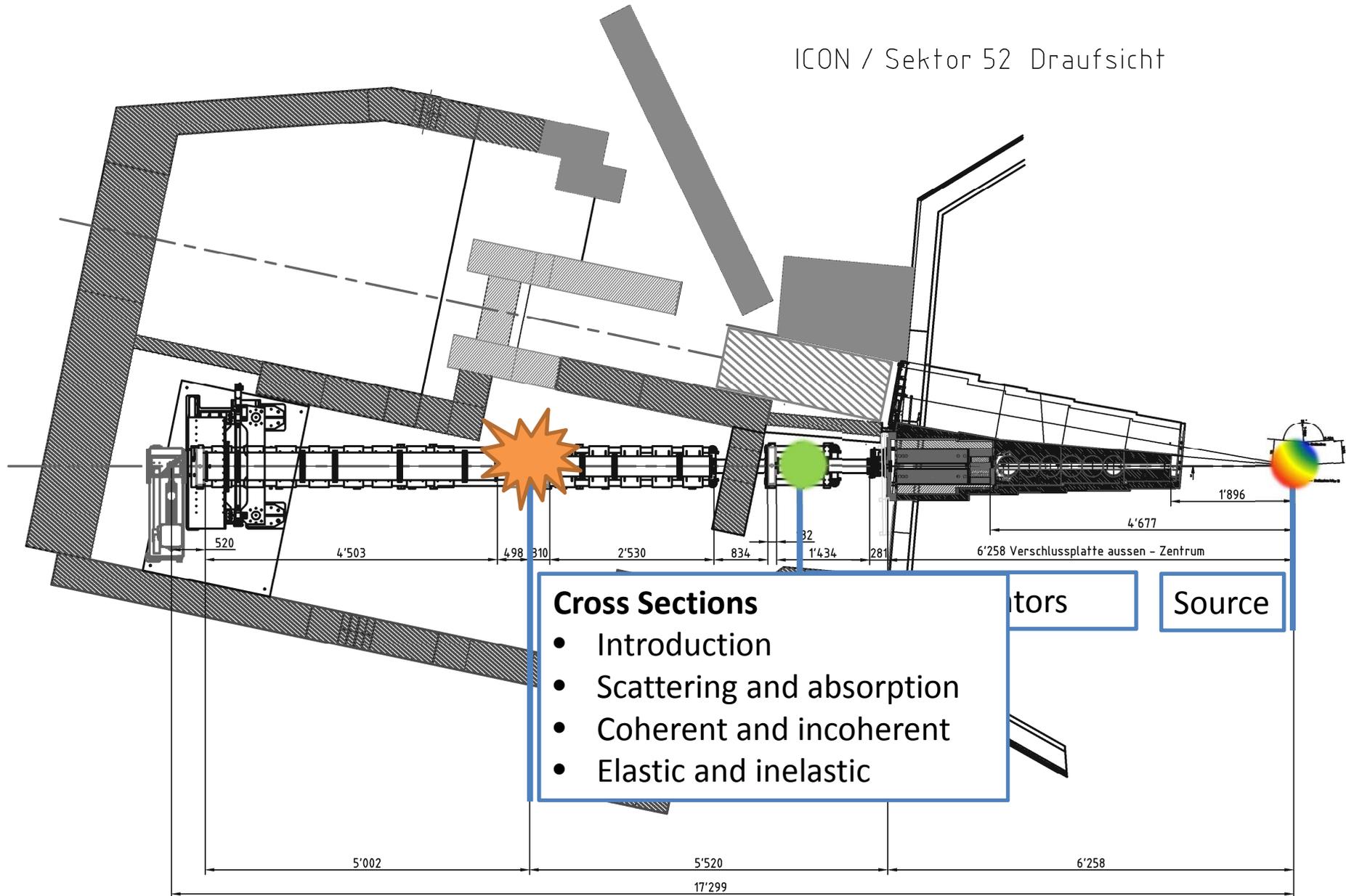
# ICON / Sektor 52 Draufsicht



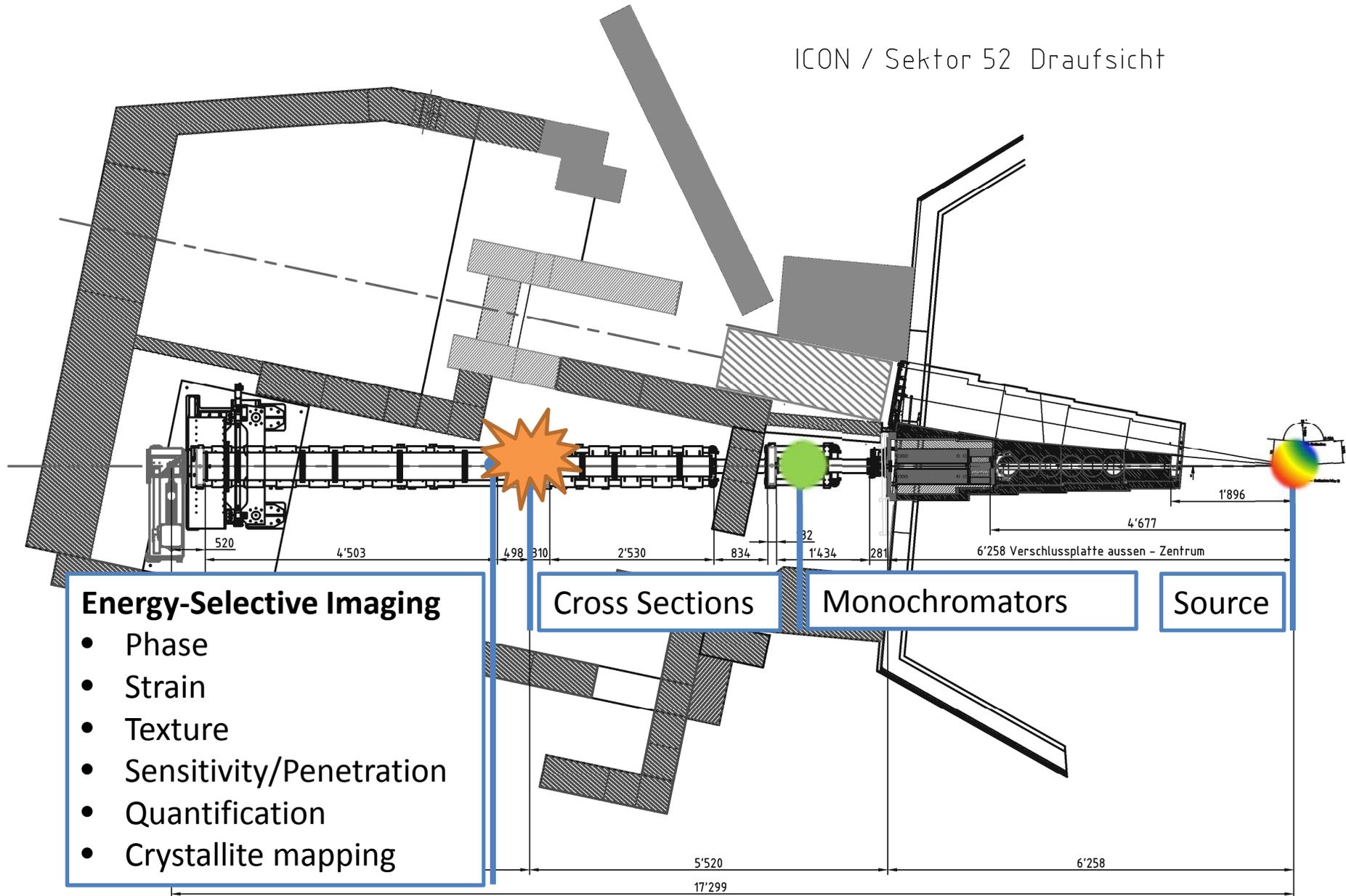
# ICON / Sektor 52 Draufsicht



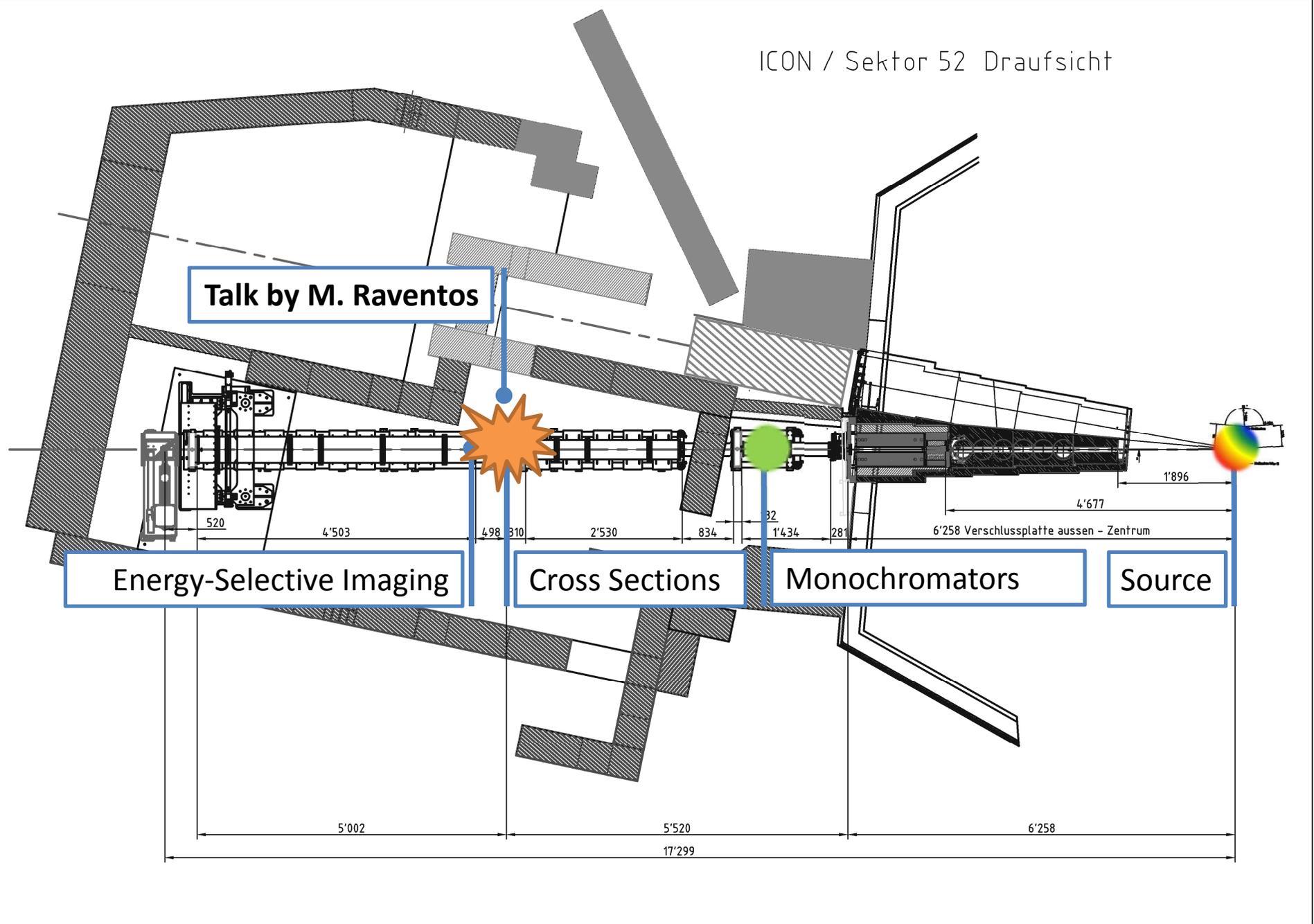
# ICON / Sektor 52 Draufsicht



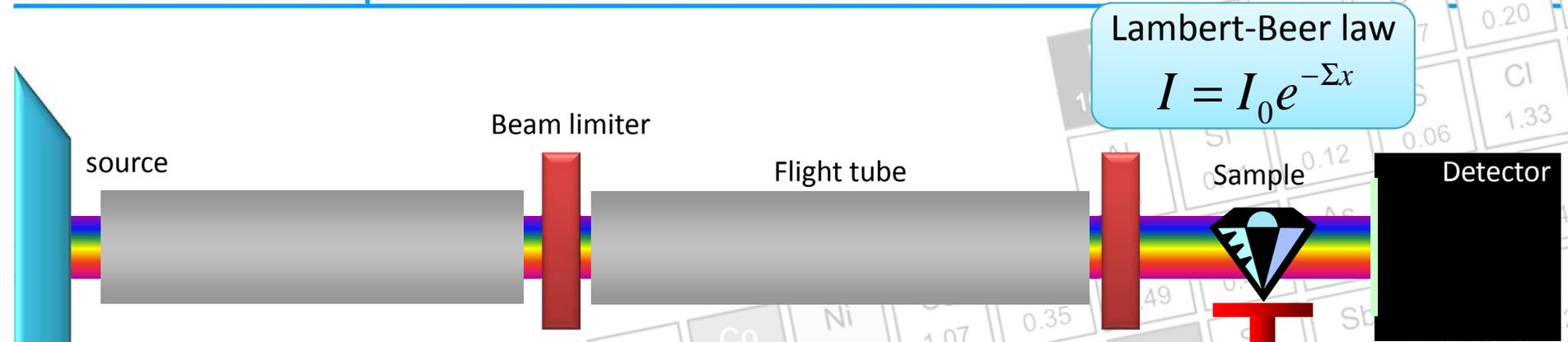
# ICON / Sektor 52 Draufsicht



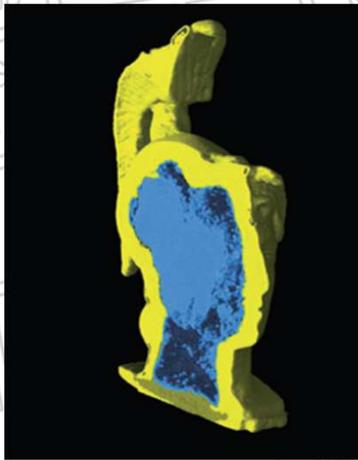
ICON / Sektor 52 Draufsicht



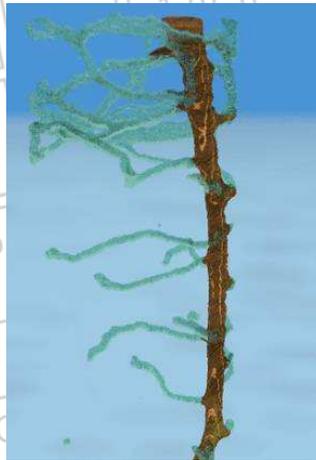
# Traditional transmission imaging



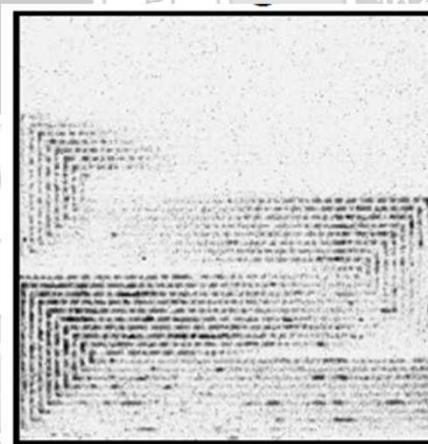
Spatial discrimination between materials of different cross-section



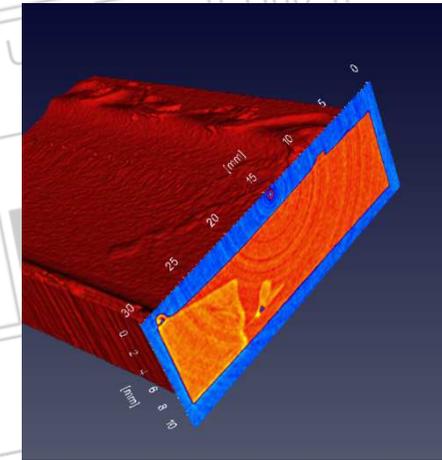
Archaeology



Biology

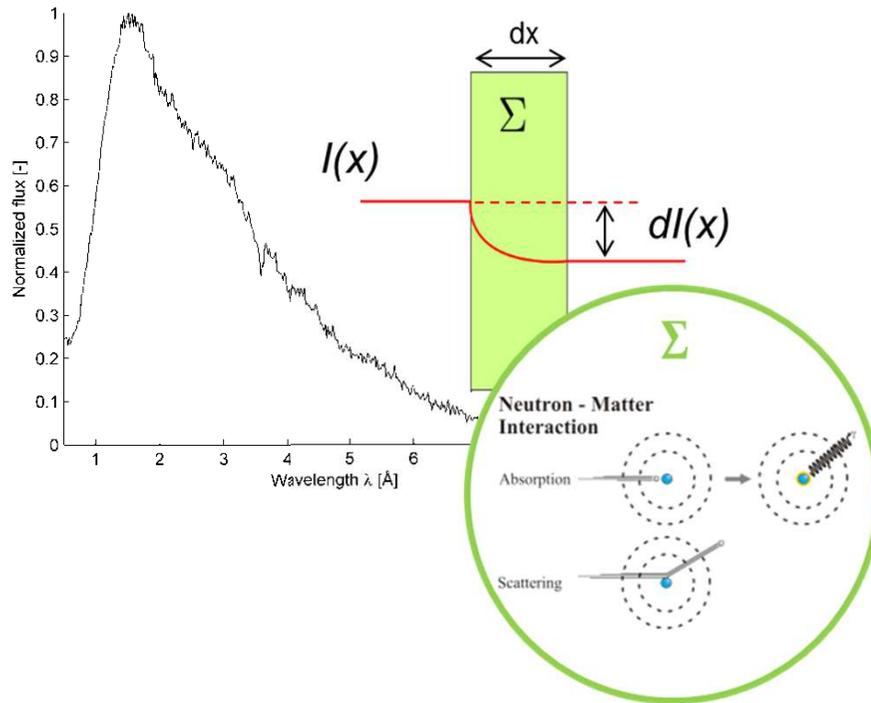


Fuel cells



Materials science

But  $\Sigma$  is more than just a number...

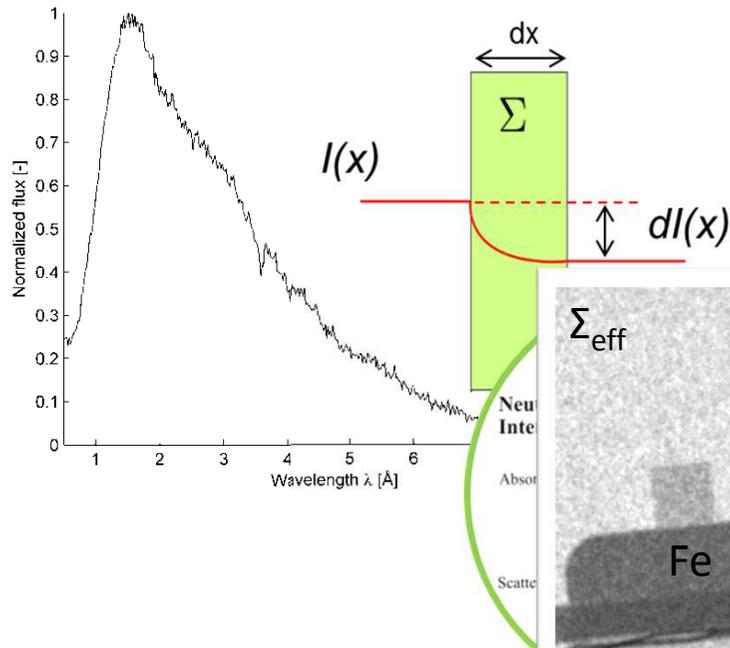


$$\int_{\lambda} I(\lambda) d\lambda = \int_{\lambda} I_0(\lambda) e^{-\Sigma(\lambda)x} d\lambda$$

Interaction ( $\Sigma$ ) depends on the neutron energy  
Incident beam spectrum  $I_0(\lambda)$

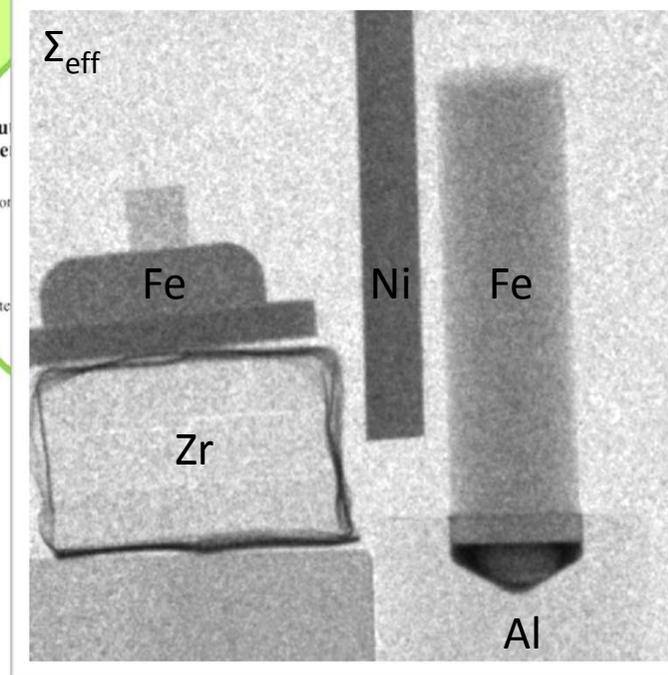
Traditionally imaging: spectral averaged  $\Sigma_{\text{eff}}$   
Information is lost

But  $\Sigma$  is more than just a number...



$$\int_{\lambda} I(\lambda) d\lambda = \int_{\lambda} I_0(\lambda) e^{-\Sigma(\lambda)x} d\lambda$$

