



Swiss Accelerator  
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Technology



# WireChar Multiphysical characterization of $\text{Nb}_3\text{Sn}$ wires and REBCO coated conductors

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# Outline

- **Transverse stress tolerance of Nb<sub>3</sub>Sn : results of the measurement campaign on state-of-the-art and R&D wires**
  - Irreversible reduction of  $I_c$  in PIT and RRP wires submitted to transverse loads, a method to identify the dominant mechanism behind the degradation
  - Effect of the wire rolling on the irreversible stress limit, FE simulations and experiments
  - First experiments on Distributed Tin (DT) Nb<sub>3</sub>Sn wires
- **A new experiment to test the delamination strength of REBCO coated conductors**
- **Summary and conclusions**

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# Stresses in Nb<sub>3</sub>Sn-based accelerator magnets

There is a need for **higher-field** magnets in view of particle accelerators **beyond LHC**

The projects of the **11 T dipoles** and the **MQXF quadrupoles** for **HL-LHC** provided very relevant experience on how critical is stress management for Nb<sub>3</sub>Sn-based magnets

The **combination of the electromagnetic forces with the pre-compression and the thermomechanical effects** exposes the **brittle and strain sensitive** Nb<sub>3</sub>Sn to the risk of degradation

Several works investigated **Rutherford cables** subjected to **transverse pressure** at room temperature and at low temperature

P. Ebermann *et al.*, Supercond. Sci. Technol., **31** (2018) 065009 DOI: [10.1088/1361-6668/aab5fa](https://doi.org/10.1088/1361-6668/aab5fa)

K. Puthran *et al.*, IEEE TAS, **33** (2023) 8400406 DOI: [10.1109/TASC.2023.3241568](https://doi.org/10.1109/TASC.2023.3241568)

G. De Marzi *et al.*, Sci. Rep., **11** (2021) 7369 DOI: [10.1038/s41598-021-86563-x](https://doi.org/10.1038/s41598-021-86563-x)

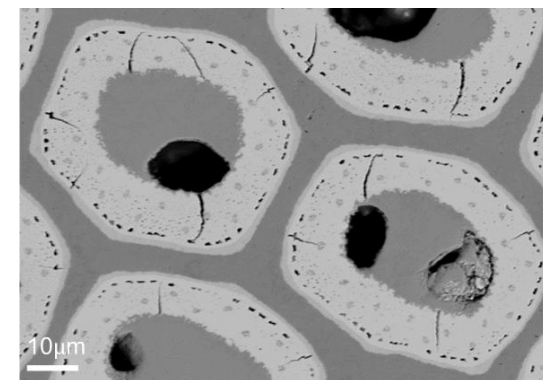
Our experiments focused on the **critical current degradation** of state-of-the-art and R&D **wires** submitted to **transverse pressure at 4.2 K**

# Degradation mechanisms and irreversible reduction of $I_c$

Two mechanisms govern the irreversible reduction of the critical current

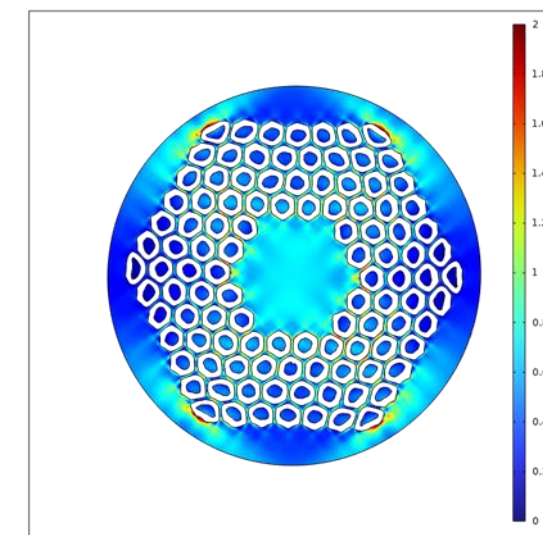
- Formation of **cracks** in the  $Nb_3Sn$  filaments due, for instance, to the stress concentration at the voids

Cracks generate a reduction of the current carrying cross section  $\Rightarrow I_c^{\text{unload}}/I_{c0}$  is independent of the magnetic field



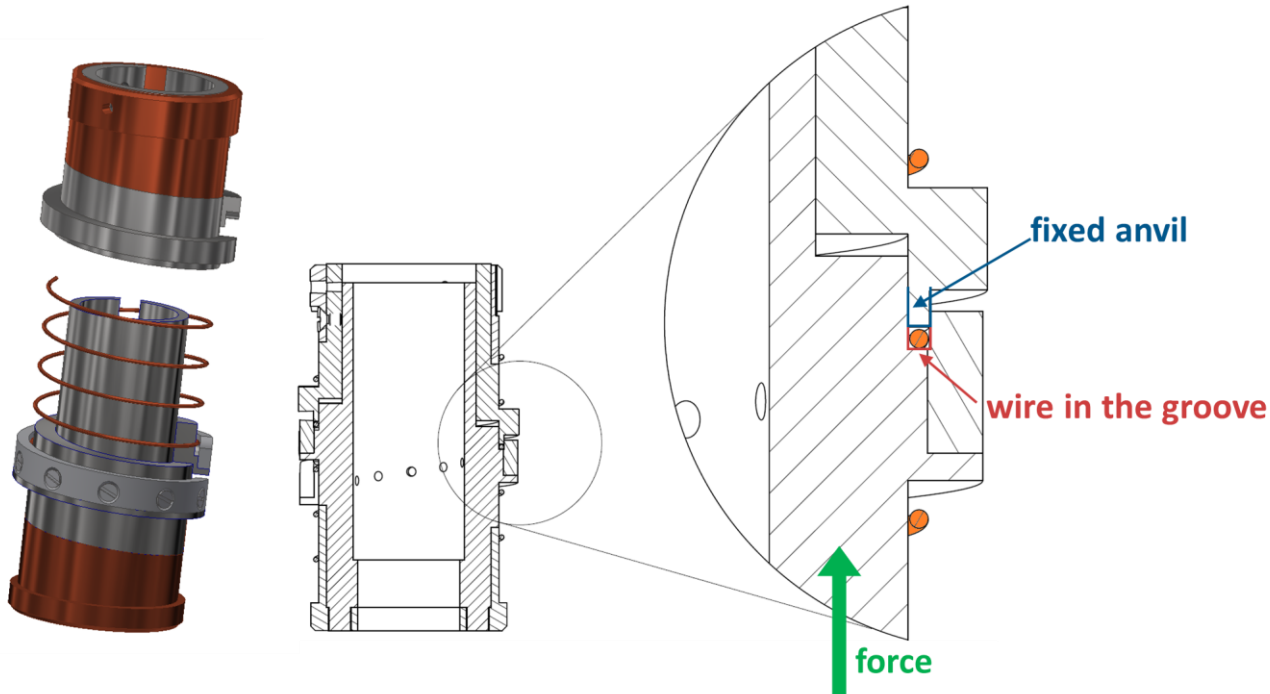
- **Plastic deformation** of the matrix and residual stress on the  $Nb_3Sn$  filaments