Interaction ( $\Sigma$ ) depends on the neutron energy

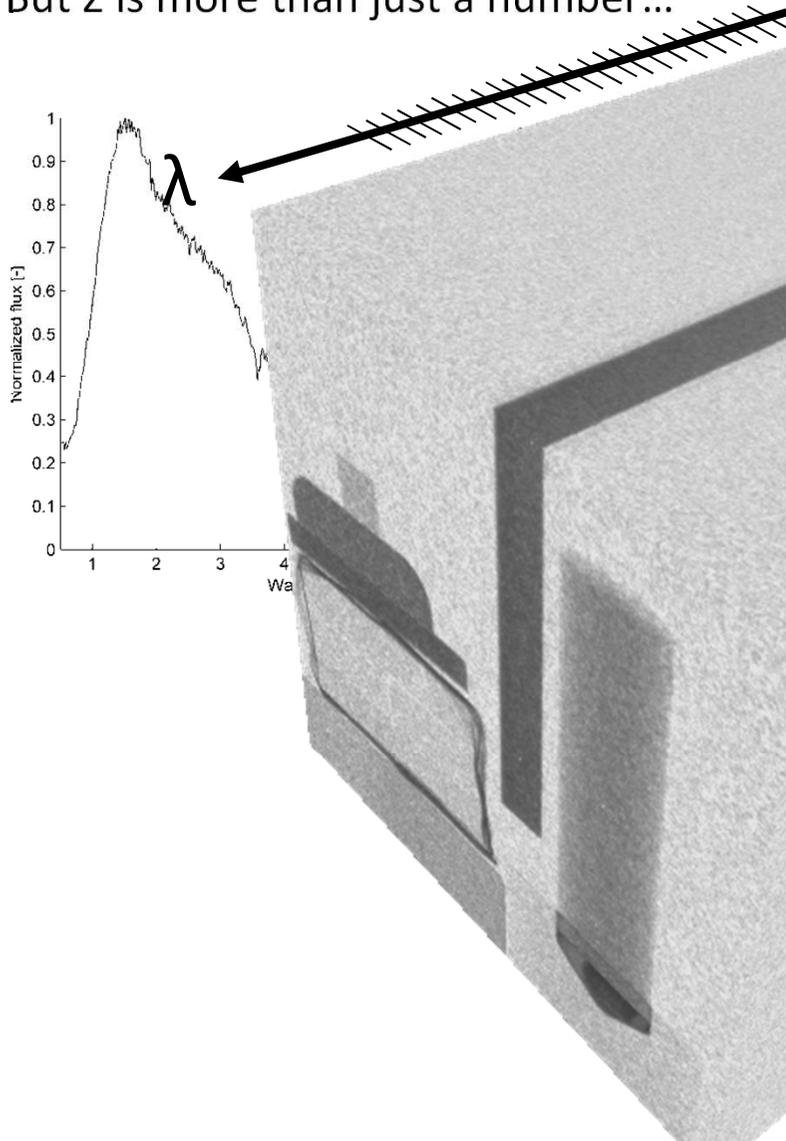


spectrum  $I_0(\lambda)$

imaging: spectral averaged  $\Sigma_{\text{eff}}$

st

But  $\Sigma$  is more than just a number...



$$I(\lambda) d\lambda = \int_{\lambda} I_0(\lambda) e^{-\Sigma(\lambda)x} d\lambda$$

... depends on the neutron energy

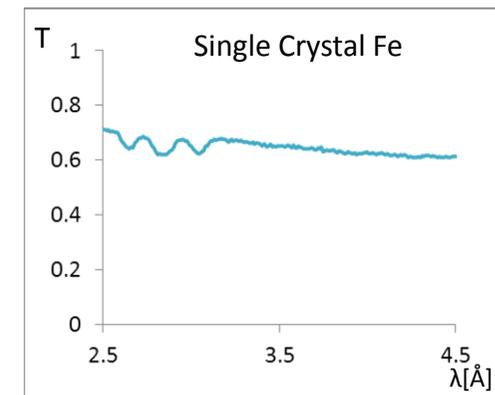
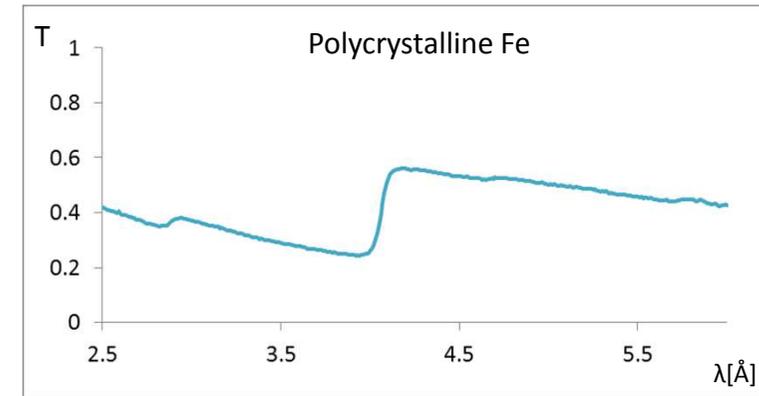
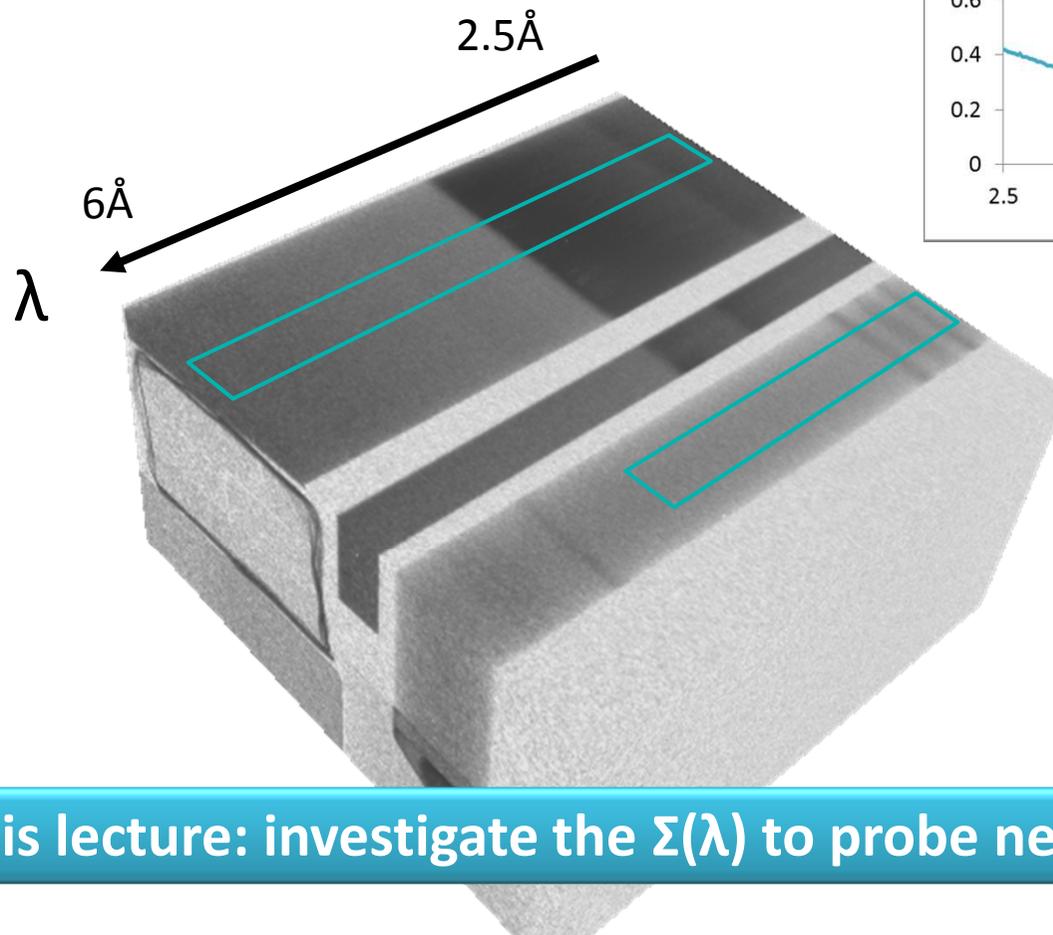
... spectrum  $I_0(\lambda)$

... averaging: spectral averaged  $\Sigma_{\text{eff}}$

... cost

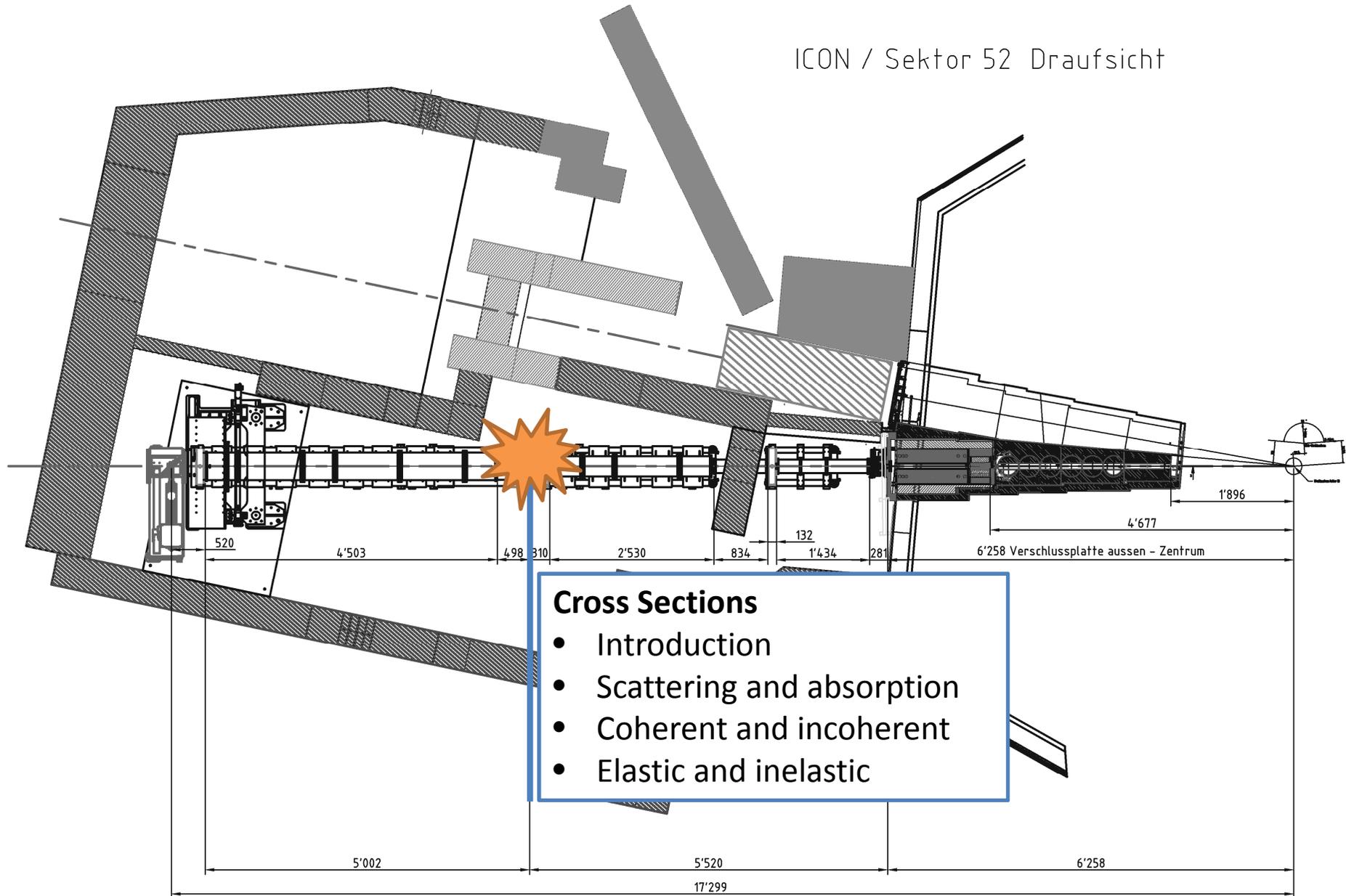
A radiograph for each wavelength

A closer look at the iron samples



This lecture: investigate the  $\Sigma(\lambda)$  to probe new sample properties

# ICON / Sektor 52 Draufsicht



Cross section = interaction probability

Microscopic cross section  $\sigma$ :

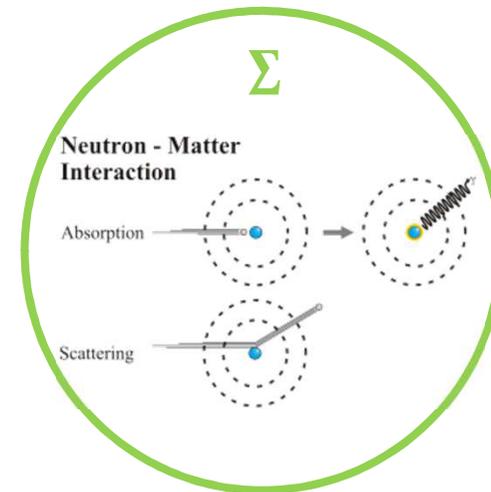
$$\frac{\# \text{ neutrons undergoing interaction per target nucleus}}{\text{incident neutron flux}}$$

Macroscopic cross section  $\Sigma$ :

$$\Sigma = N\sigma$$

$$\Sigma = \frac{\rho N_A}{M} \sigma$$

Clearly the interaction (cross-section) is wavelength dependent. What interactions are there possible?



$N$  = density of nuclei  
 $\rho$  = mass density  
 $M$  = molar mass  
 $N_A$  = Avogadro's constant (6.022e23#/mol)

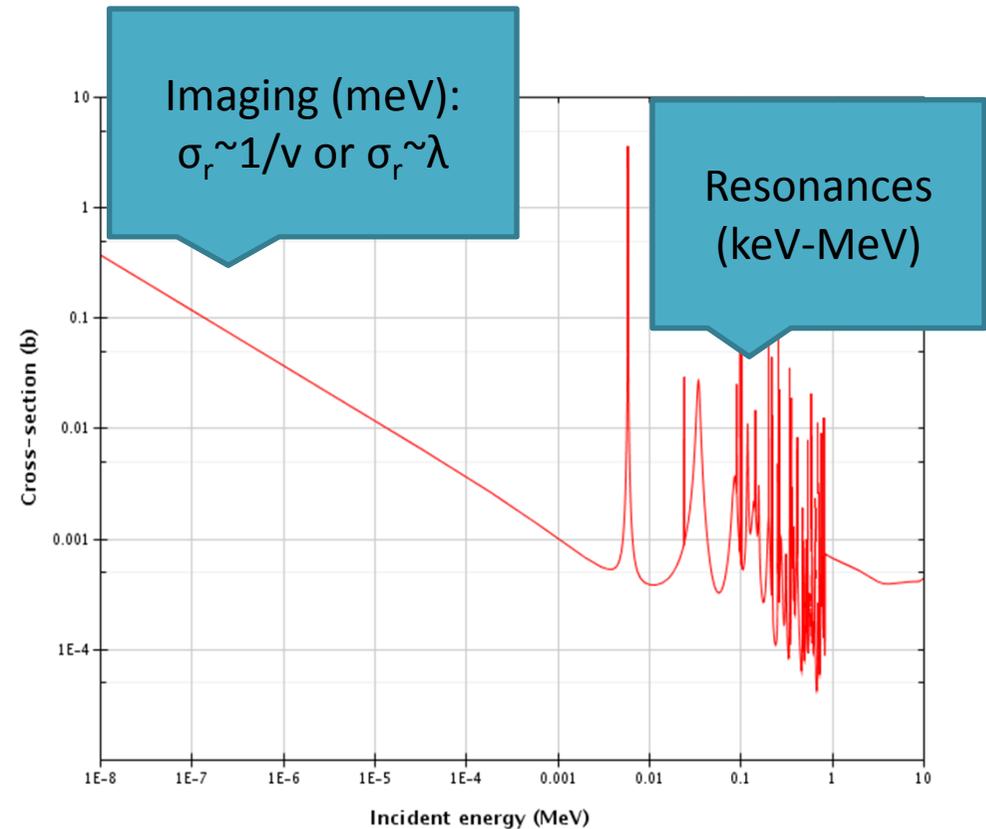
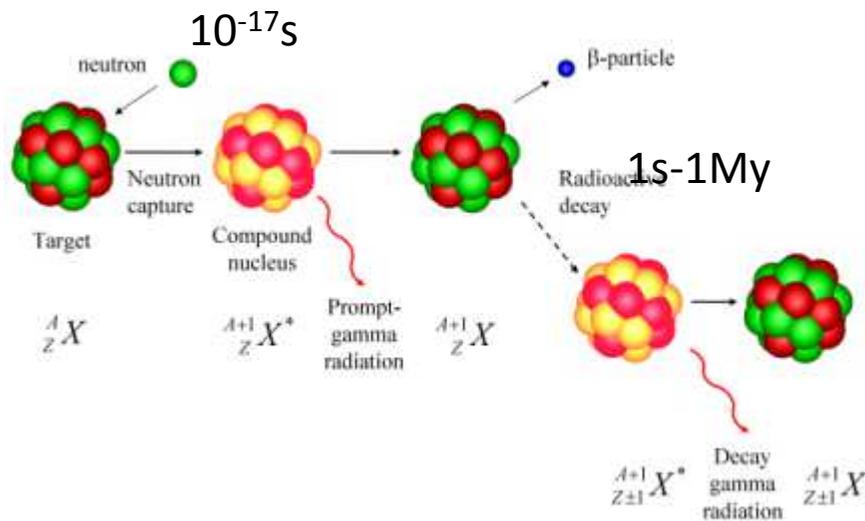


An incident neutron can be absorbed by the nucleus:

Formation of an excited compound nucleus

Prompt  $\gamma$ -emission

Radioactive decay ( $\beta, \gamma$ )



“The slower the neutron, the more time it has to interact and be absorbed”

Or Limit of the Breit-Wigner equation for describing resonances



# Activation of samples: example Cu

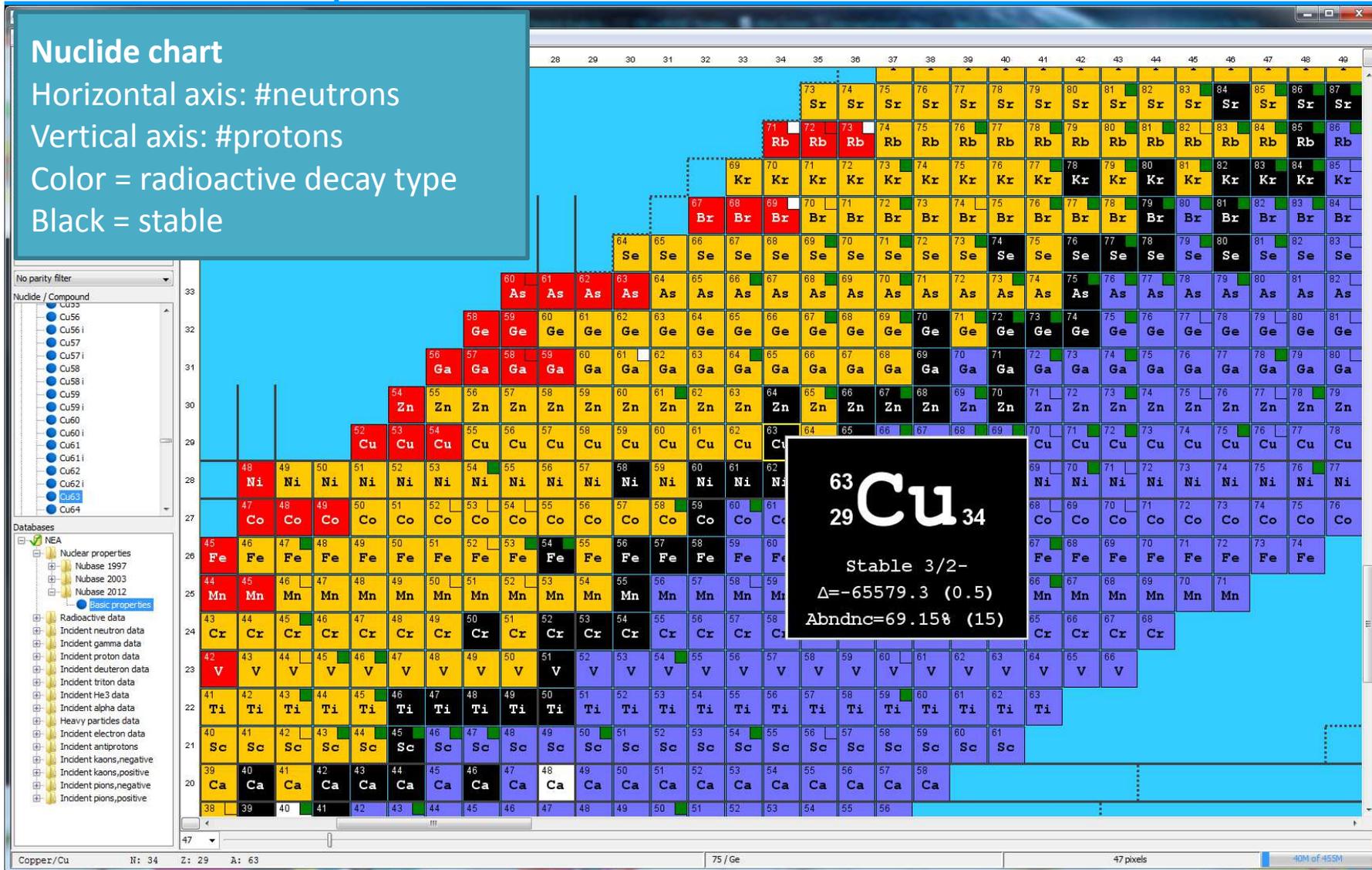
## Nuclide chart

Horizontal axis: #neutrons

Vertical axis: #protons

Color = radioactive decay type

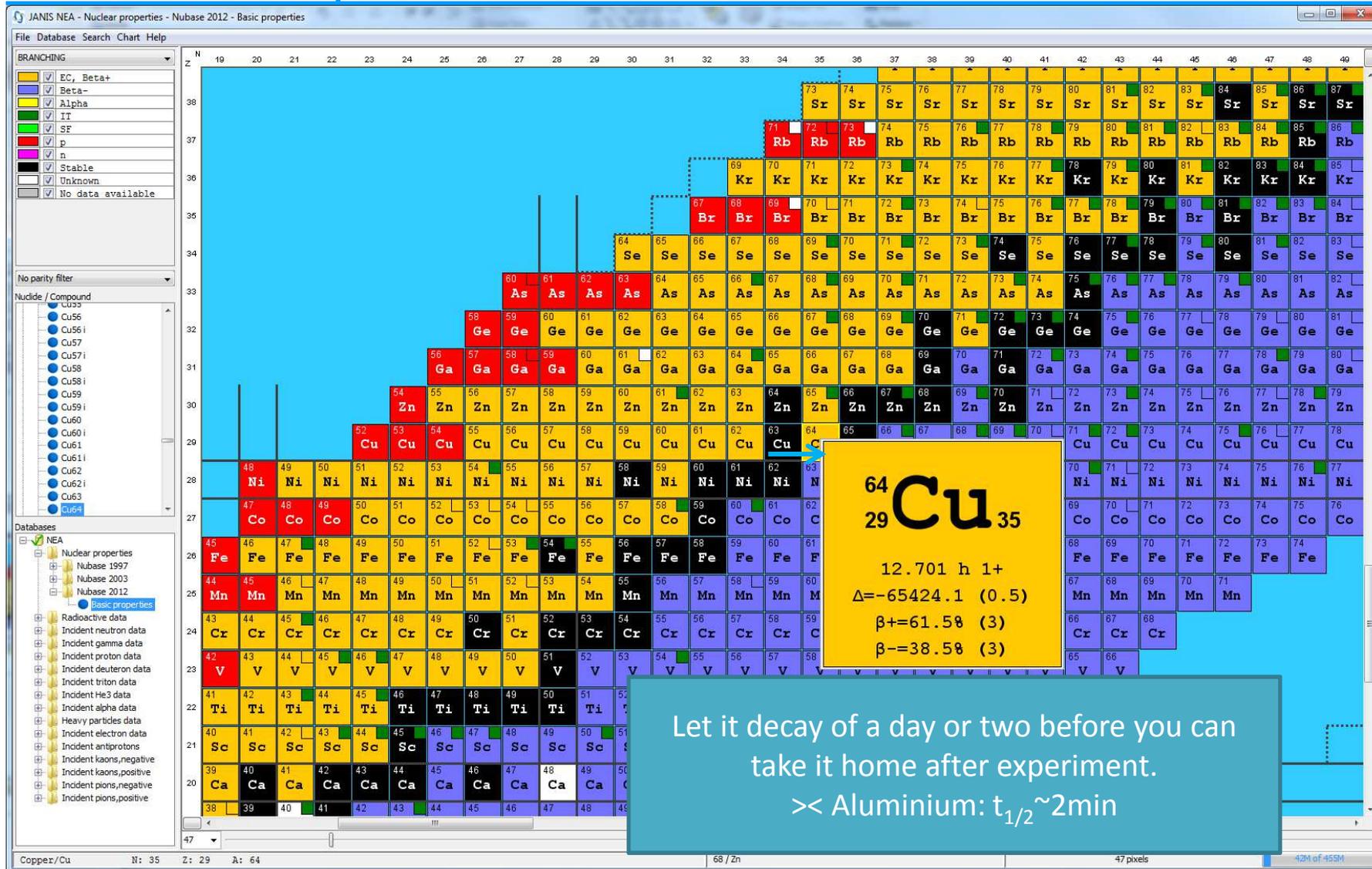
Black = stable



<http://www.oecd-nea.org/janis/> Check the nuclide chart to see what you can expect.

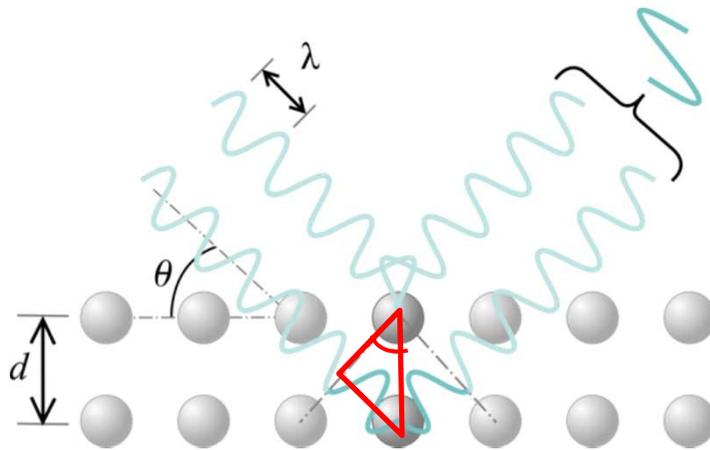


# Activation of samples: example Cu



<http://www.oecd-nea.org/janis/>

In many materials the nuclei form an ordered, periodic structure.

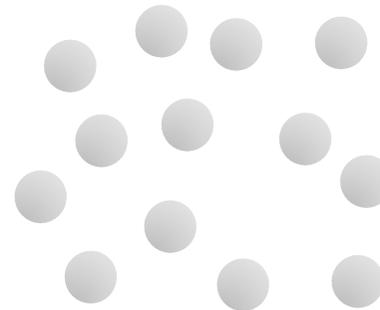


If phase difference equals  $n\lambda$ , with  $n= 1,2,3,\dots$   
reflected waves in phase (constructive  
interference):

Scattering is *Coherent*

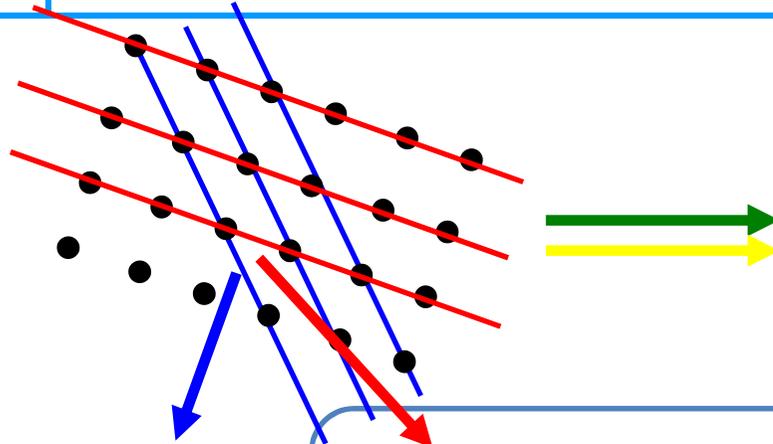
Condition:  $2d\sin\vartheta=n\lambda$  **Bragg Law**

Amorphous materials:  
no periodic structure is present  
scattering is *Incoherent*



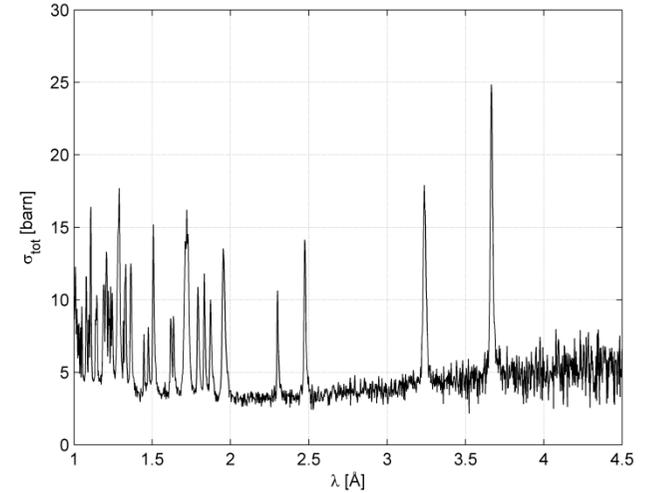
# Single Crystals and Polycrystalline material

Single crystal



Bragg's law

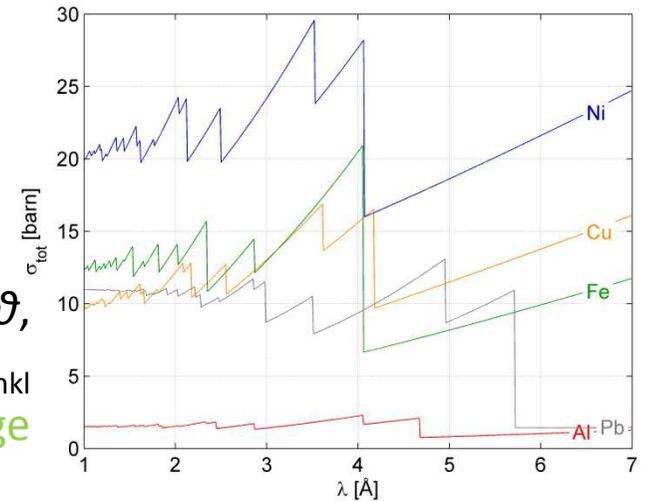
$$2d_{hkl} \sin \theta = \lambda_{hkl}$$



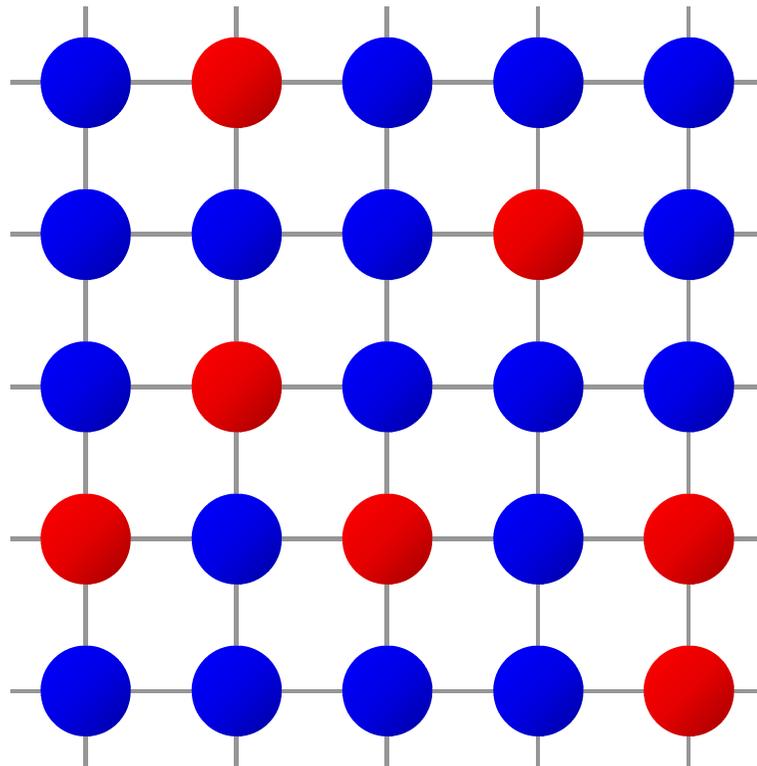
Polycrystalline material

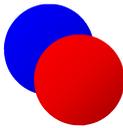
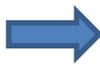


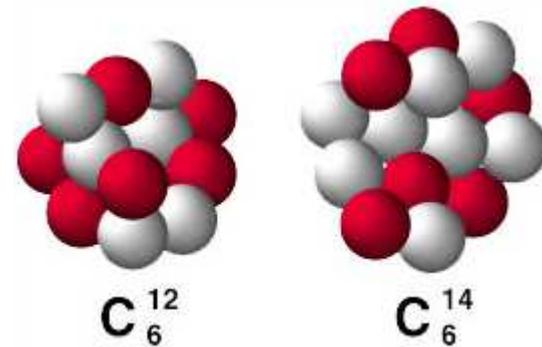
All orientations  $\vartheta$ ,  
Diffraction until  $\lambda_{hkl} = 2d_{hkl}$   
Followed by **Bragg Edge**



# Periodic (?) structure




 Different isotopes,  
 Interaction with nuclear spin  $I$   
 combines as  $J^\pm = I \pm \frac{1}{2}$   
 neutrons see this difference

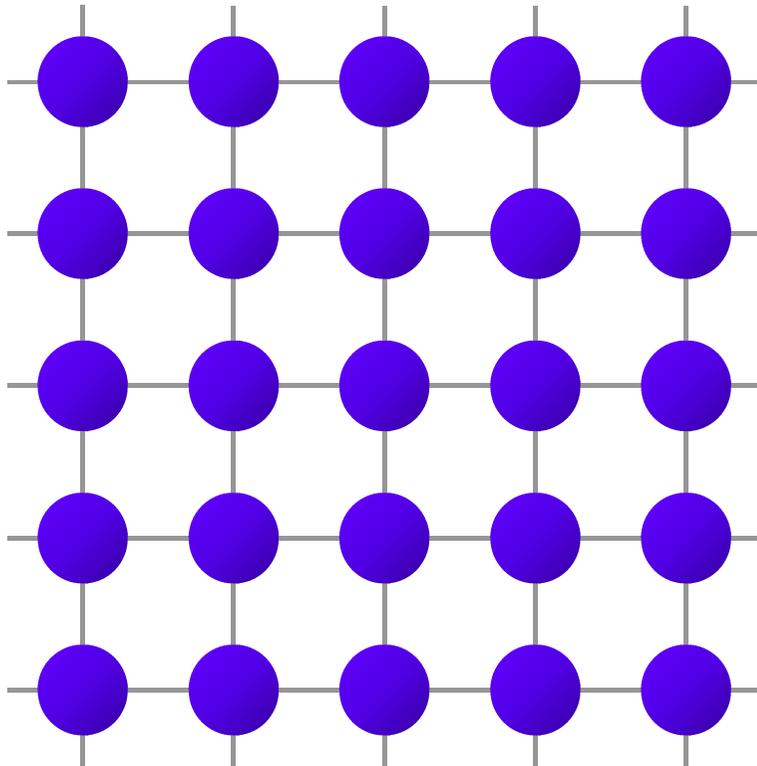


Not truly a periodic structure for our incident neutrons



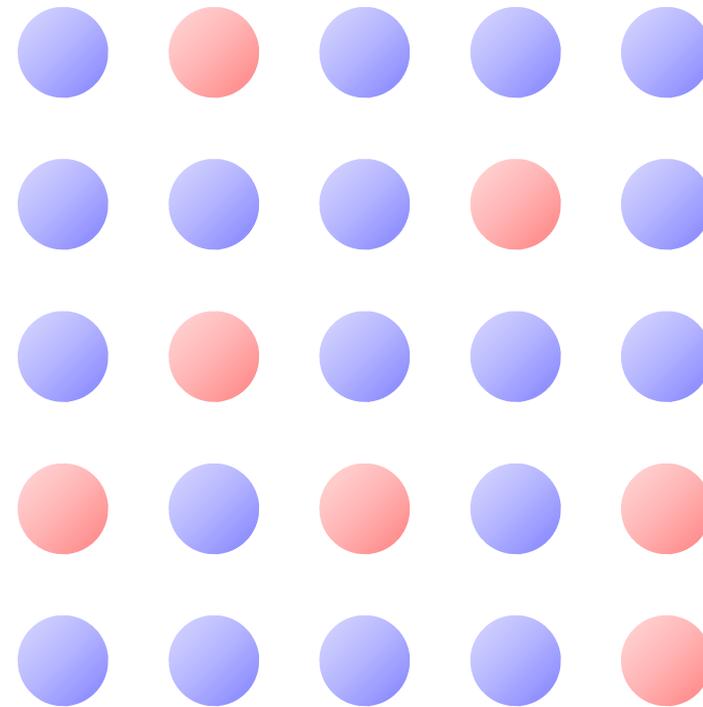
Often you have both coherent and incoherent scatt.

Periodic structure of  
average scattering length  $\bar{b}$



Coherent scattering

Corrections to it (random)

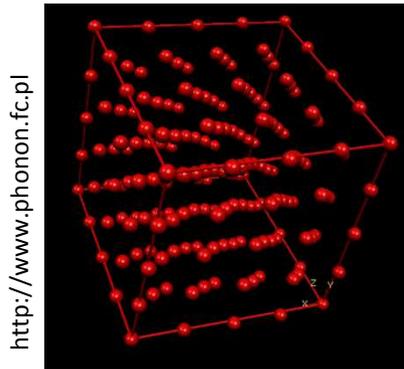


+

Incoherent scattering



Crystal at room temperature; Atoms feature thermal motion  
The energy associated with atomic vibrations is quantized: **phonons**



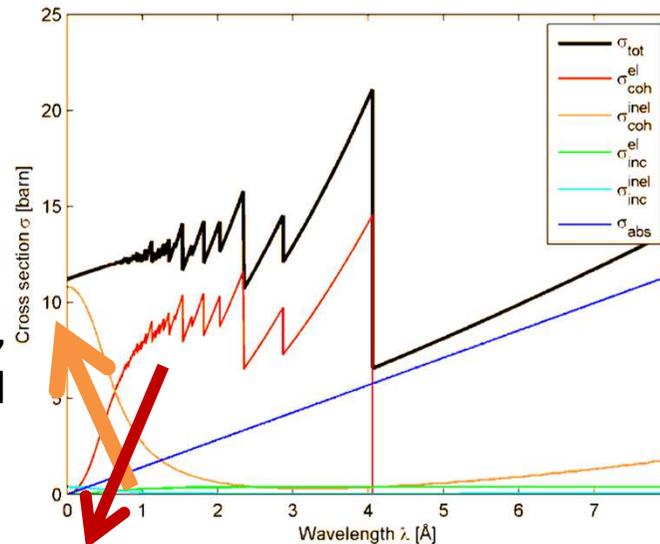
Incident neutron can  
Transfer energy to the crystal (phonon creation)  
Get energy from the crystal (phonon annihilation).

**Elastic scattering:** no energy transfer

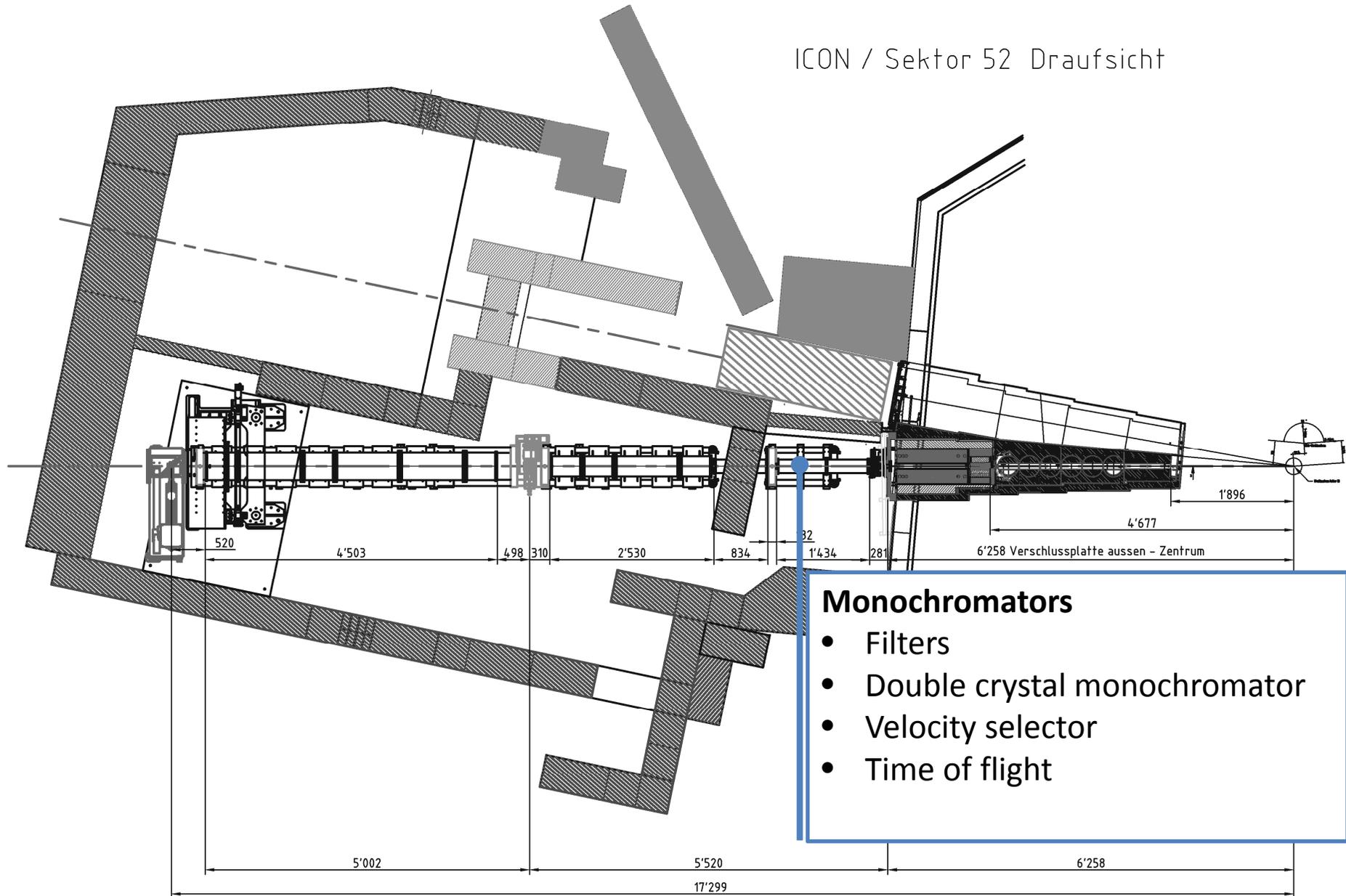
**Inelastic scattering:** energy transfer

The higher the neutron energy (lower  $\lambda$ ),  
The more phonons can be involved

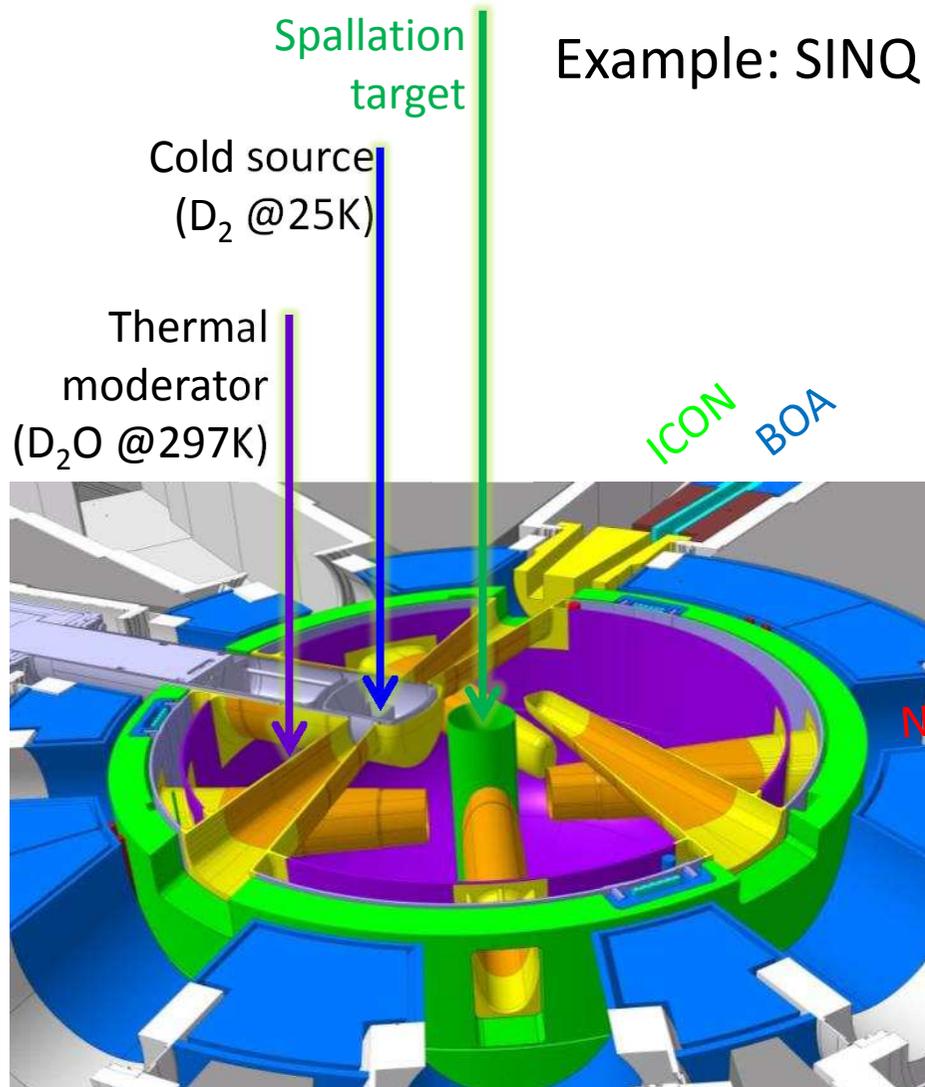
$$\sigma_{\text{inc}} \nearrow \quad \sigma_{\text{coh}} \searrow$$