Residual stress induces a permanent reduction of  $B_{c2}$  after unload  $\Rightarrow I_c^{\text{unload}}/I_{c0}$  depends on of the magnetic field

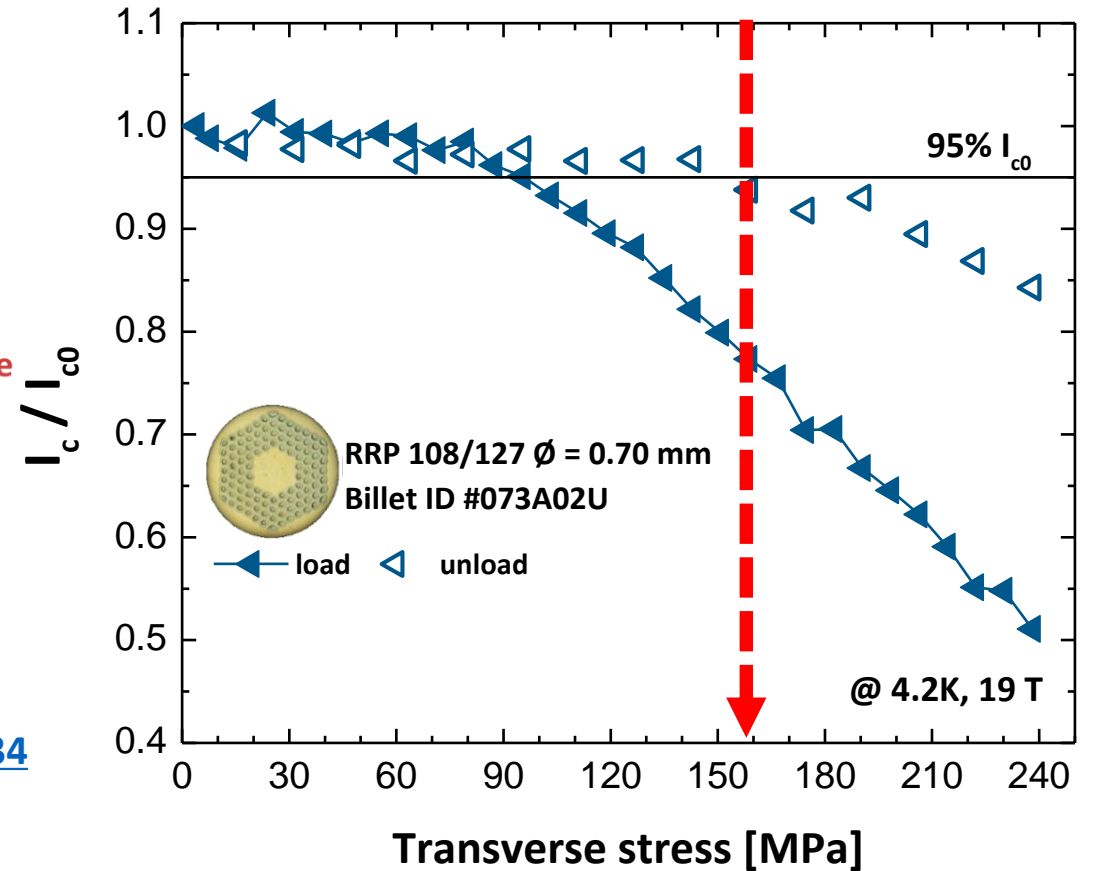


# Tolerance to transverse stress of a single wire

## Electromechanical tests on Nb<sub>3</sub>Sn wires impregnated with epoxy



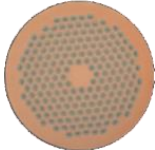

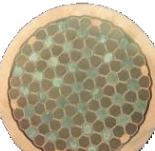
The WASP concept was adapted to transverse loads by  
 B. Seeber et al., IEEE TAS 17 (2007) 2643, DOI: [10.1109/TASC.2007.897934](https://doi.org/10.1109/TASC.2007.897934)



The **irreversible limit** is defined at the force level leading to a **95% recovery of the initial  $I_c$**  after unload

Here the irreversible stress limit is  $\sigma_{irr}$  (B=19T) = **155 MPa** (force divided by groove area)

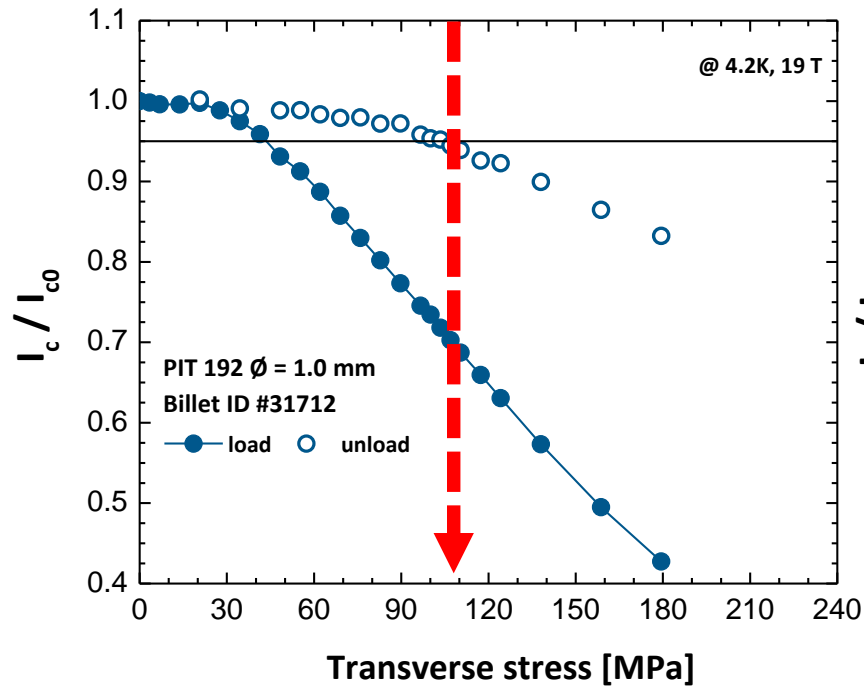
# What has been tested – Catalogue of the Wires

	Technology	# of subelements	Cu/non-Cu	Subelement size/shape	Diameter	$I_c$ (16 T)	Wire ID	
	PIT	192	1.22	50 $\mu$ m round	1.0 mm	340 A	#0904 #31712 #14310	} FRESCA-2
				43 $\mu$ m round	0.85 mm	240 A	#29992	
	RRP	108/127	1.2	~55 $\mu$ m	0.85 mm	280 A	#14753 #14516 (Ta) #76A08U #A02U4014 #271A05	} MQXF
				~45 $\mu$ m	0.7 mm	160 A	#073A02 #271A05	
	DT		0.3		1.2 mm	680 A	R&D	

# Irreversible stress limits at B = 19 T

PIT,  $\varnothing = 1.0$  mm

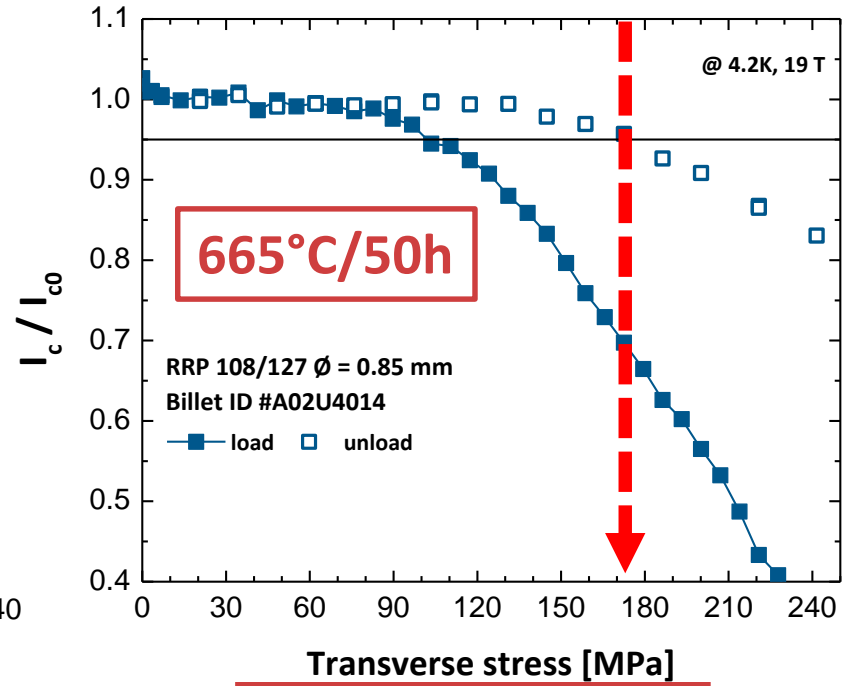
Tested in a 1.15 mm groove



$\sigma_{irr} \approx 110$  MPa

RRP,  $\varnothing = 0.85$  mm

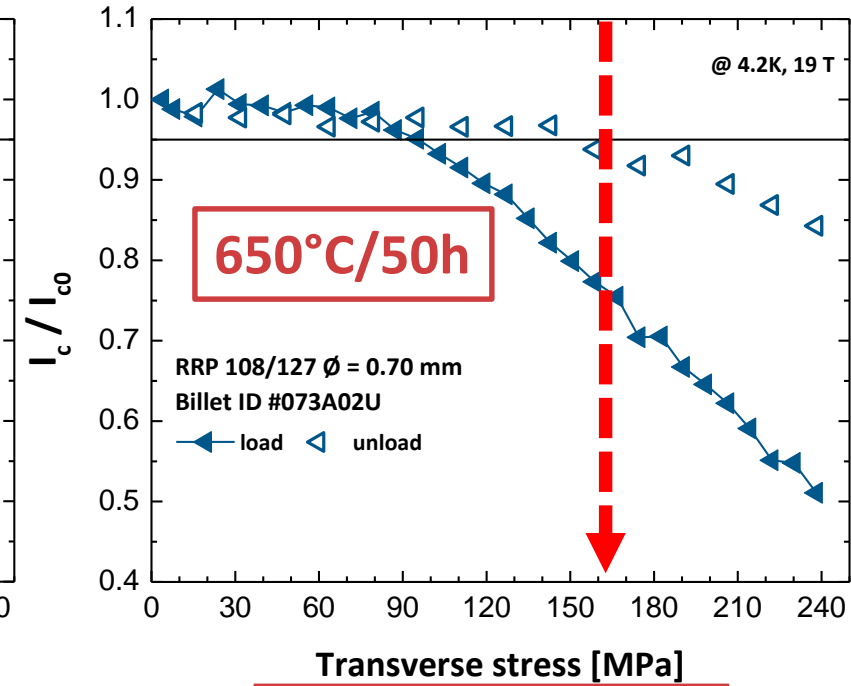
Tested in a 1.15 mm groove



$\sigma_{irr} \approx 175$  MPa

RRP,  $\varnothing = 0.7$  mm

Tested in a 1 mm groove



$\sigma_{irr} \approx 155$  MPa

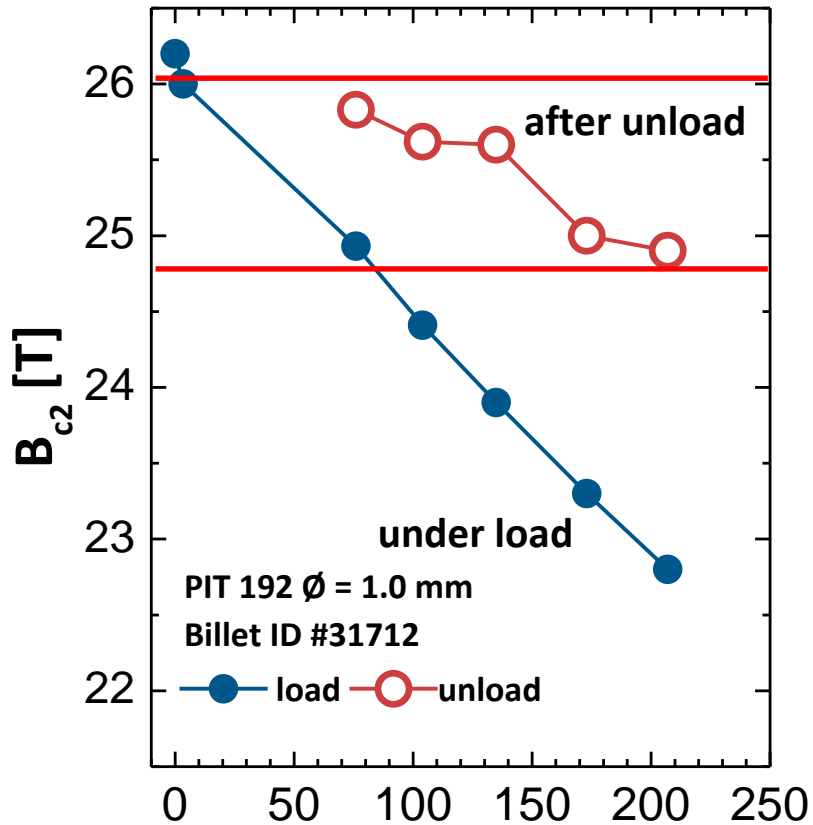
- RRP wires are definitely more robust than PIT wires
- A reduction of  $\sigma_{irr}$  is measured for the RRP wire reacted at lower temperature (650°C)



# Comparison of $B_{c2}$ under load and after unload

(as determined from Kramer extrapolation)