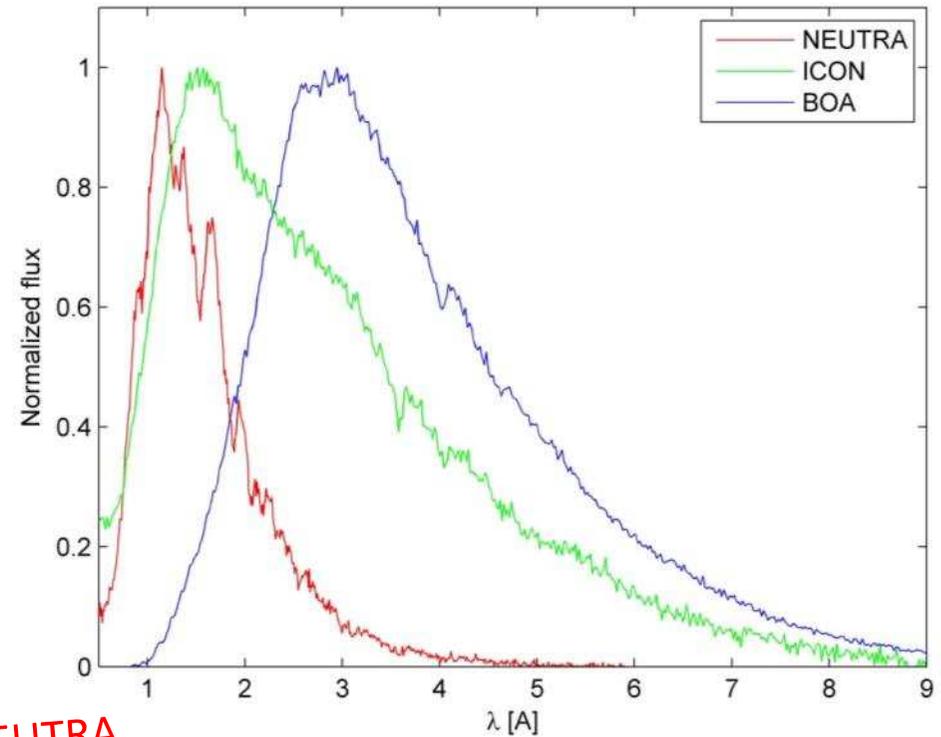
# ICON / Sektor 52 Draufsicht



## Example: SINQ



SINQ neutron source

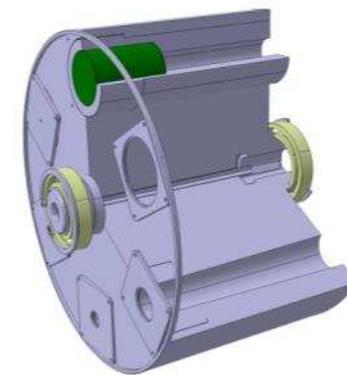
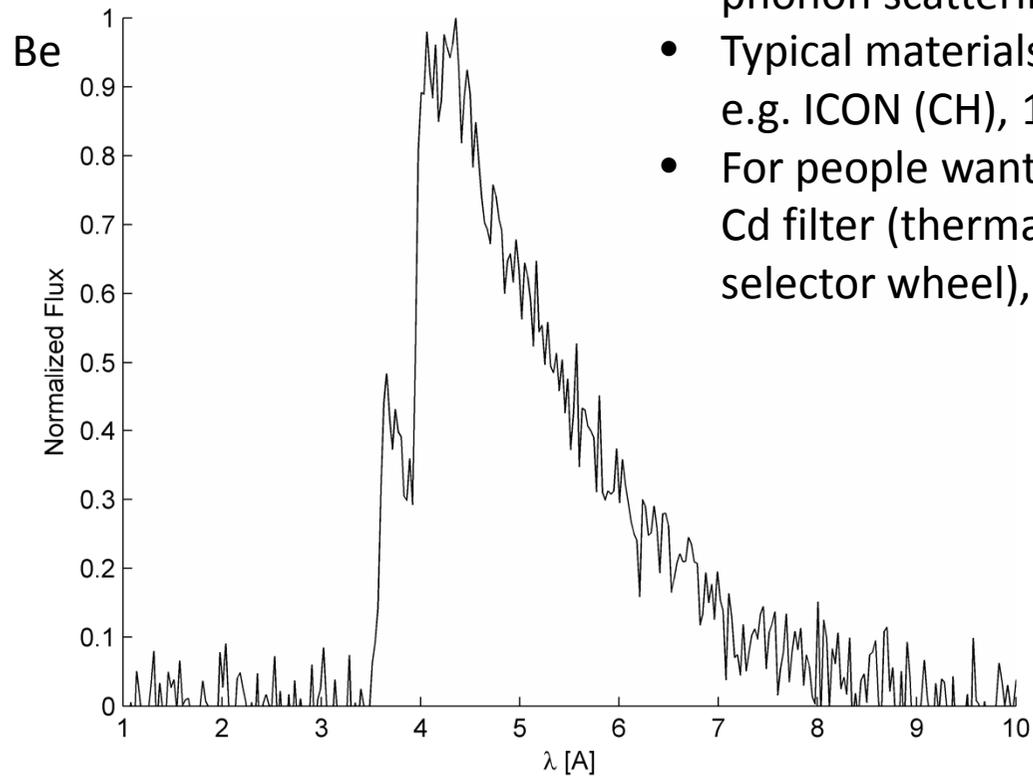


$$\int_{\lambda} I(\lambda) d\lambda = \int_{\lambda} I_0(\lambda) e^{-\Sigma(\lambda)x} d\lambda$$

If we use the full beam spectrum, we get the average **thermal** or **cold** cross-section



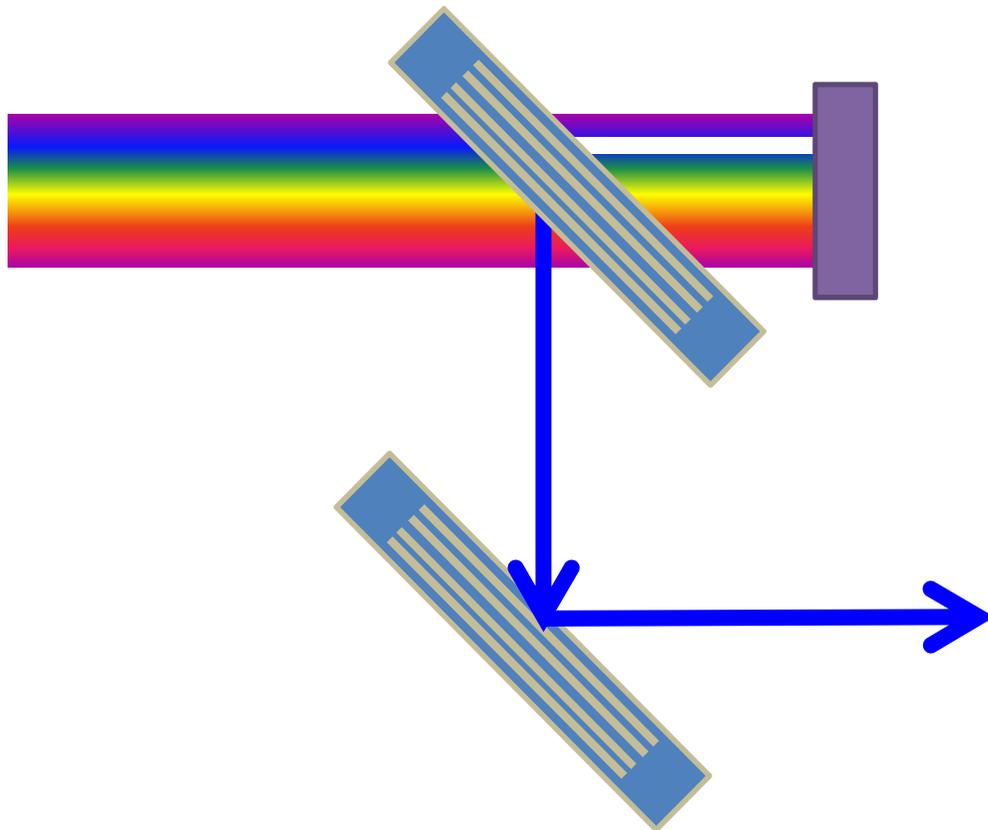
- Cross section high before the Bragg cut-off, low after
- Take a block thick enough, only neutrons  $\lambda > \lambda_{Br}$  get through
- (Cool it to limit losses above  $\lambda_{Br}$  : reduce inelastic phonon scattering)
- Typical materials: Be ( $\sim 4\text{\AA}$ ), C ( $\sim 6.7\text{\AA}$ )  
e.g. ICON (CH), 100mm in selector wheel)
- For people wanting only epithermal neutrons:  
Cd filter (thermal resonance). E.g. Antares II (D), 2mm in selector wheel), NPP control rods



ICON selector wheel



# Double crystal monochromator



## Bragg Law

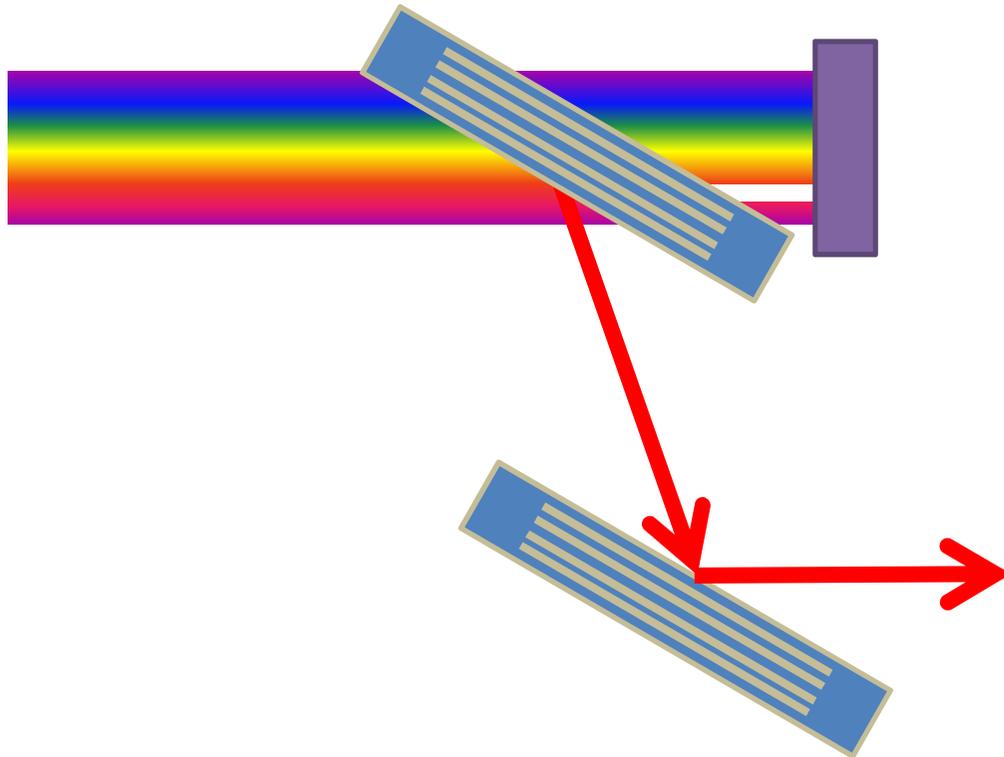
$$2d \sin \theta = n\lambda$$

Set crystal angle to  $\theta_1 \rightarrow \lambda_1$

Second crystal parallel to bring it back in the original direction



# Double crystal monochromator



## Bragg Law

$$2d \sin \theta = n\lambda$$

Set crystal angle to  $\theta_1 \rightarrow \lambda_1$

Second crystal parallel to bring it back in the original direction

Set crystal angles to  $\theta_2 \rightarrow \lambda_2$ ,  
+ translate

Typical crystals: PG002, Ge, Si

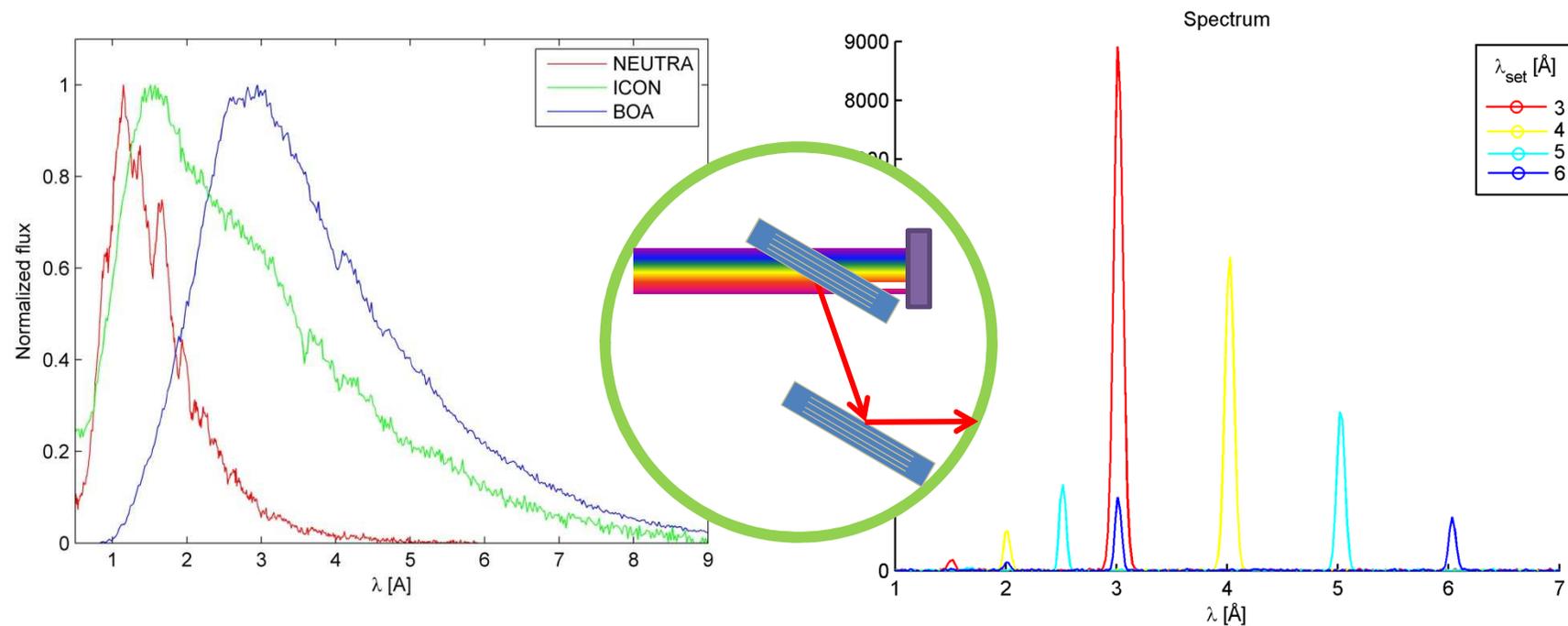


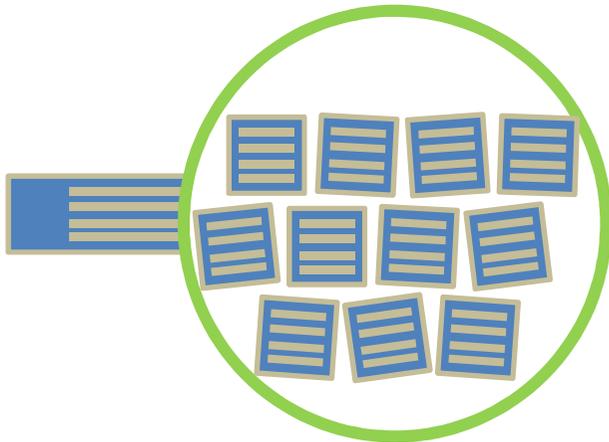
Higher order contributions (n=2,3,... reflections)

$$2d \sin \theta = n\lambda$$

Crystal set to  $\theta \rightarrow \lambda$ , also  $\lambda/2$  (n=2),  $\lambda/3$  (n=3)...

Solution: PG002 + Be-filter / Ge, Si: 2<sup>nd</sup> order forbidden (F=0)





Single crystal  
= small blocks ( $\sim 100\mu\text{m}$ ) of small misorientation

This is called **mosaicity**

So  $\theta$  and hence  $\lambda$  are no  $\delta$ -function

The better the monochromaticity, the less neutrons you get on your detector!  
(So the longer the exposure time for one image, let alone a tomography)

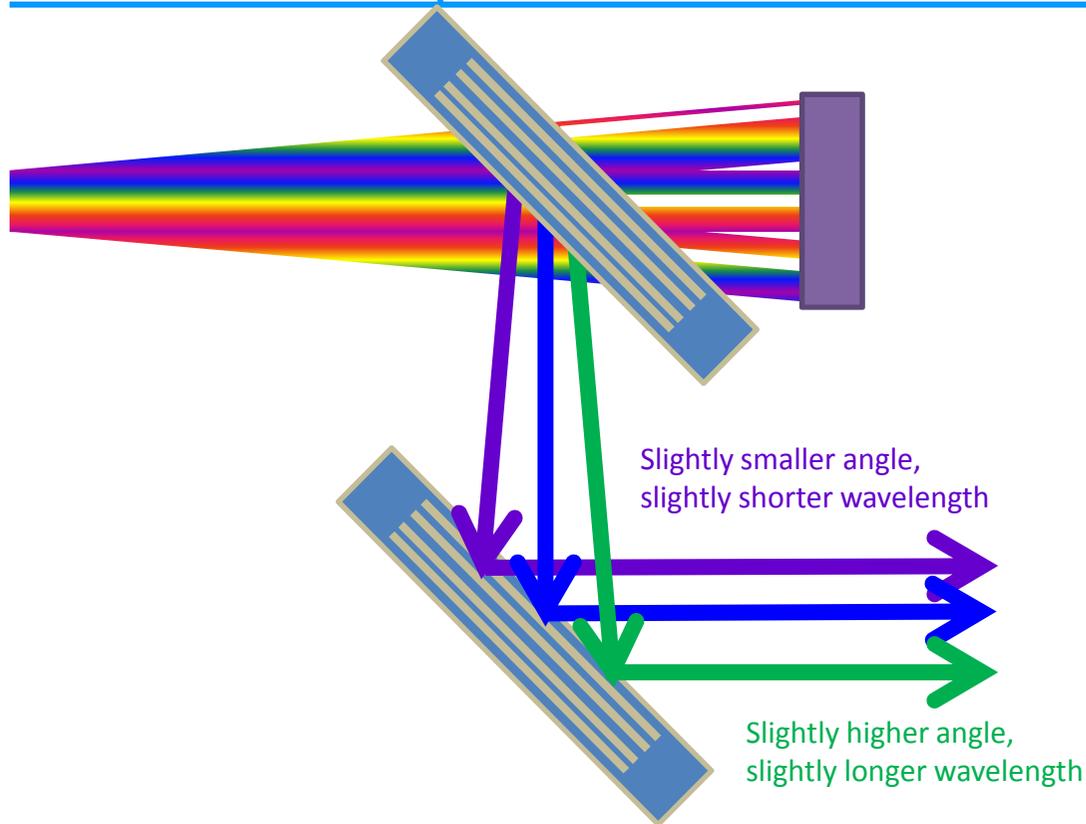
Si, Ge: too perfect ( $\Delta\lambda/\lambda < 0.01\%$ ), hot bending to introduce defects

PG002: Commercially available, only ordered in the [002] direction, the choice in imaging

Imaging:

- Mosaic spread  $\sim 0.4^\circ$
- Monochromaticity  $\Delta\lambda/\lambda \sim 1-3\%$

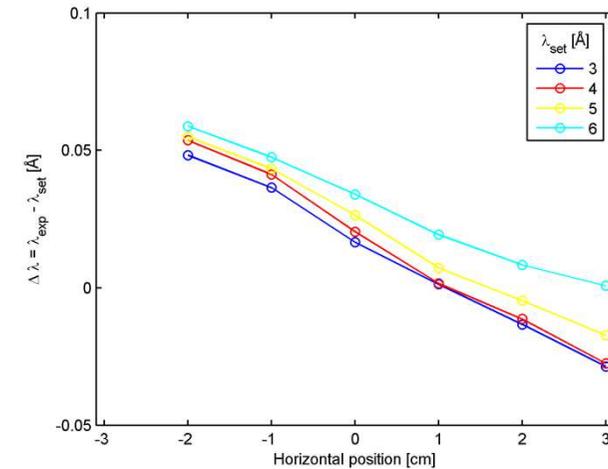




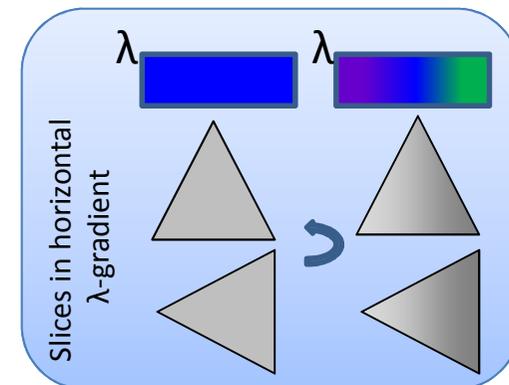
Imaging implication:

- Crystals horizontal: same wavelength within the slice, vertical gradient across sample
- Crystals vertical: broader effective wavelength band, vertically homogeneous

Beam hitting left vs right side FOV:  
 $\Delta\theta \sim 0.5^\circ$  (+5 to -5cm FOV @ 5m)

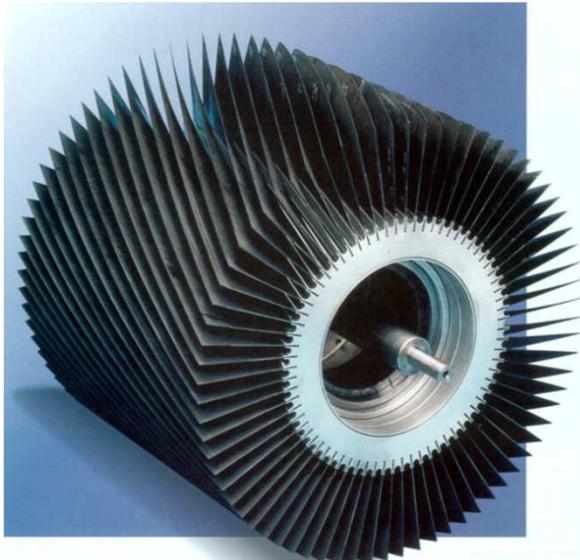


Morgano, M.; Peetermans, S.; Lehmann, E.; Panzner, T. & Filges, U.  
Neutron imaging options at the BOA beamline at Paul Scherrer Institut  
*Nuclear Instruments and Methods, section A*, **2014**, 754, 46-56