PIT,  $\phi = 1.0$  mm

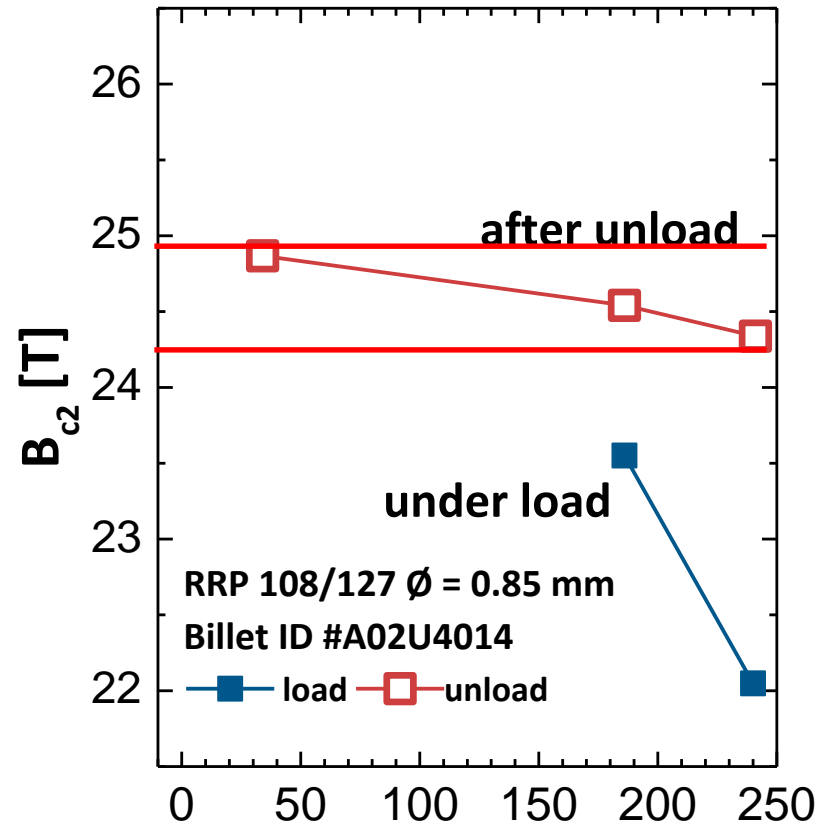


Transverse stress [MPa]

After unload from  $\sigma = 210$  MPa

$$\Delta B_{c2}^{\text{unload}} \approx 1 \text{ T}$$

RRP,  $\phi = 0.85$  mm

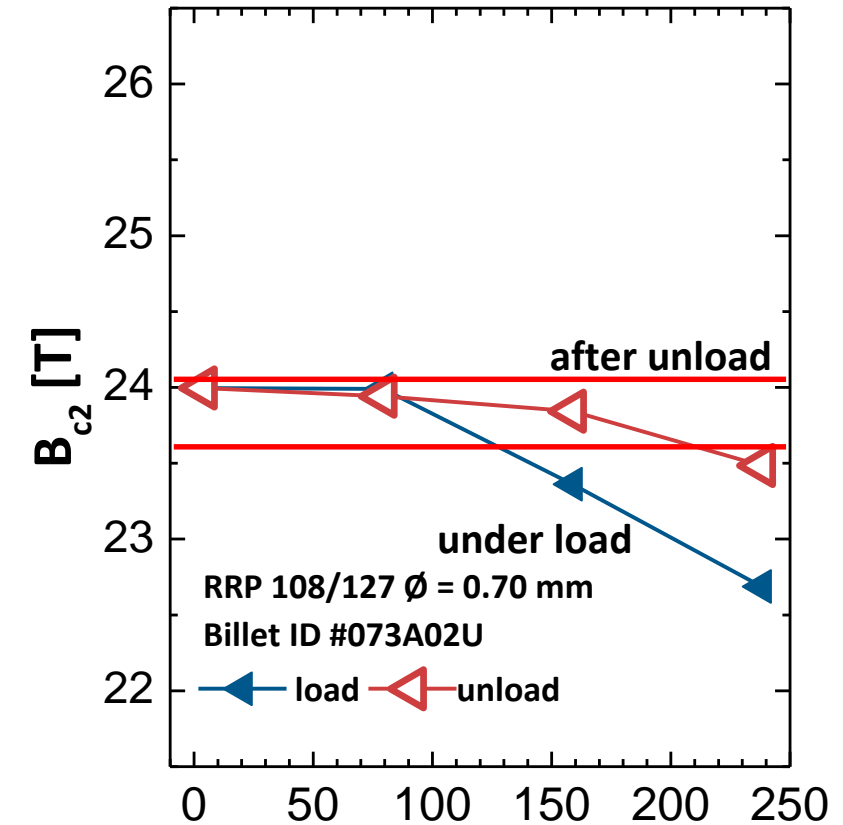


Transverse stress [MPa]

After unload from  $\sigma = 240$  MPa

$$\Delta B_{c2}^{\text{unload}} \approx 0.5 \text{ T}$$

RRP,  $\phi = 0.7$  mm



Transverse stress [MPa]

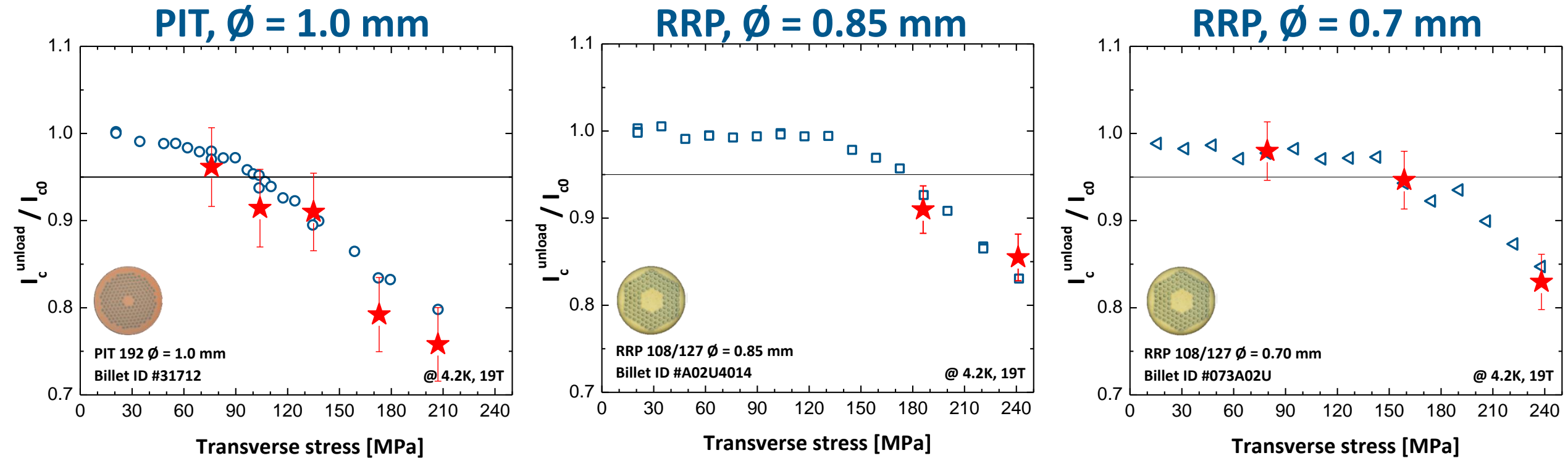
After unload from  $\sigma = 240$  MPa

$$\Delta B_{c2}^{\text{unload}} \approx 0.5 \text{ T}$$

$$I_c^{\text{unload}}(B) = f C \left[ \frac{B_{c2}^{\text{unload}}}{B} \right]^{0.5} \left[ 1 - \frac{B}{B_{c2}^{\text{unload}}} \right]^2$$

# Measured $I_c^{\text{unload}}$ vs expected degradation from residual stress

(open blue points) vs (solid red stars)

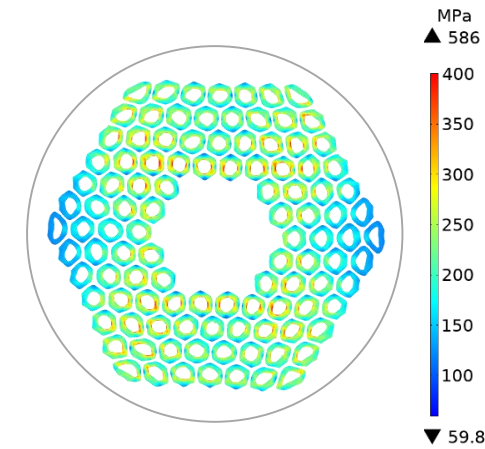
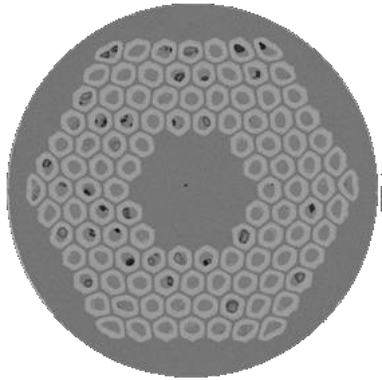


The observed permanent degradation of  $I_c$  after unload from  $\sigma$  up to 240 MPa originates mainly from the residual stress on  $\text{Nb}_3\text{Sn}$  due to the plastic deformation of Cu

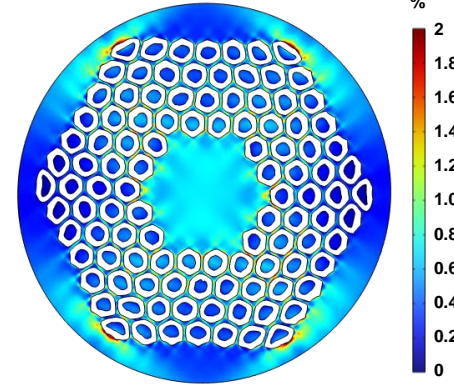
The effect of cracks seems negligible up to very high transverse stress values

# 2D FE electromechanical models with

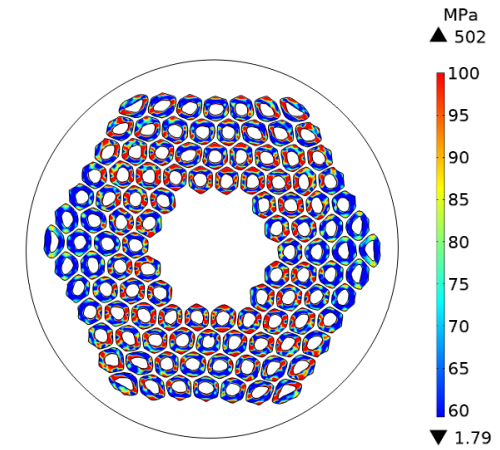
## An independent confirmation of the conclusions



von Mises stress in  $\text{Nb}_3\text{Sn}$   
under load at 30 kN



Plastic strain in Cu after  
unload from 30 kN

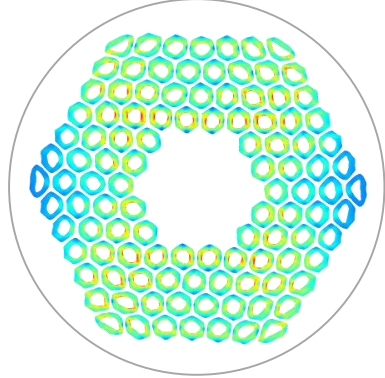
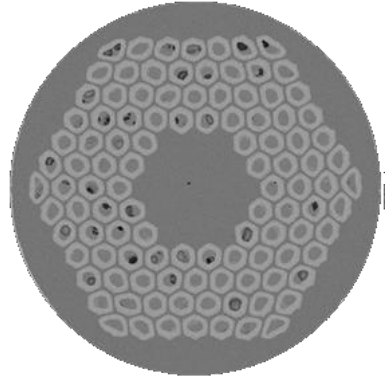


von Mises stress in  $\text{Nb}_3\text{Sn}$   
after unload from 30 kN  
(residual stress)

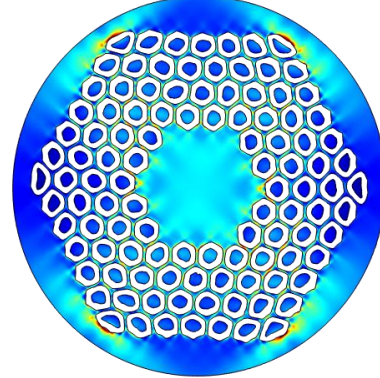
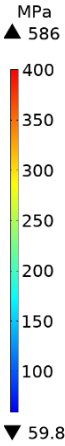
2D plain strain model to visualize the stress and strain distributions in the wire reproducing the experimental conditions of an epoxy impregnated wire under transverse load

Three domains:  $\text{Nb}_3\text{Sn}$ , Cu and bronze – no voids

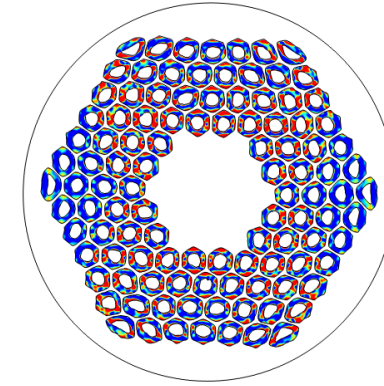
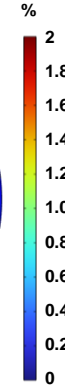
# 2D FE electromechanical models with Round vs 15%-rolled wire