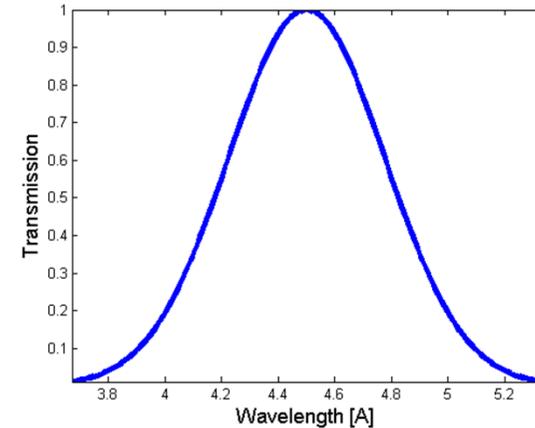
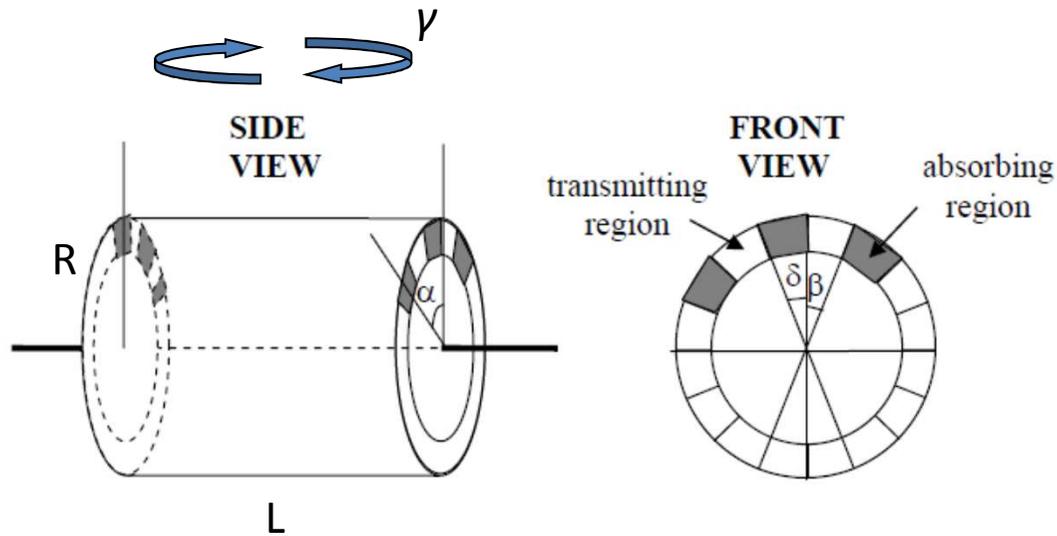
$\Delta\lambda/\lambda=15\%$

- + Relatively high count rates
- + ES Imaging past Bragg cut-off
- Sharp Bragg edge imaging



[http://www.sii.co.jp/jp/segg/files/2013/03/file\\_PRODUCT\\_MASTER\\_1381\\_GRAPHIC02.pdf](http://www.sii.co.jp/jp/segg/files/2013/03/file_PRODUCT_MASTER_1381_GRAPHIC02.pdf)





Central wavelength

$$\lambda = \frac{\alpha h}{L m_n \omega}$$

Wavelength Spread

$$\frac{\Delta\lambda}{\lambda} = \frac{\beta}{\alpha}$$

Tilt w.r.t. the beam

$$\alpha_{eff} = \alpha + \gamma \frac{L}{R}$$

Beam divergence  $\longleftrightarrow$  Velocity selector tilt  $\longrightarrow$  Spectral shift over the FOV!

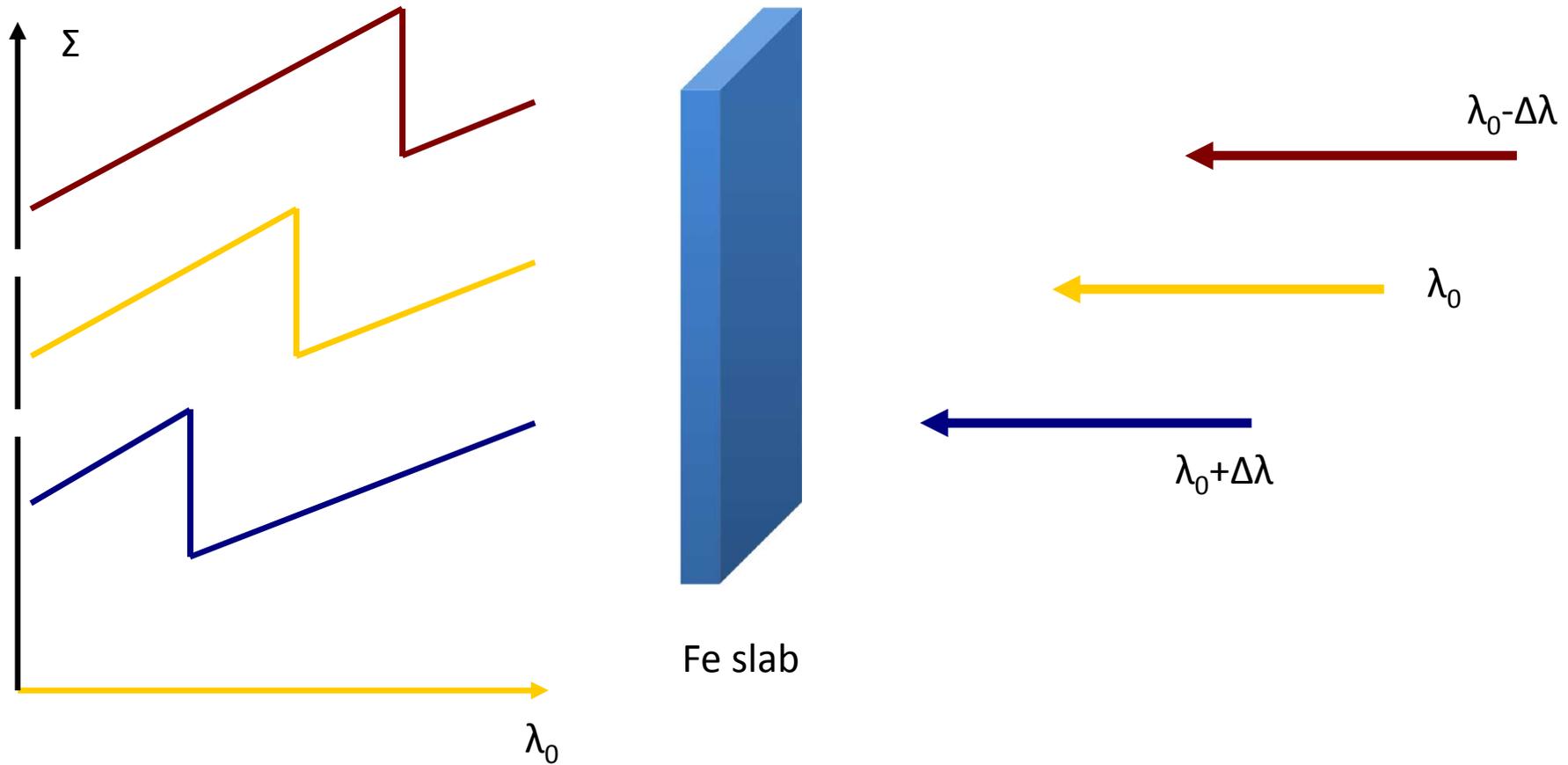
Case of ICON:  $L=400\text{mm}$ ,  $R_c=82.5\text{mm}$ ,  $\alpha=12.5^\circ$ ,  $\beta=1.6^\circ$ ,  $\gamma=0.5^\circ$ ,  $L/D=343$

Shift up to  $1\text{\AA}$  over the Field Of View!

Image taken from B. Hammouda, Probing nanoscale structures – the SANS toolbox



# Measuring the wavelength shift

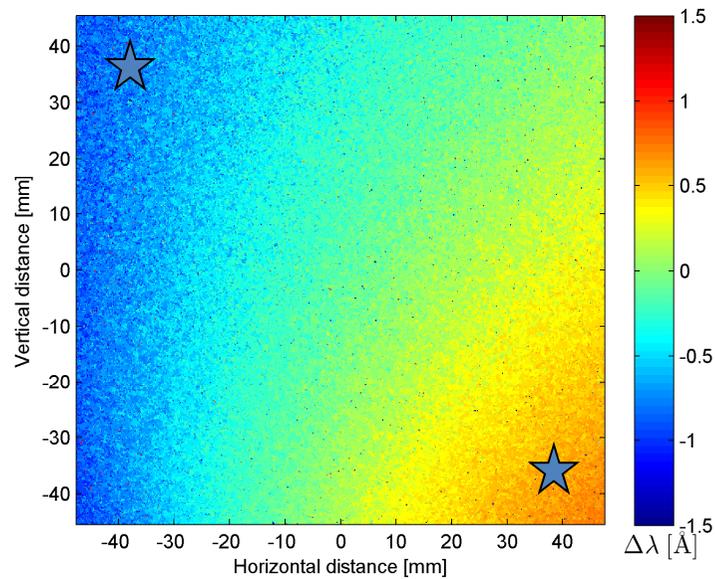


Calibration energy-scan of an iron plate (5mm)

A shift in the incoming spectrum will also induce a shift in the observed Bragg edge position

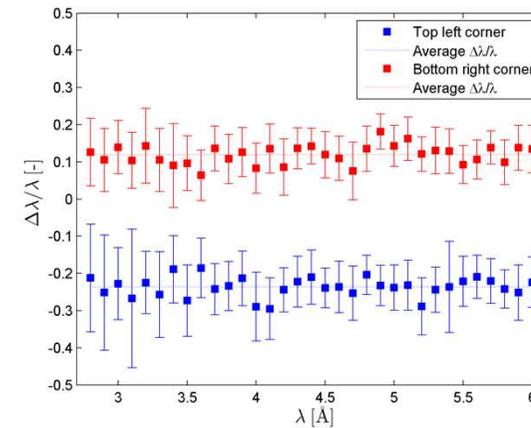


But know the shiftmap at one wavelength, you know it for all



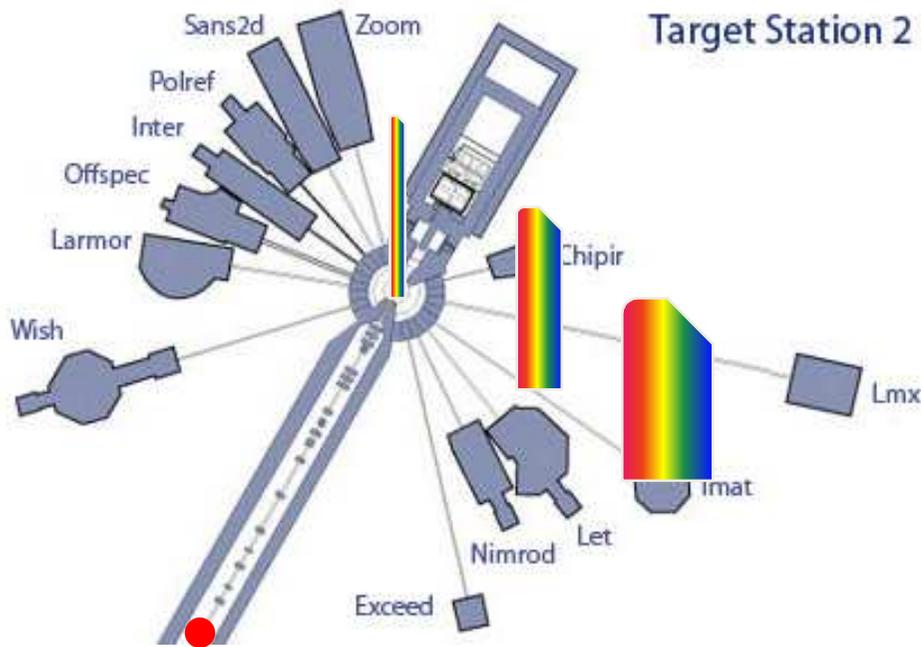
$$\frac{\Delta\lambda_1}{\lambda_1} = \frac{\gamma L}{\alpha R}$$

$$\frac{\Delta\lambda_1}{\lambda_1} = \frac{\Delta\lambda_0}{\lambda_0}$$



Peetermans, S.; Grazi, F.; Salvemini, F. & Lehmann, E.  
Spectral characterization of a velocity selector type monochromator for energy-selective neutron imaging  
*Physics Procedia*, **2013**, *43*, 121-127

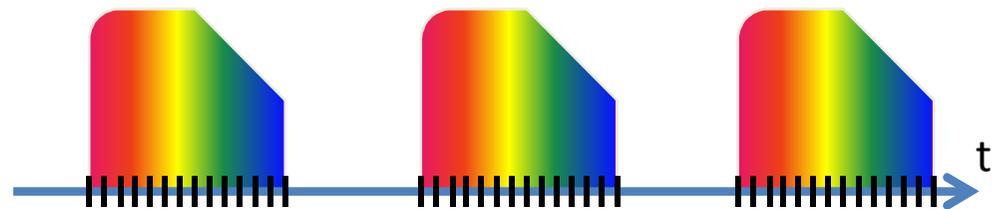




ISIS TS2 lay out (UK)

Pulsed spallation source  
e.g. ISIS runs at 50Hz

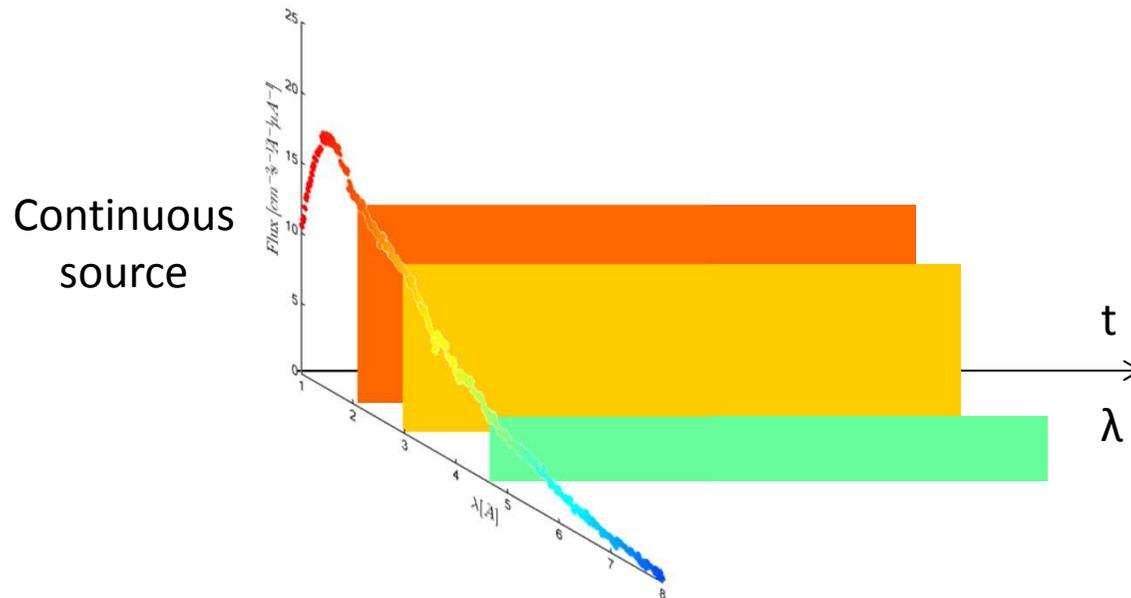
Record flight time  
→ Neutron velocity  
→ Neutron energy/wavelength



Each short frame ( $\mu\text{s}$ ) is a monochromatic image  
Stack frames (pulses) for sufficient statistics  
Whole spectrum recorded at once  
Very high energy resolution  
Limited by initial pulse width

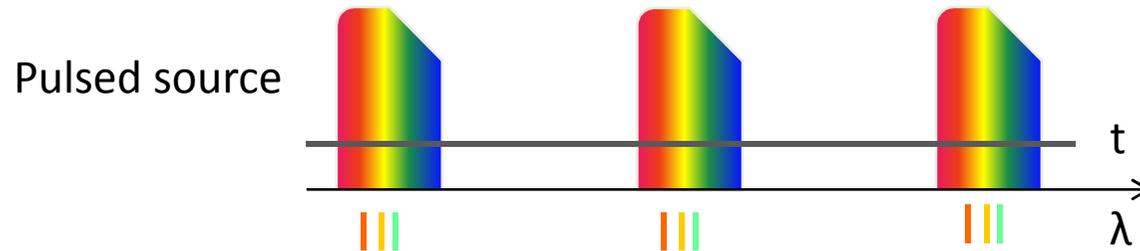


What if you don't want a full spectrum, just work with a single monochromatic wavelength?