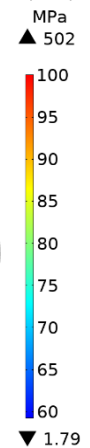
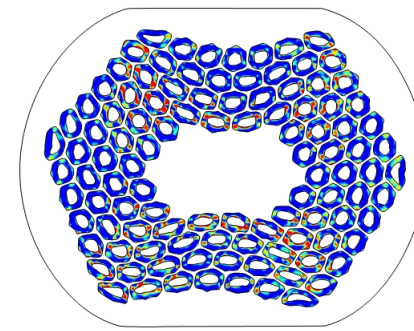
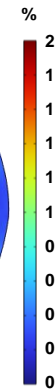
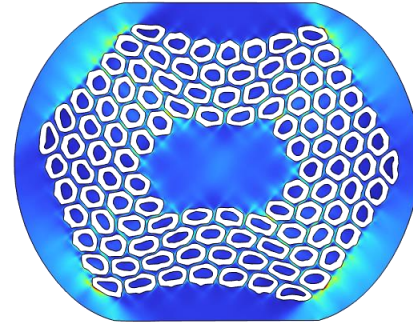
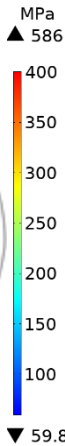
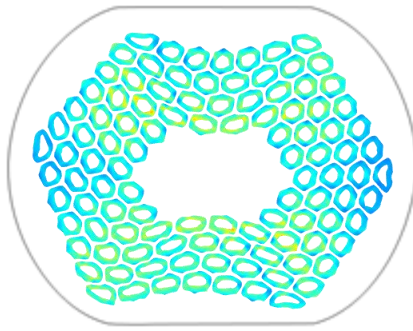
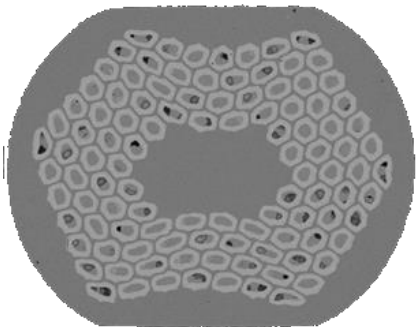
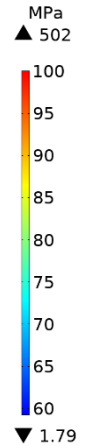
von Mises stress in  $Nb_3Sn$   
under load at 30 kN



Plastic strain in Cu after  
unload from 30 kN



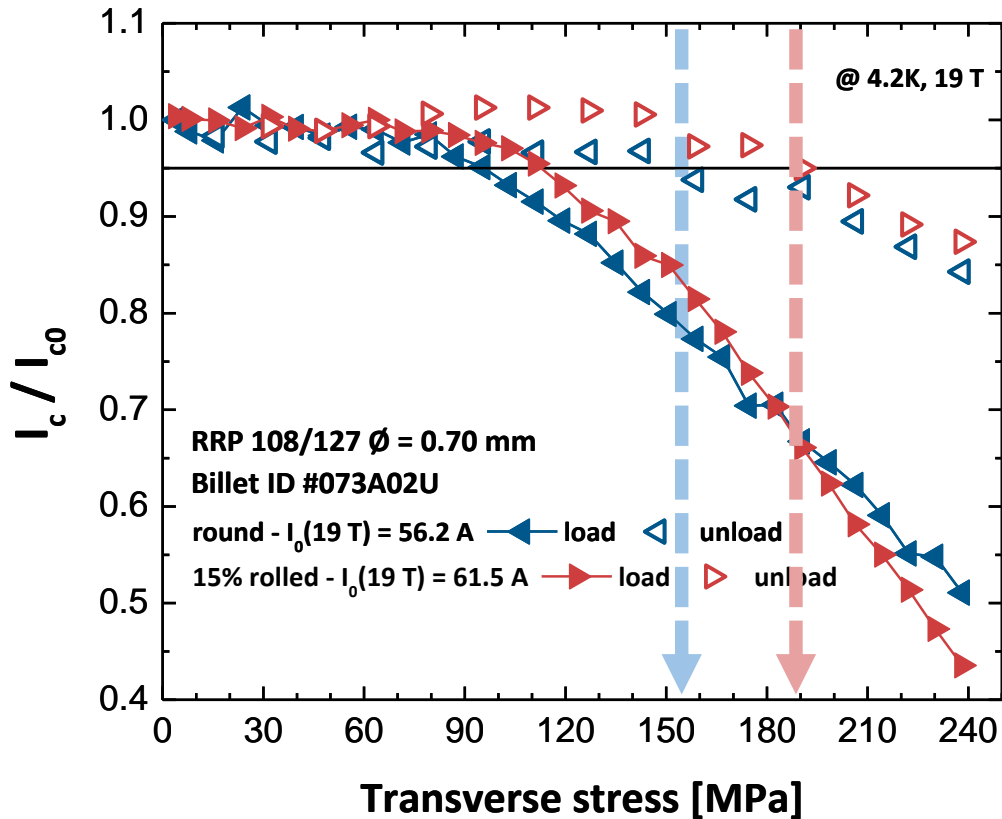
von Mises stress in  $Nb_3Sn$   
after unload from 30 kN  
(residual stress)



According to the simulation, 15% rolling is associated with a reduction of the residual stresses  
This is due to a better redistribution of the load between matrix and superconductor

# Effects of wire rolling on the stress tolerance

## Experimental verification of the outcome of the simulations



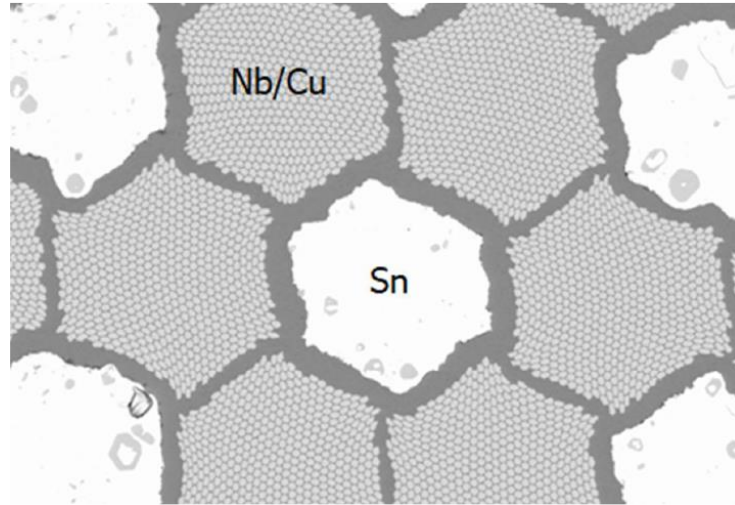
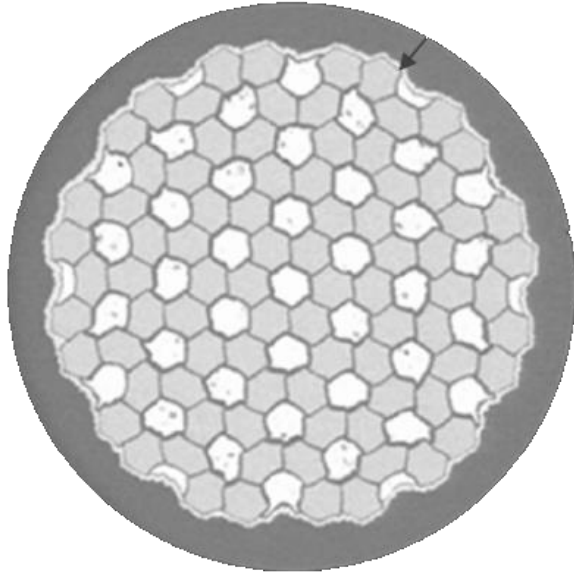
Round wire  
 $I_{c0}(B=19\text{T}) = 56.2$  A  
 $\sigma_{irr} \approx 155$  MPa