All wavelengths are continuously provided for

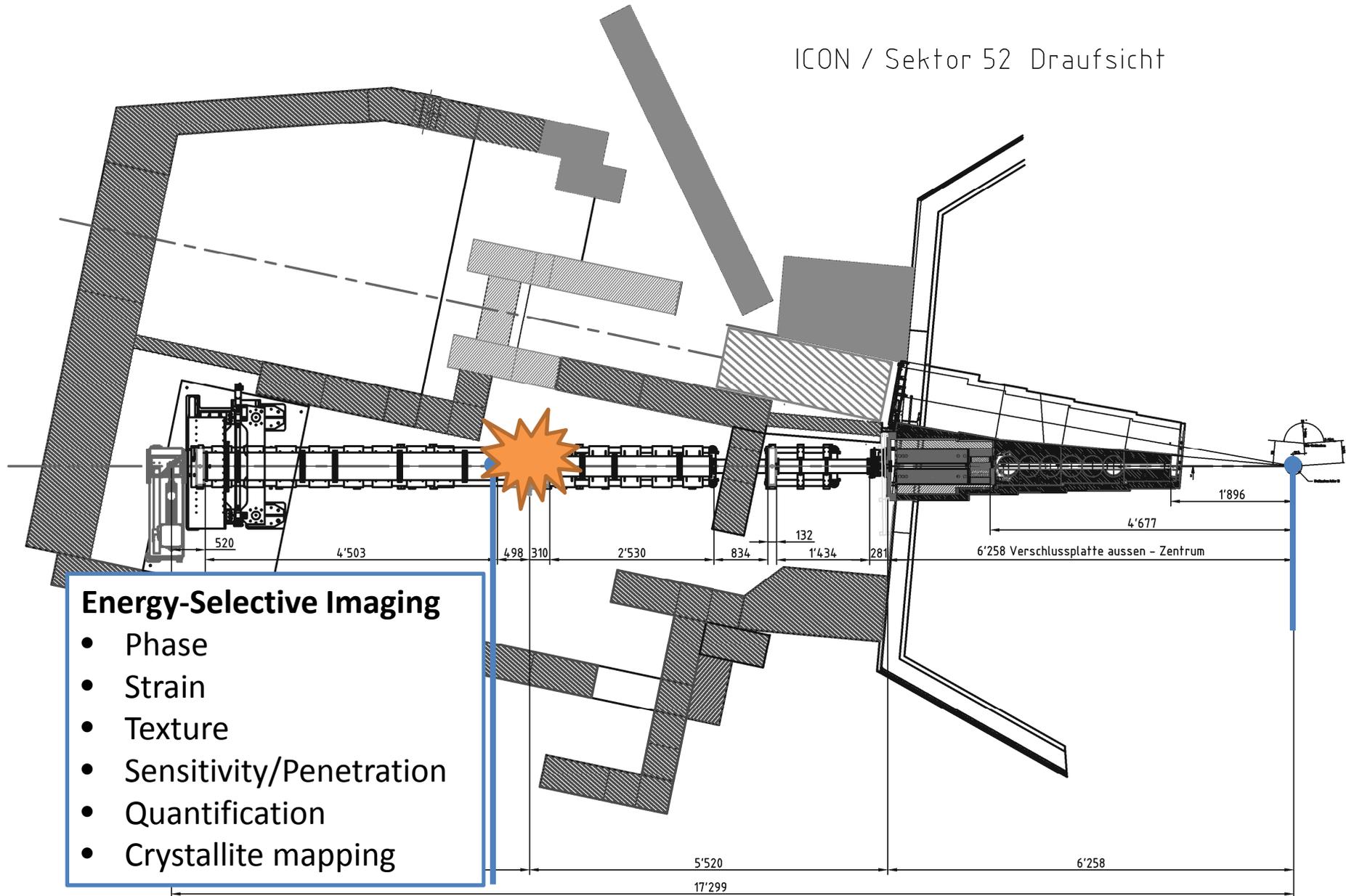
Select one with monochromator

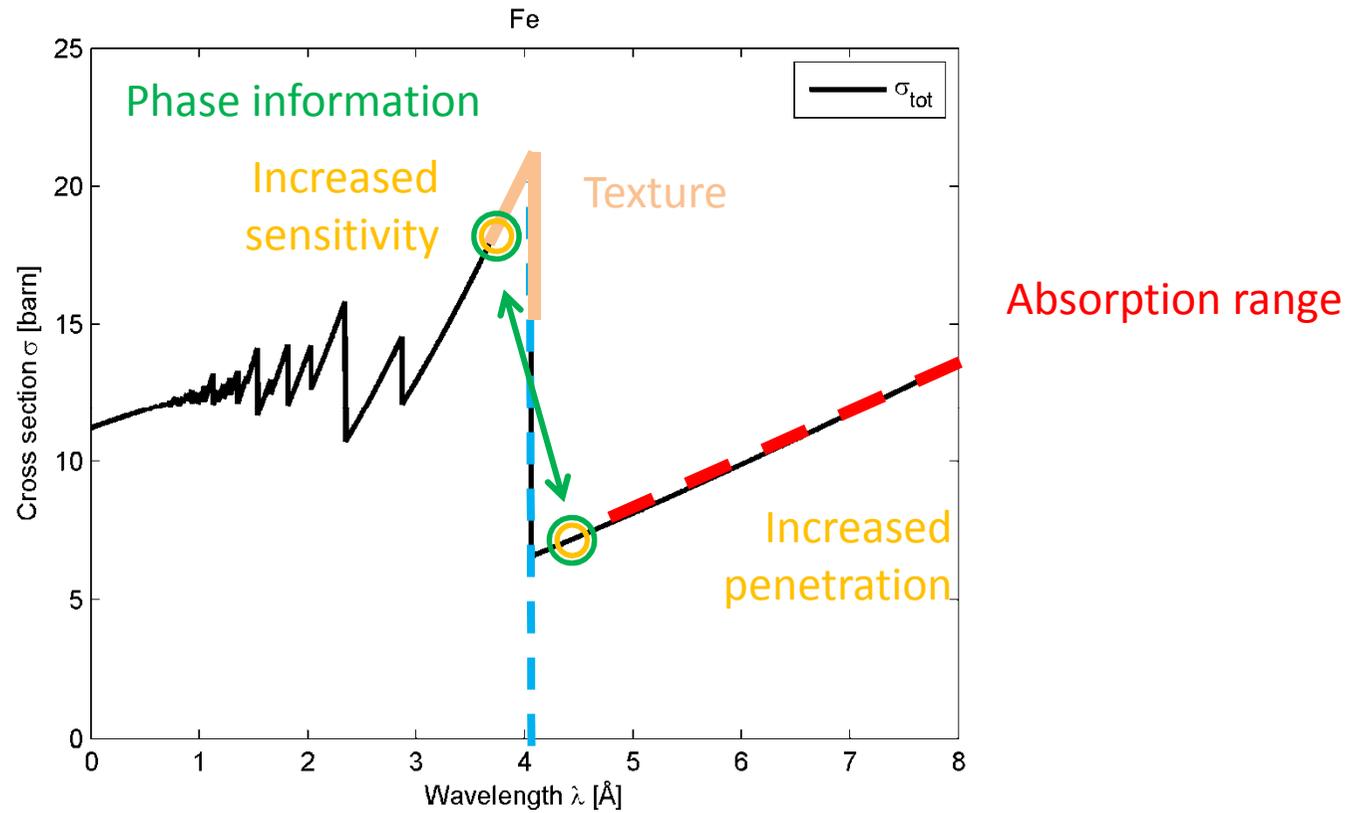


Record your frame  
Wait until next pulse

Traditional white beam: time integrated flux important







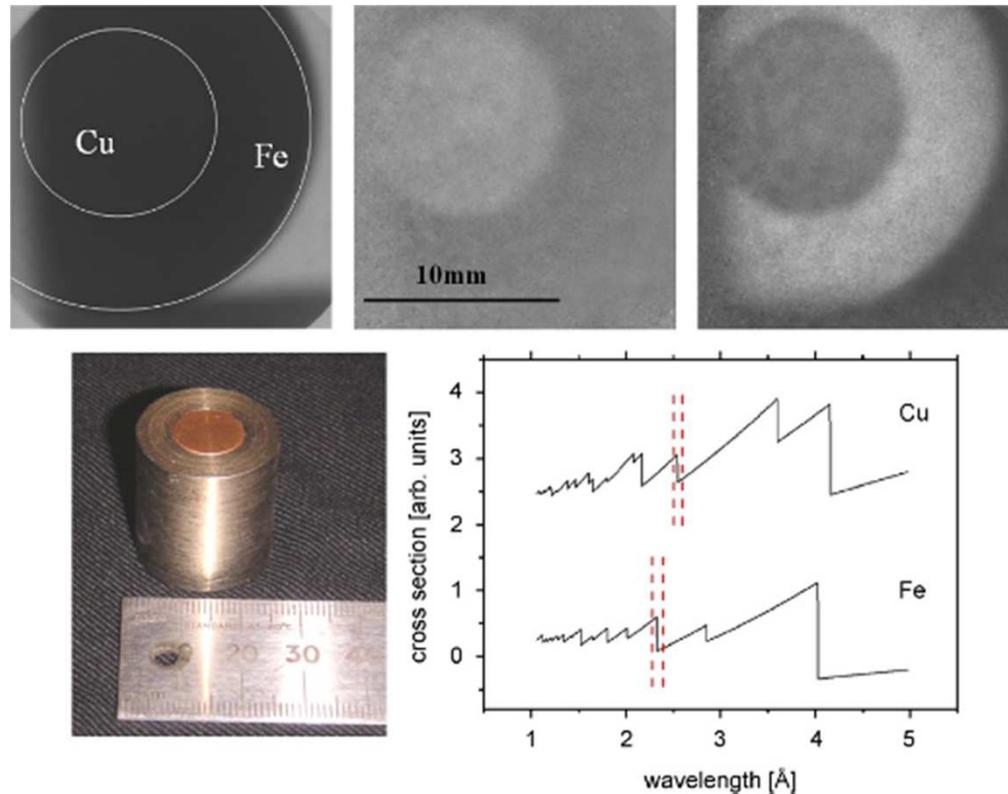
Strain Imaging



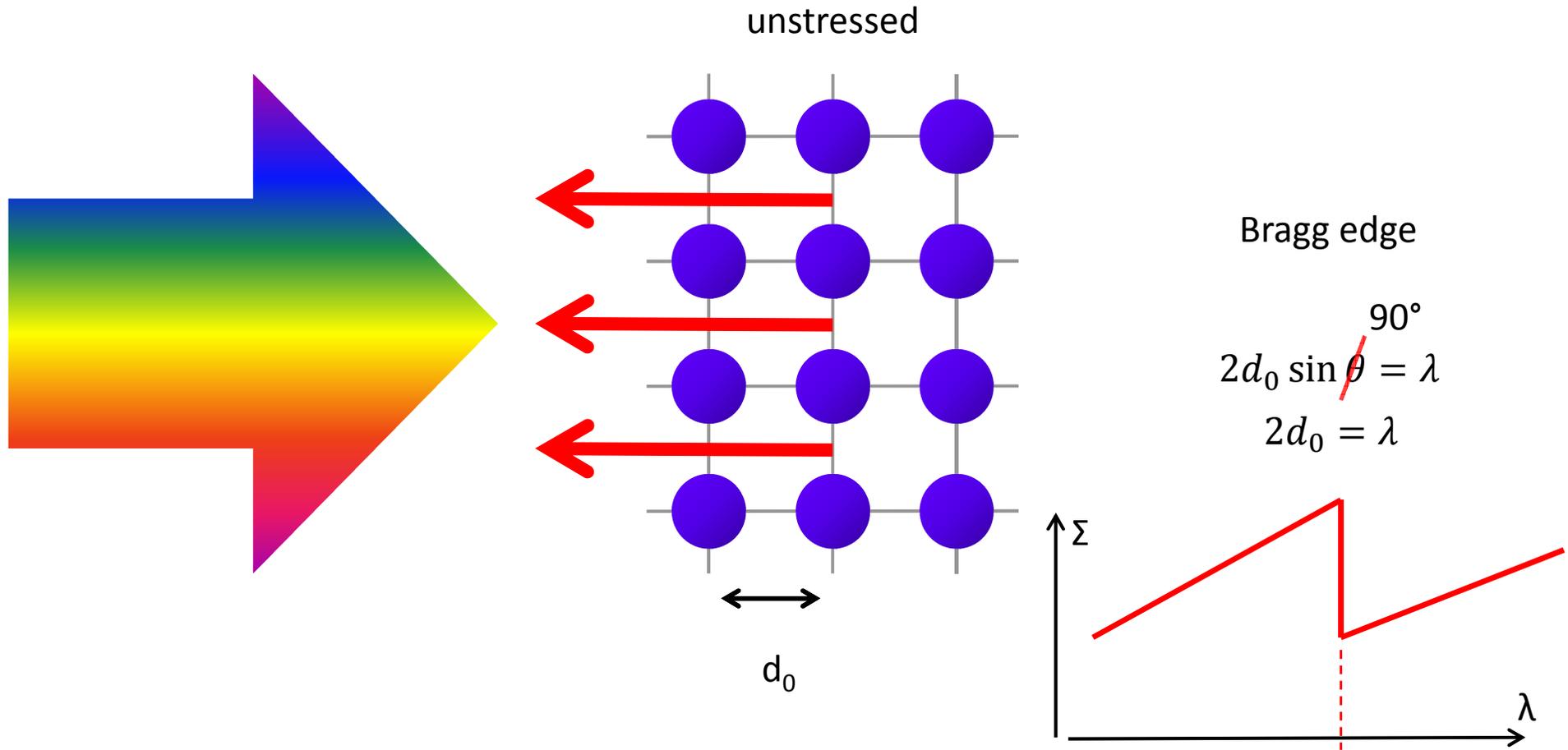
Imaging just before and just after the Bragg edge for Copper

→  $\Delta\Sigma_{\text{Cu}}$  large,  $\Delta\Sigma_{\text{Fe}}$  small (its Bragg edge is elsewhere)

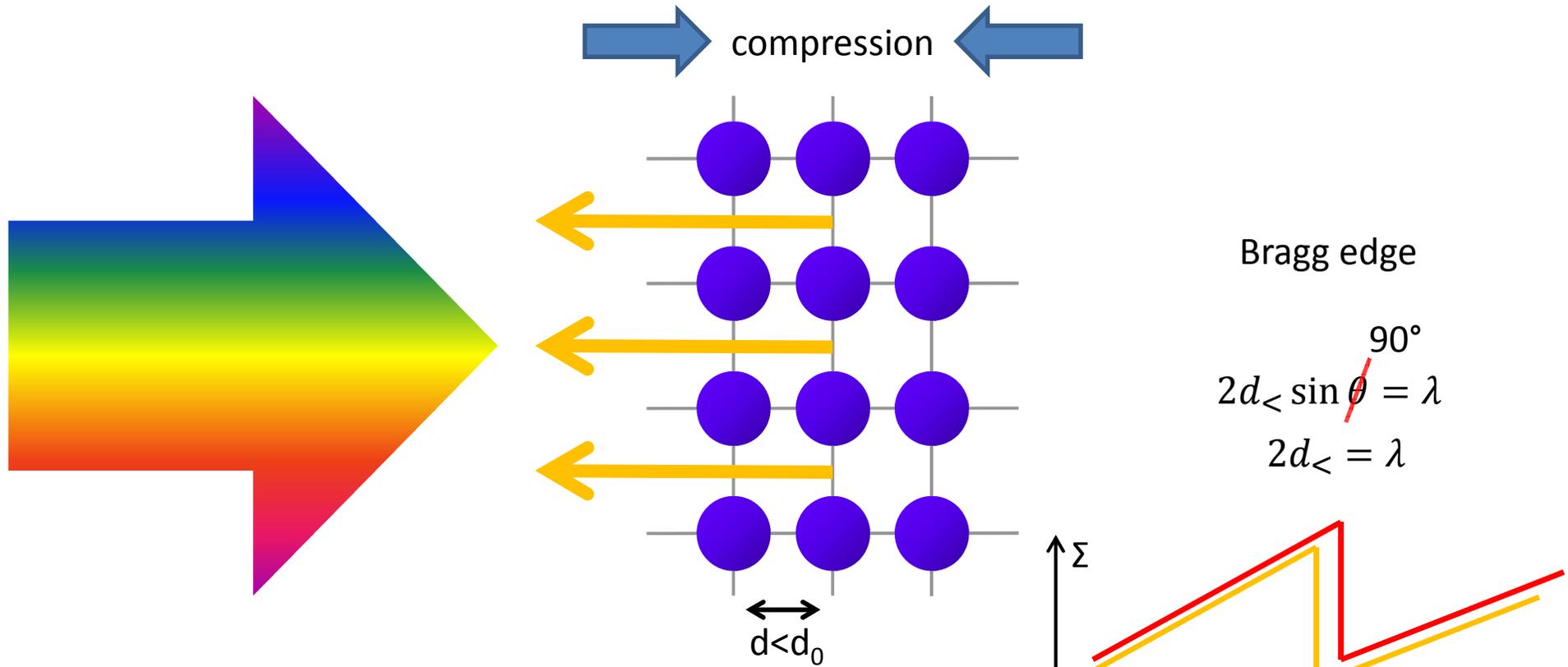
Reconstruction of  $\Delta\Sigma$  provides phase mapping



# Bragg edges - unstressed



# Bragg edges - compression

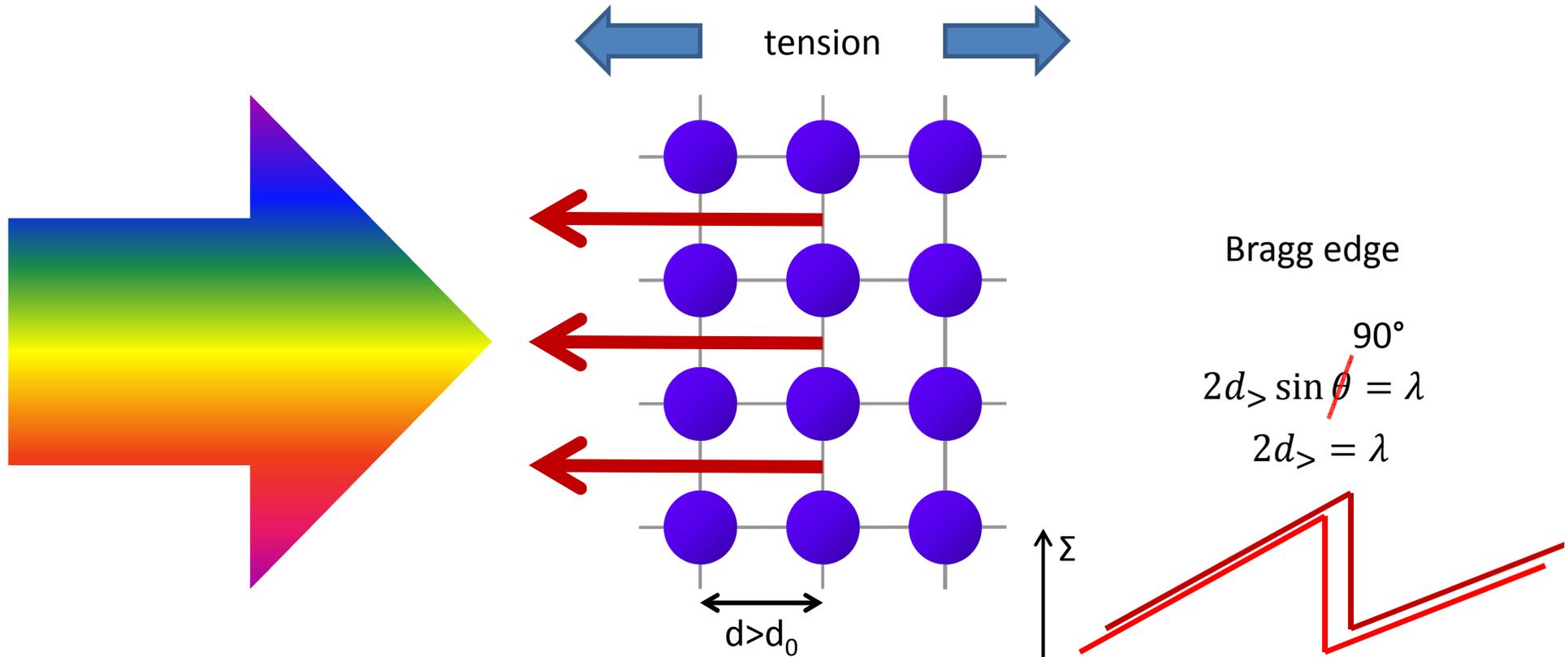


Strain  $\epsilon = \frac{d - d_0}{d_0}$

Stress  $\sigma = E\epsilon$



# Bragg edges - tension

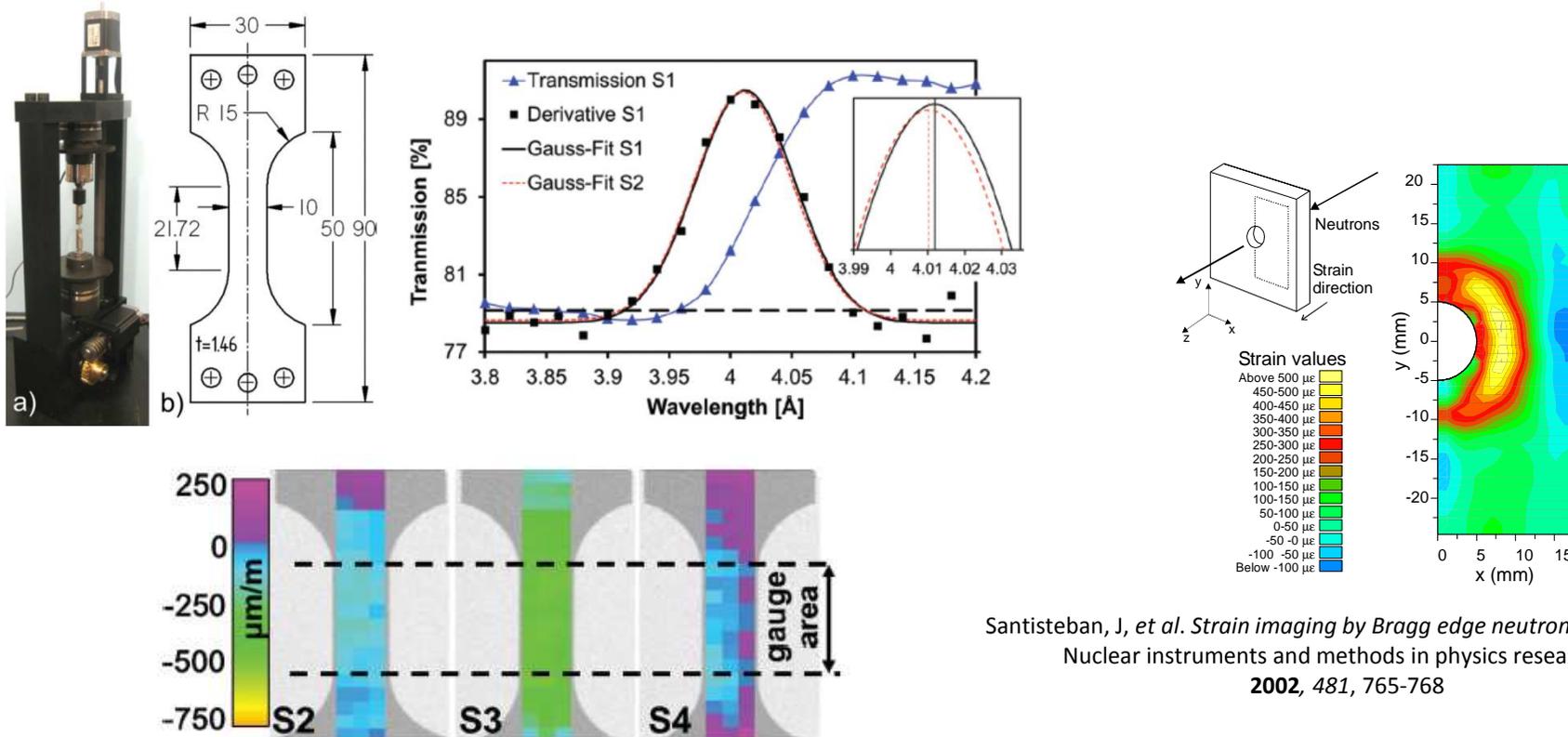


Strain  $\epsilon = \frac{d - d_0}{d_0}$

Stress  $\sigma = E\epsilon$



Stressing a dogbone sample and perform energy scan  
Fit of derivative for increased accuracy in Bragg edge position

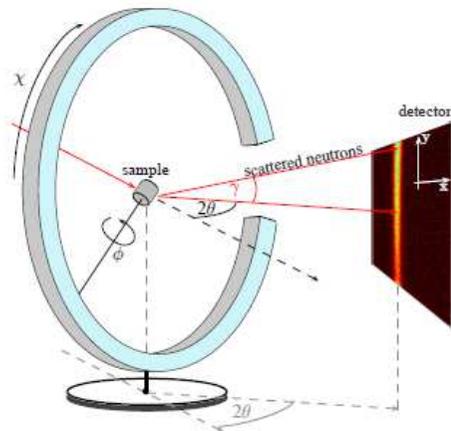


Santisteban, J, et al. *Strain imaging by Bragg edge neutron transmission*  
Nuclear instruments and methods in physics research A,  
**2002**, 481, 765-768

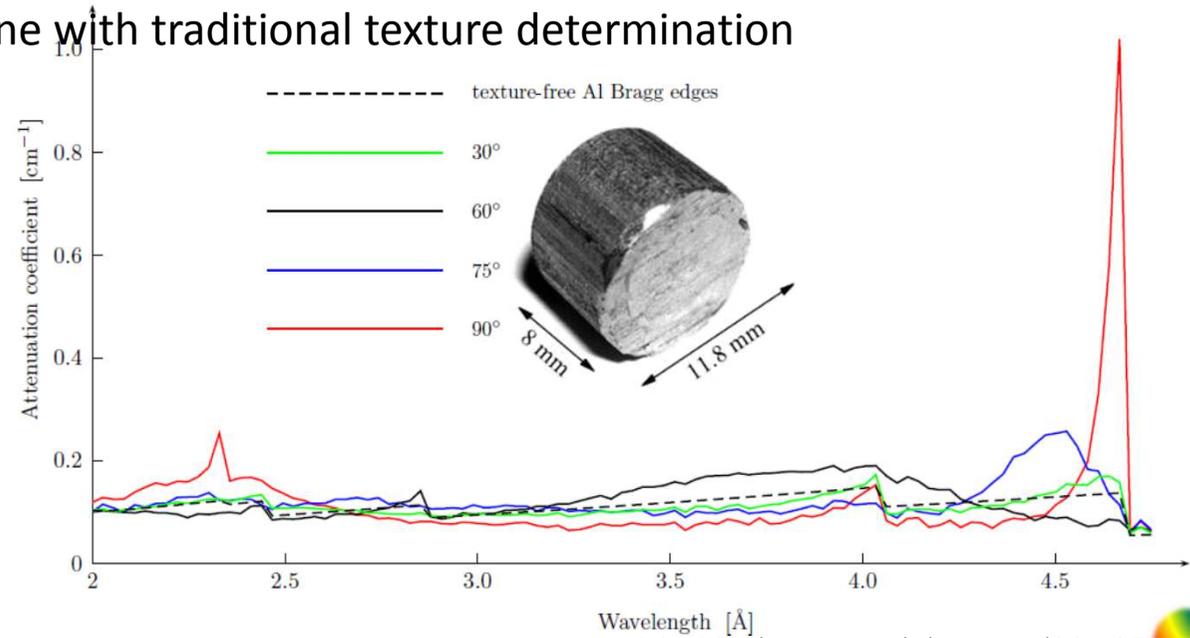
Woracek, R.; Penumadu, D.; Kardjilov, N.; Hilger, A.; Strobl, M.; Wimpory, R. C.; Manke, I. & Banhart, J.  
Neutron Bragg-edge imaging for strain mapping under in situ tensile loading  
*Journal of Applied Physics*, **2011**, 109, 093506-1 - 093506-4