15%-rolled wire  
 $I_{c0}(B=19\text{T}) = 61.5$  A  
 $\sigma_{irr} \approx 185$  MPa

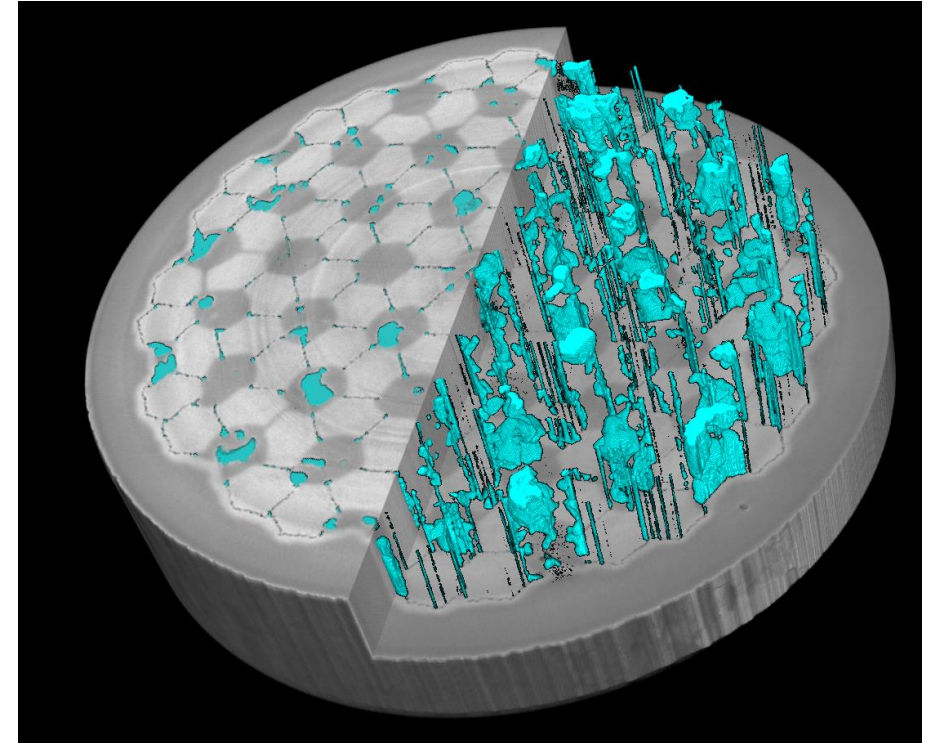
A favorable stress redistribution in the 15%-rolled wire leads to an increased  $\sigma_{irr}$  by  $\sim 30$  MPa

**Is the degradation under transverse stress of Nb<sub>3</sub>Sn wires always driven by residual stresses?**

# Distributed Tin (DT) wires from

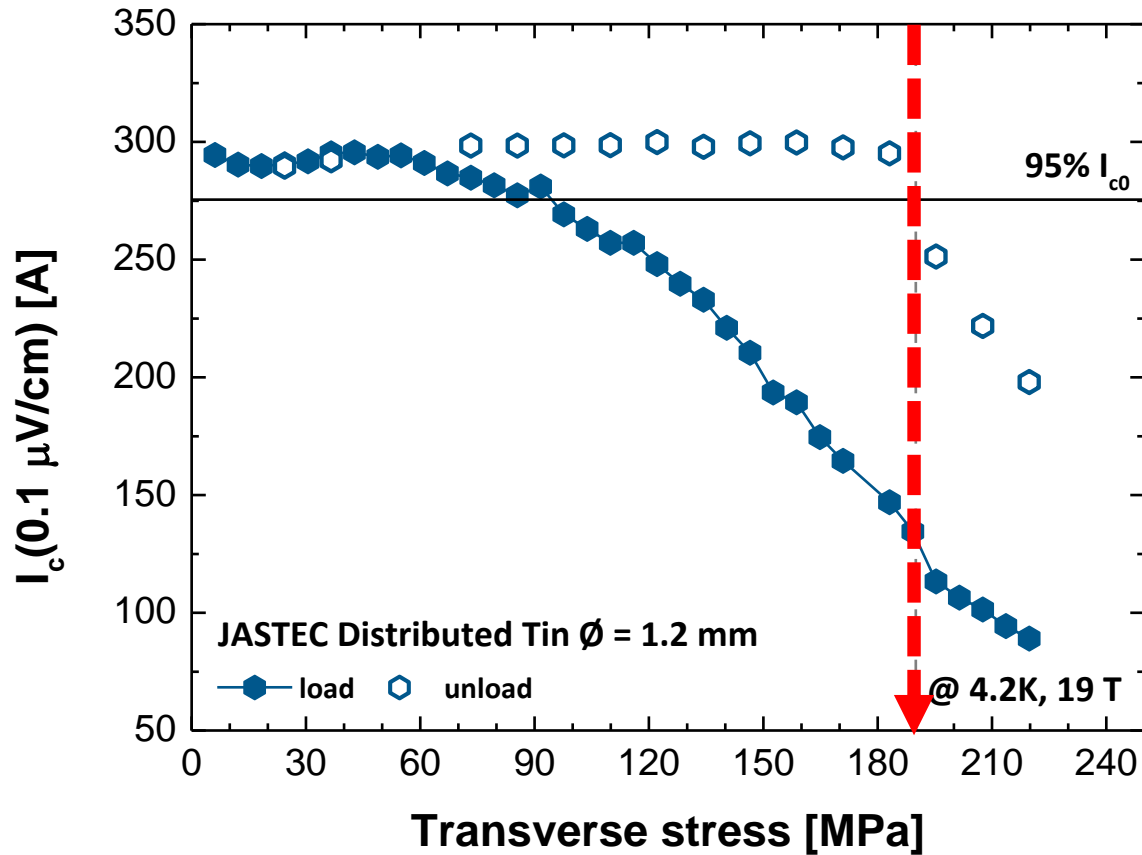


SEM picture courtesy of Dr. Yoshito Fukumoto from JASTEC



X-ray tomography reconstruction and void detection by machine learning, preliminary analysis

# Irreversible degradation driven by cracks in DT wires



The abrupt decrease of  $I_c$  after unload at 190 MPa is a clear sign of degradation driven by the formation of cracks

The Cu/non Cu ratio is only 0.3  
This can explain the marginal role of residual stresses