- No longer all random orientations
- Some orientations occur more than others (preferred orientation / texture)
- Corresponding wavelengths will be scattered out more, others less
- Deformed Bragg edge pattern
- Rotation dependent – be careful in tomography
- Rather qualitative (only March-Dollase texture can be treated mathematically)
- But spatial resolved – combine with traditional texture determination



Experimentally determining diffraction strength for each sample orientation in eulerian cradle



Bragg edge pattern for a textured aluminium cylinder

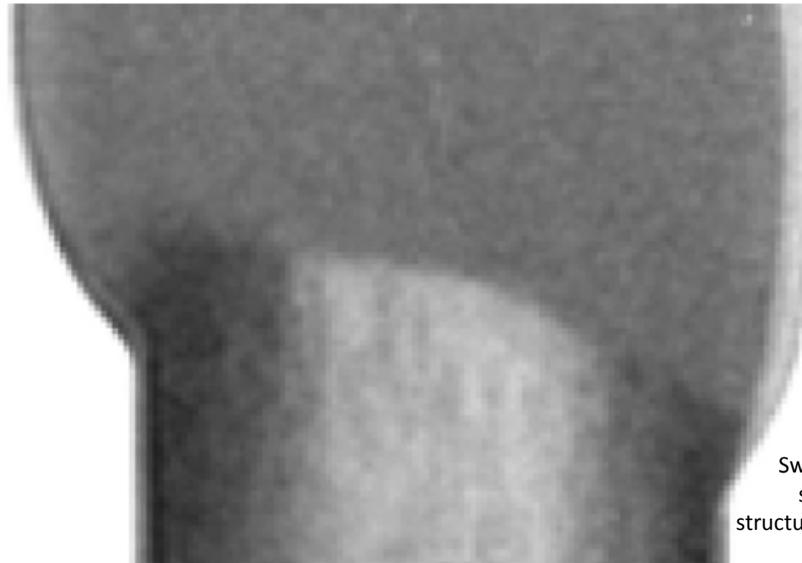
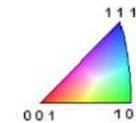
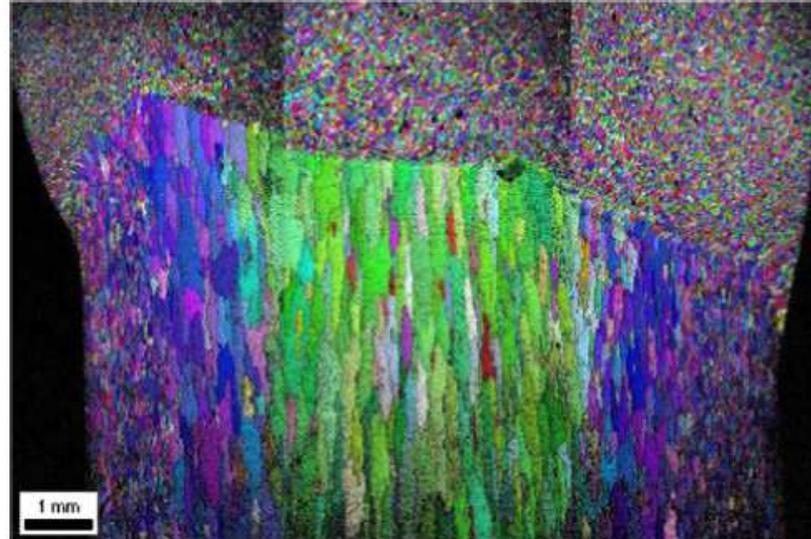
Boin, M. Developments towards the tomographic imaging of crystallographic structures. *The Open University*, 2011

# Texture example

Weld on  
rolled aluminium

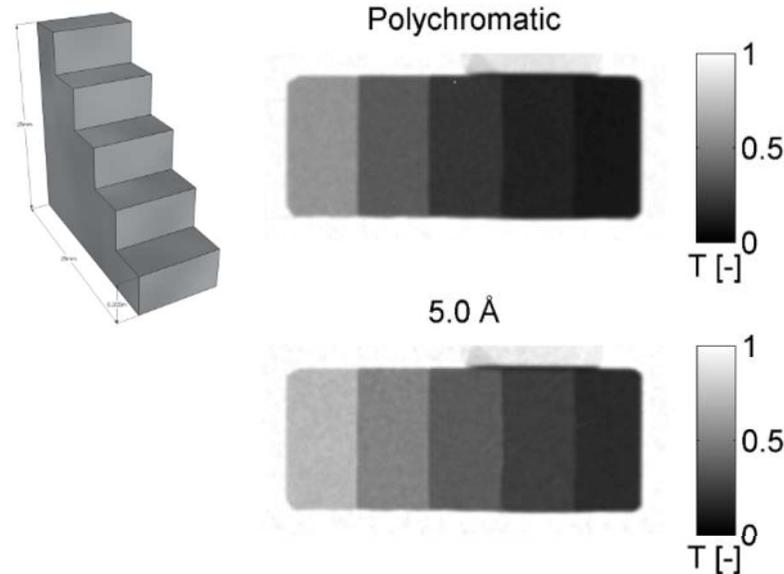
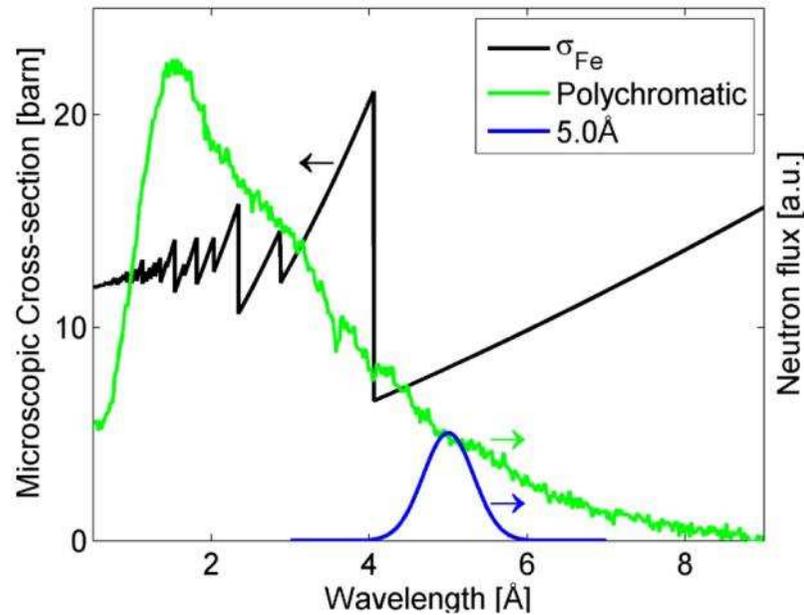
Orientation map from  
electron microscopy

Surface mapping  
~ 1day

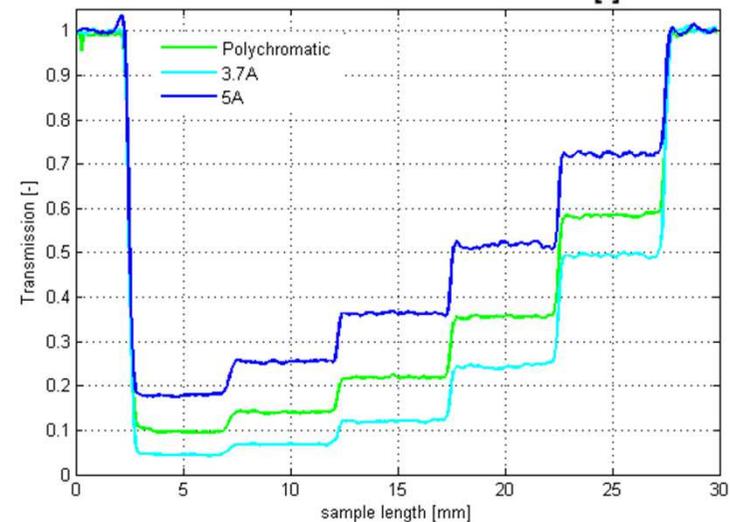


Radiograph at 4.6Å  
Bulk, ~1min

Lehmann, E.; Peetermans, S.; Josic, L.; Leber, H. & van Swygenhoven, H. Energy-selective neutron imaging with high spatial resolution and its impact on the study of crystalline-structured materials. *Nuclear Instruments and Methods in Physics Research, Section A*, **2014**, 735, 10.



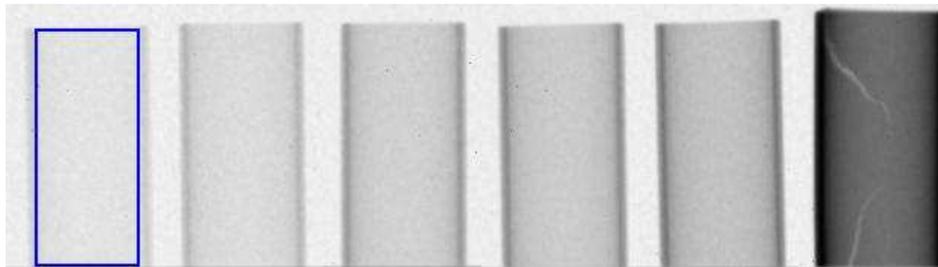
- Increased transmission just past the Bragg cut-off
- Maximum attenuation just before the Bragg cut-off: highest sensitivity to small amounts



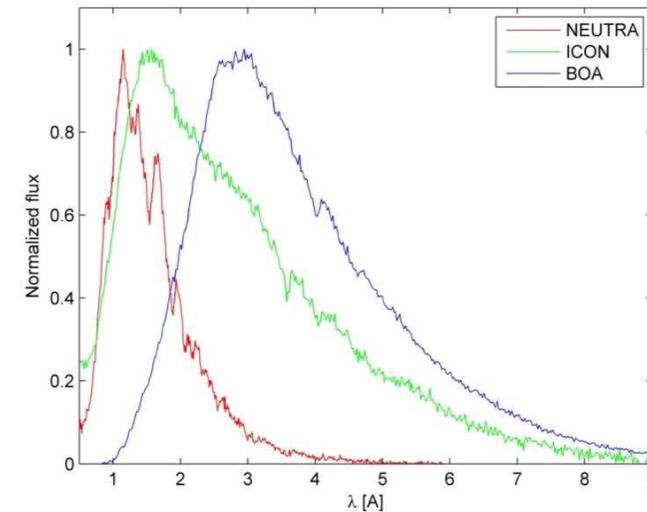
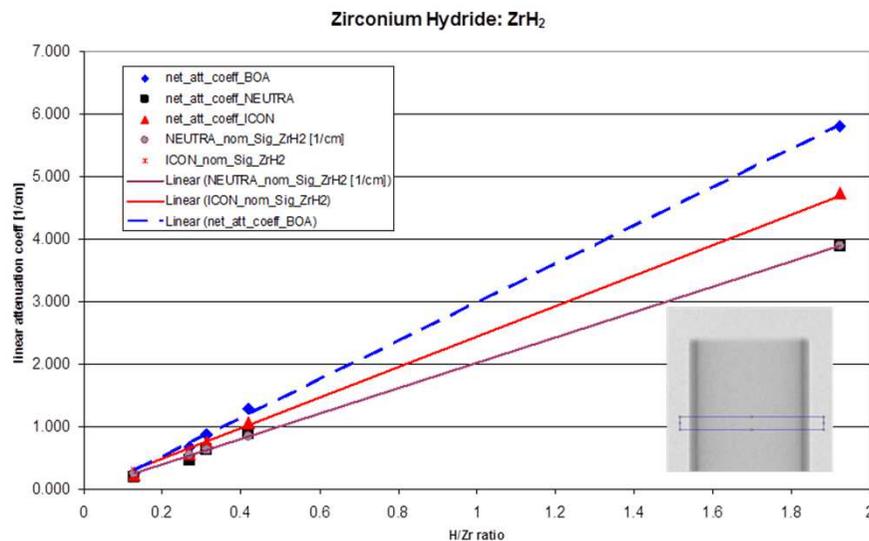
# Sensitivity example

Sample	H/Zr ratio
8	0
9	0.127
10	0.268
11	0.312
12	0.418
13	1.924

- Hydrogen content in Zirconium (LOCA accidents NPP, e.g. Fukushima)
- The colder the spectrum, the more sensitive to small amounts of H:  
NEUTRA – ICON – BOA

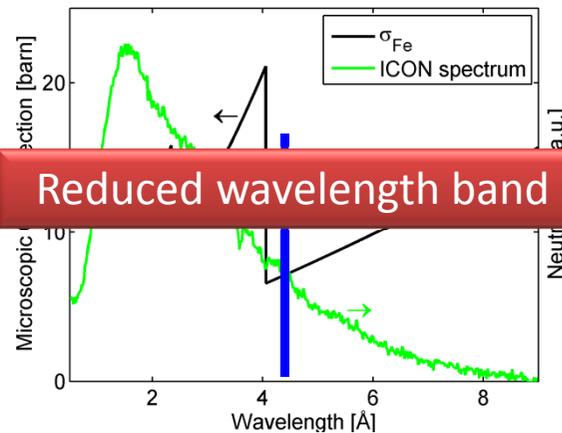


**Samples:** M. Grosse, KIT  
**Experiment:** P. Vontobel, A. Kaestner, S. Peetermans, T. Panzner, PSI  
**Analysis:** P. Vontobel, PSI



## White beam problems...

### Beam hardening



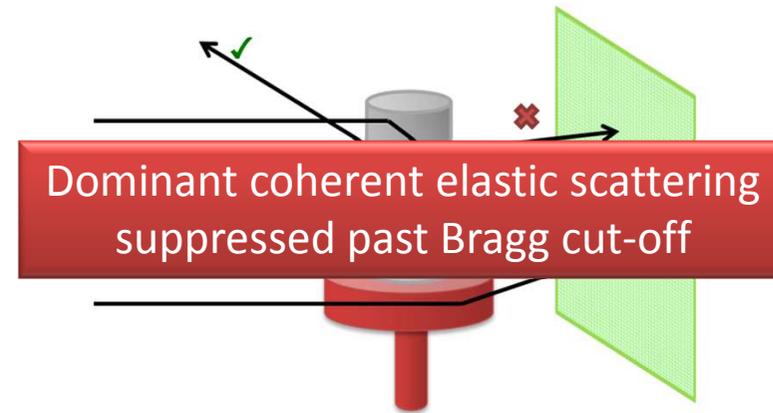
Short wavelengths attenuated more

Thickness increases

→ colder beam

→ lower effective cross-section

### Scattering contributions

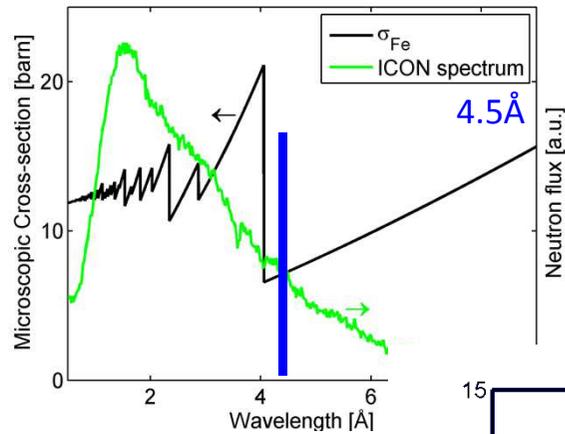


Scattered neutrons still hit the detector behind the sample

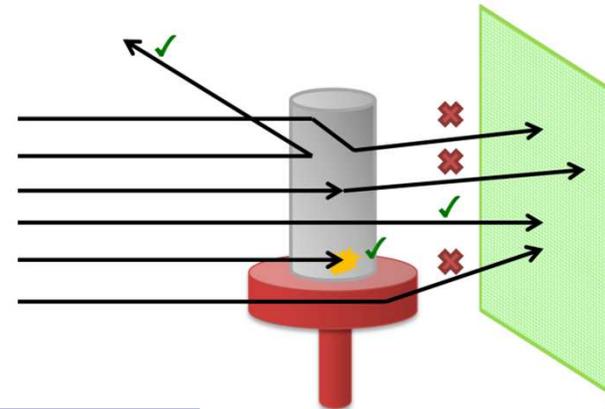
Transmission through sample overestimated  
Cross-section underestimated

## White beam problems

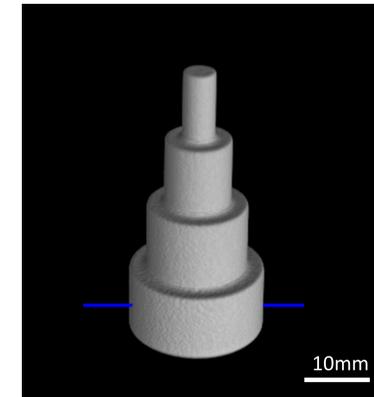
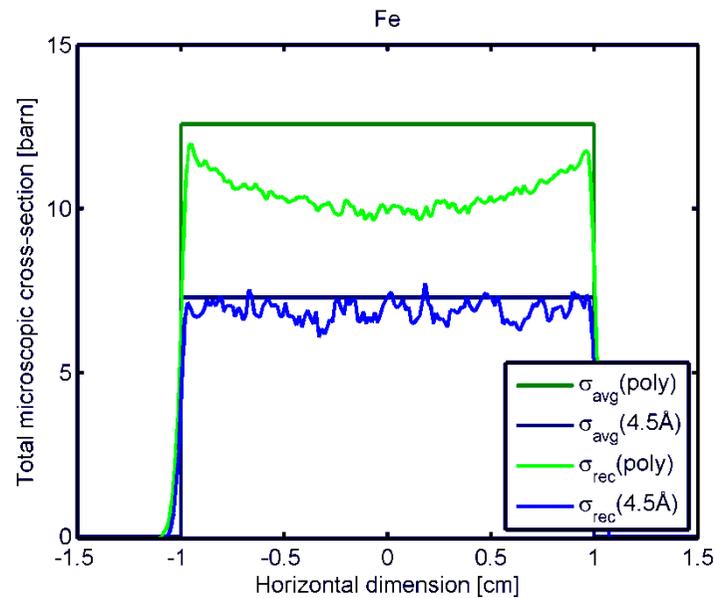
### Beam hardening



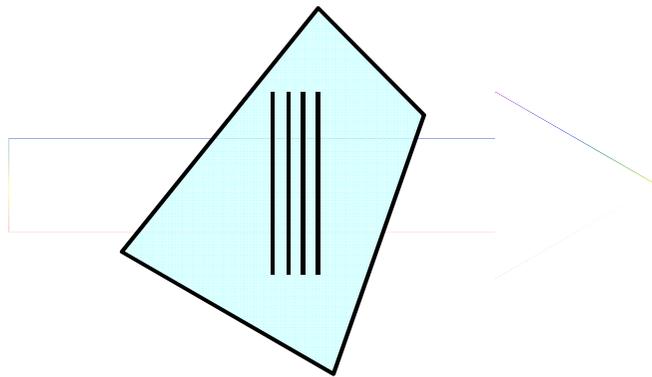
### Scattering contributions



Iron step cylinder  
Tomography  
(polychromatic/4.5 Å)  
ø 20mm  $\sigma$ -profile



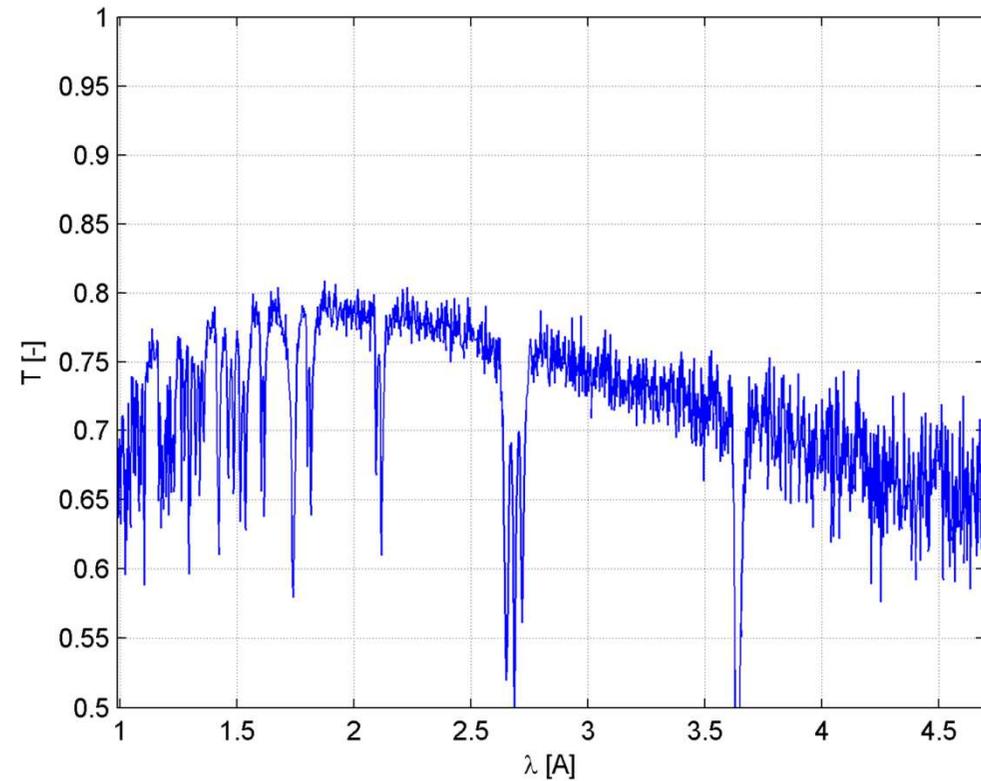
$$2d_{hkl} \sin \theta = \lambda_{hkl}$$



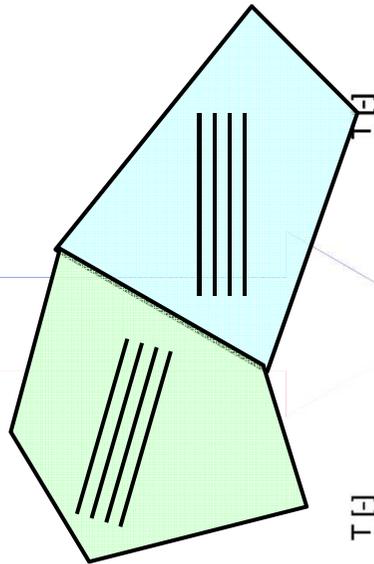
Single crystal: Bragg dips

Position: crystallographic phase,  
crystal orientation

Width: crystal quality (mosaicity)

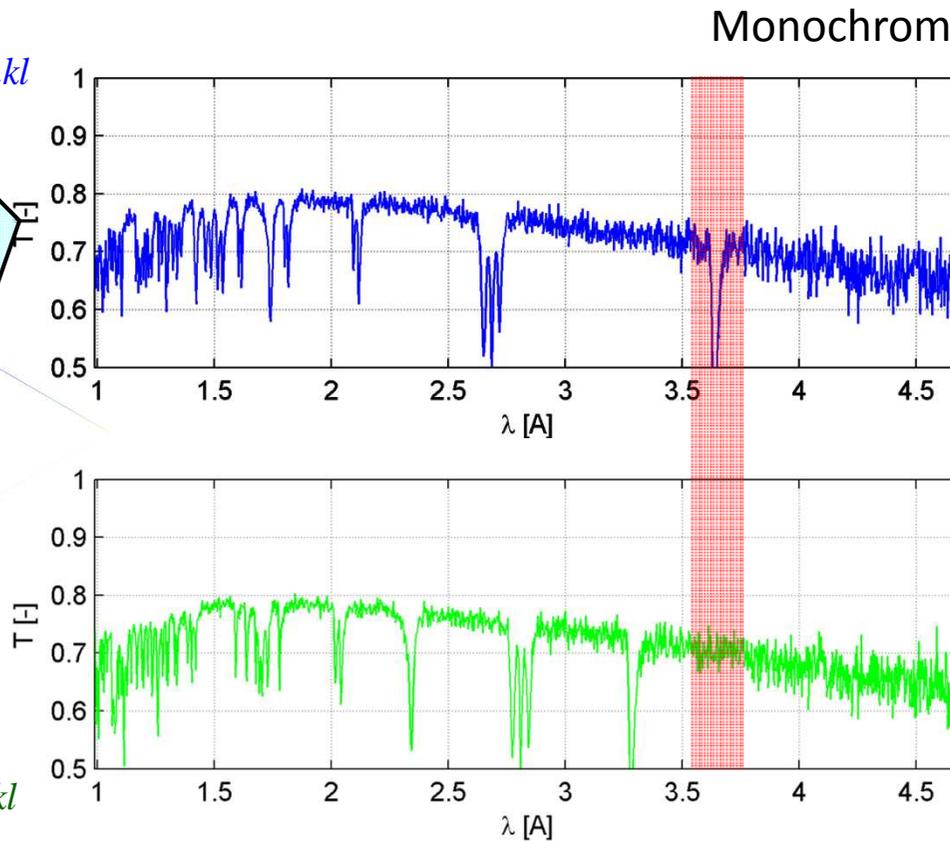


$$2d_{hkl} \sin \theta_1 = \lambda_{hkl}$$



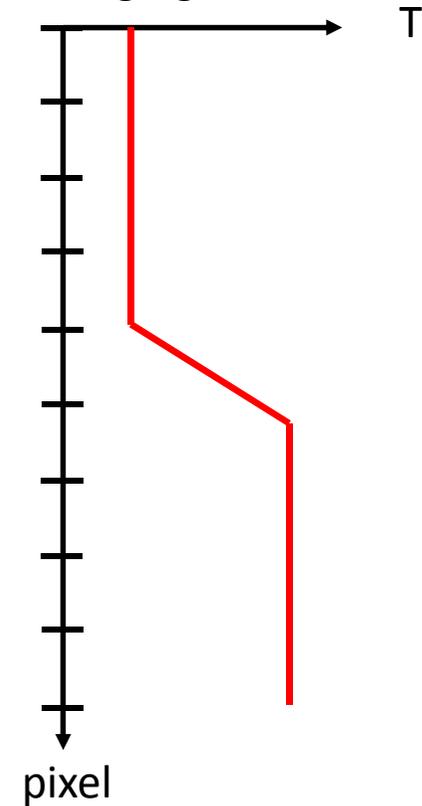
$$2d_{hkl} \sin \theta_2 = \lambda_{hkl}$$

Add 2nd crystal



Different orientation,  
Different Bragg dip location

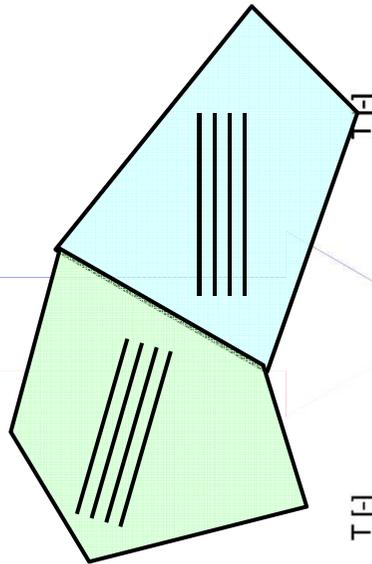
Monochromatic imaging



Crystallite contrast  
if above spatial  
resolution

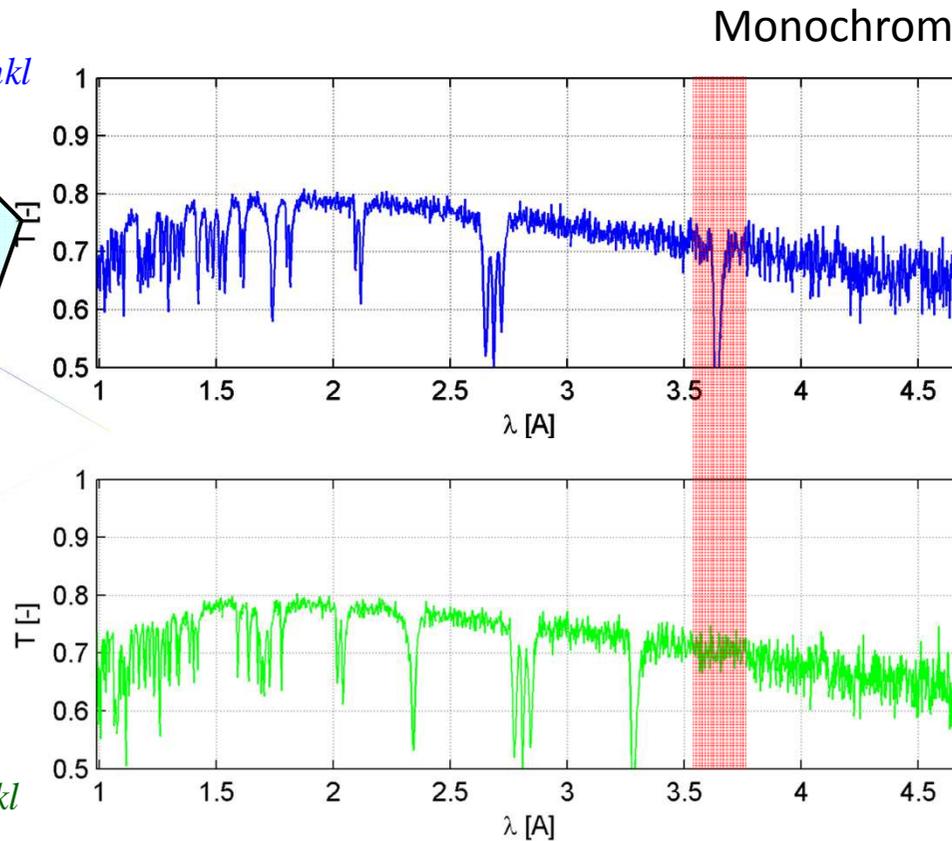


$$2d_{hkl} \sin \theta_1 = \lambda_{hkl}$$



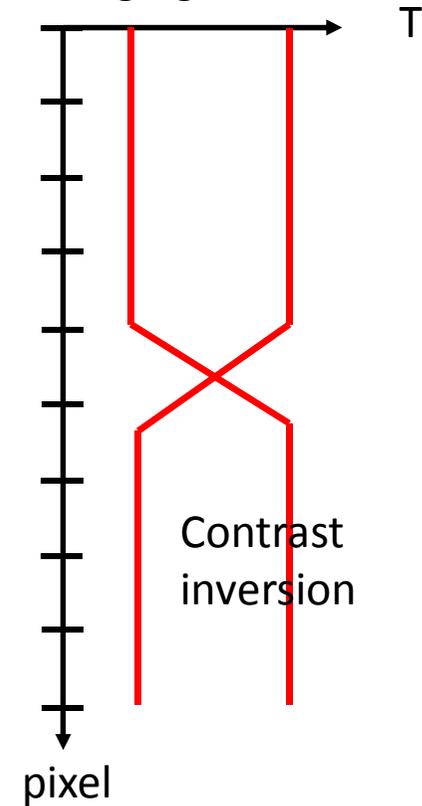
$$2d_{hkl} \sin \theta_2 = \lambda_{hkl}$$

Add 2nd crystal



Different orientation,  
Different Bragg dip location

Monochromatic imaging

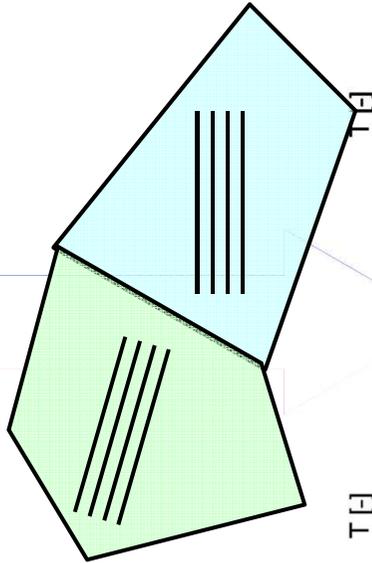


Crystallite contrast  
if above spatial  
resolution



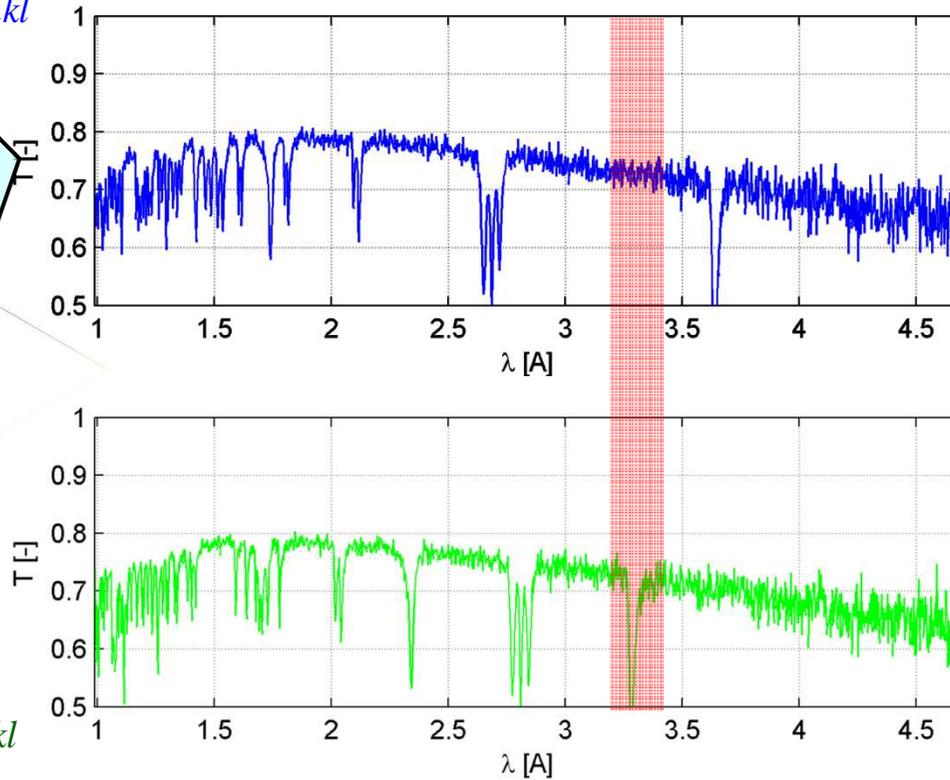
# Single Crystal: Bragg dips

$$2d_{hkl} \sin \theta_1 = \lambda_{hkl}$$



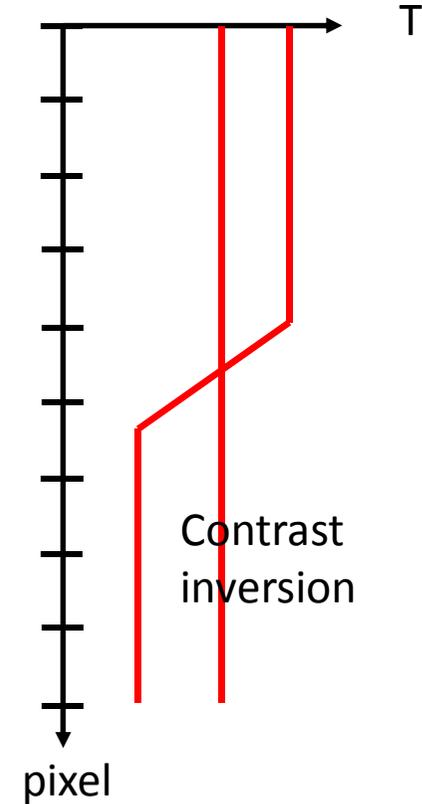
$$2d_{hkl} \sin \theta_2 = \lambda_{hkl}$$

Add 2nd crystal



Different orientation,  
Different Bragg dip location

Monochromatic imaging



Crystalized contrast  
No diffuse scattering, pure elemental contrast



# Example: monochromatic imaging of the Mont-Dieu meteorite

Photo

Mineral inclusions

Crystallite morphology

→ Classification of meteorites

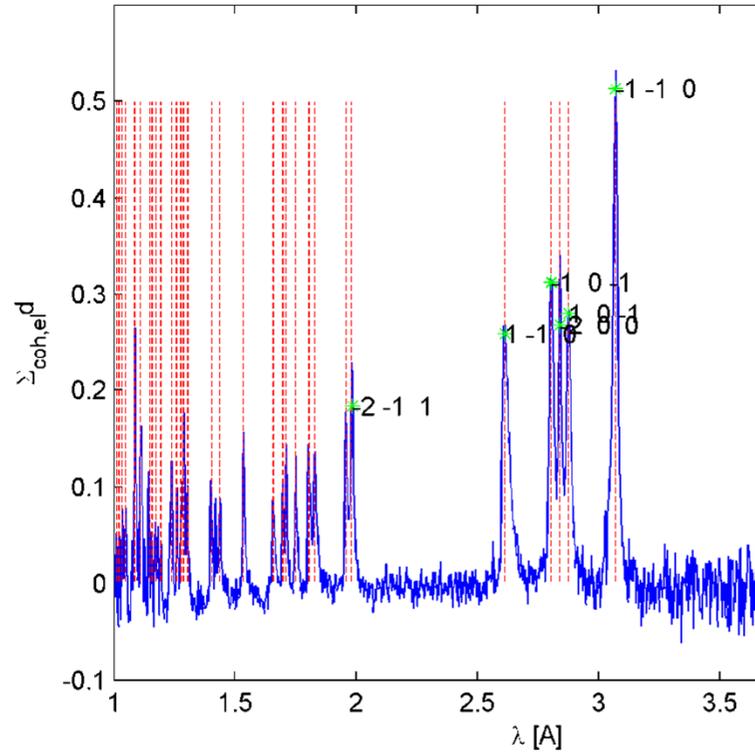
5Å

2.6Å

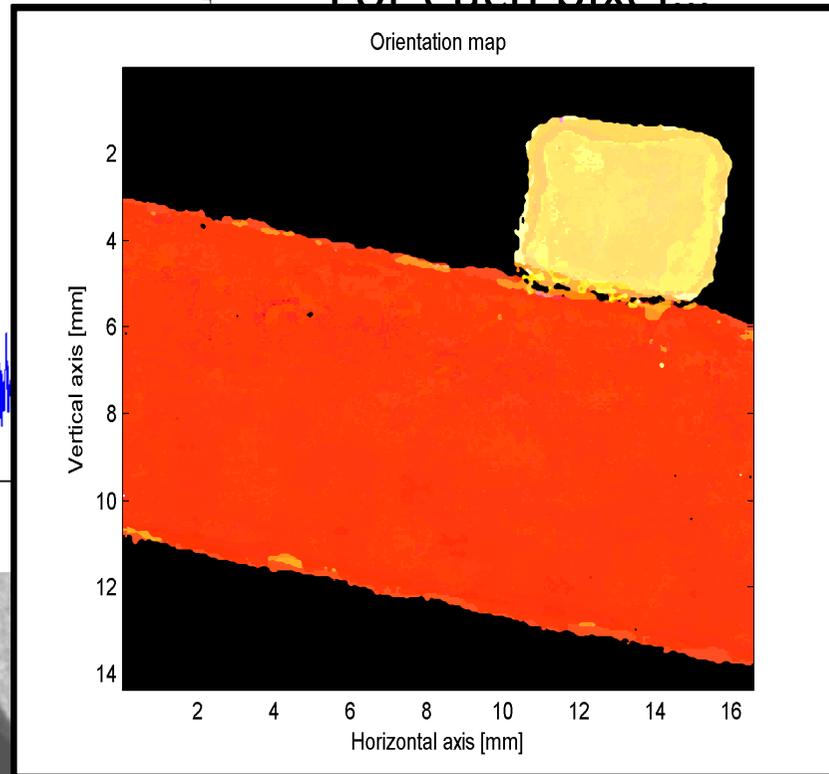
www.voicechronicle.com

Peetermans, S. *et al.*, *Analyst*, **2013**, *138*, 5303-5308

# Energy scan of single crystal



Mapping of crystal properties  
 (orientation, phase, mosaicity)  
 at 100 $\mu$ m spatial resolution  
 for each pixel

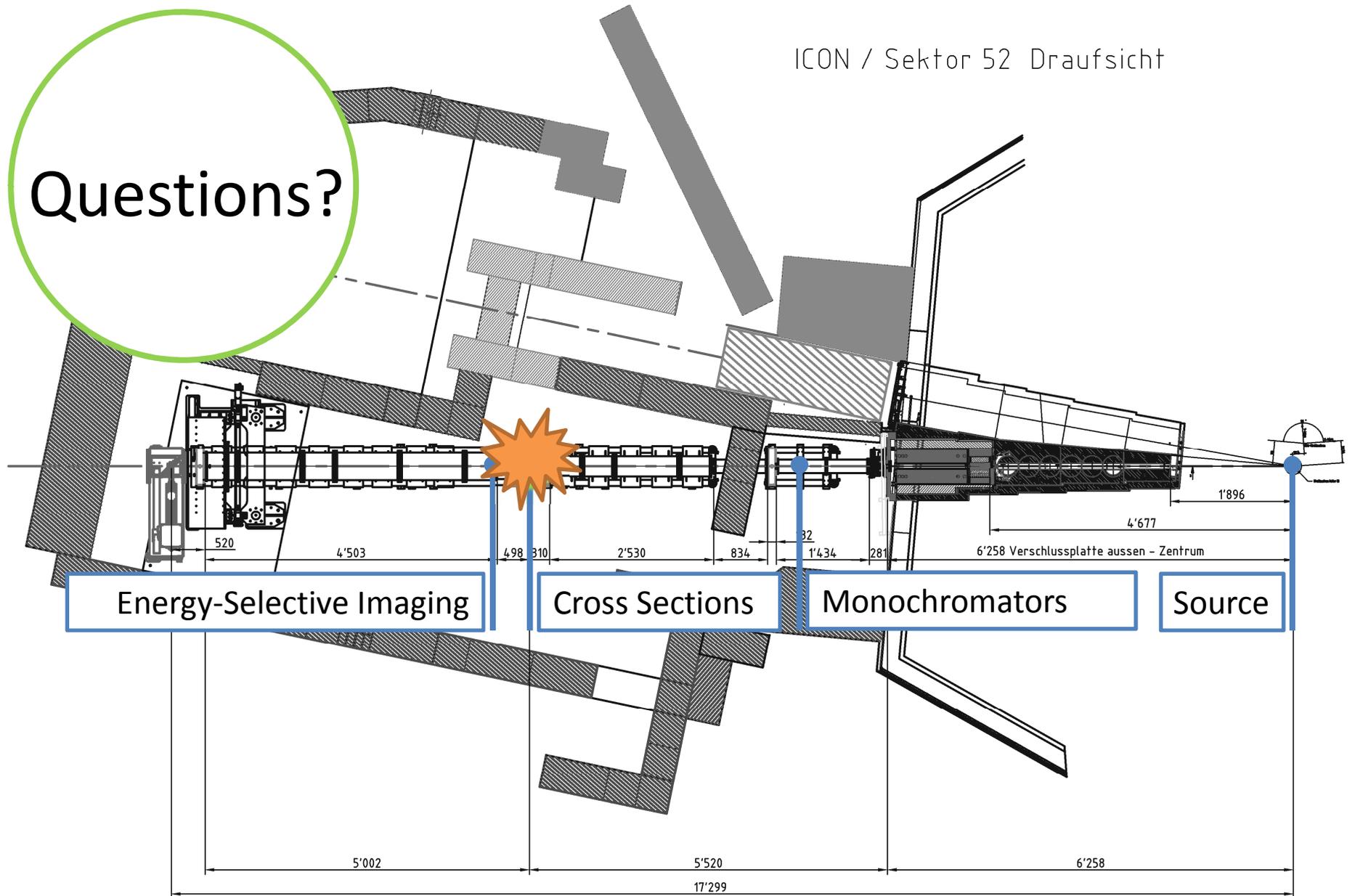


- (1 0 0)
- (0 1 0)
- (0 0 1)

X3

$(1\ 0\ 0)$  closely aligned to the beam  
 15° difference between both

Questions?

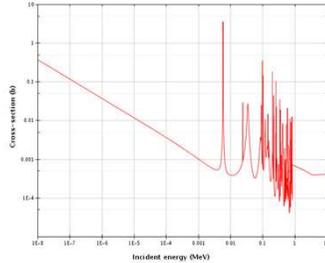




The absorption cross-section is 3 barn at 2.5Å, what's it at 5Å?

A. 6 barn

$$\sigma_{\text{abs}} \sim \lambda$$



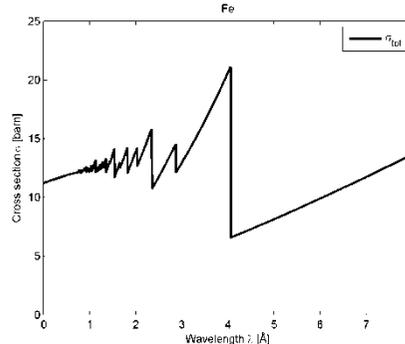
C. 9 barn

D. I'll have to look it up

If I want exploit the Bragg edges to see a phase transition, I will choose a spectrum that is

A. Thermal

B. Cold



I expect the Bragg edge position to shift by 0.01Å due to strain. The best option is to use:

A. Velocity selector

B. Double Crystal Monochromator

C. Time of flight

Velocity selector

Double crystal monochromator

Time of flight

$$\Delta\lambda/\lambda \approx 15\%$$

$$\Delta\lambda/\lambda \approx 2-5\%$$

$$\Delta\lambda/\lambda \approx 0.5\%$$