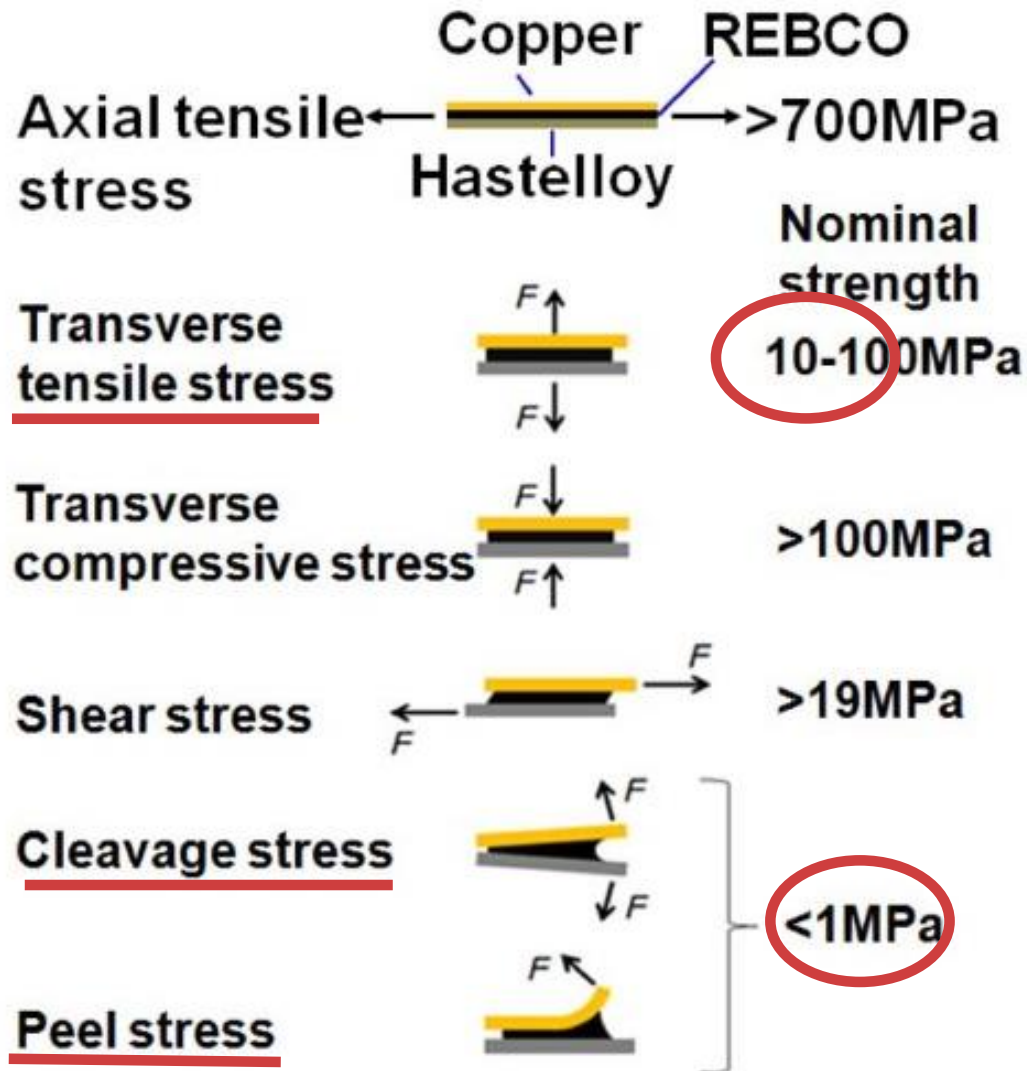
Similar conclusions reached in the paper of M. Nakamoto *et al.*, IEEE TAS, 33 (2023) 8400505  
DOI: [10.1109/TASC.2023.3242919](https://doi.org/10.1109/TASC.2023.3242919)



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  - Irreversible reduction of  $I_c$  in PIT and RRP wires submitted to transverse loads, a method to identify the dominant mechanism behind the degradation
  - Effect of the wire rolling on the irreversible stress limit, FE simulations and experiments
  - First experiments on Distributed Tin (DT) Nb<sub>3</sub>Sn wires
- **A new experiment to test the delamination strength of REBCO coated conductors**
- Summary and conclusions

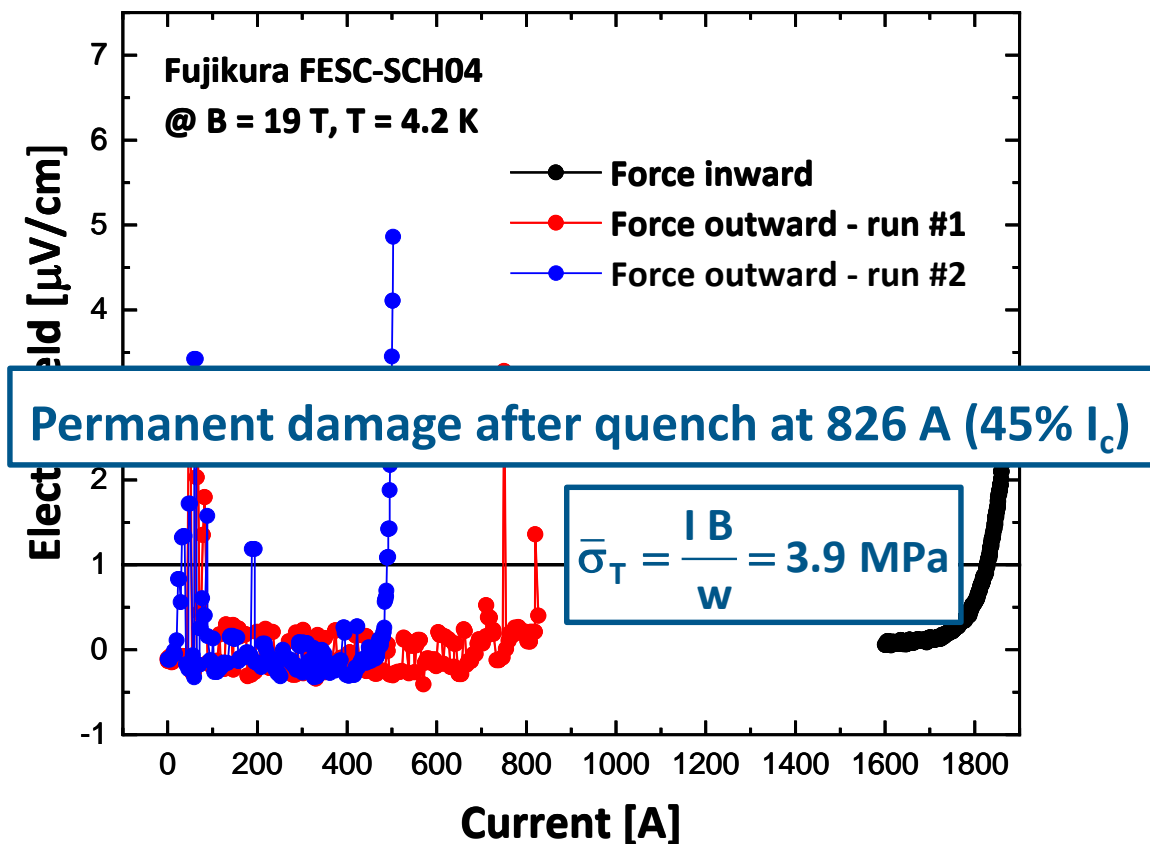
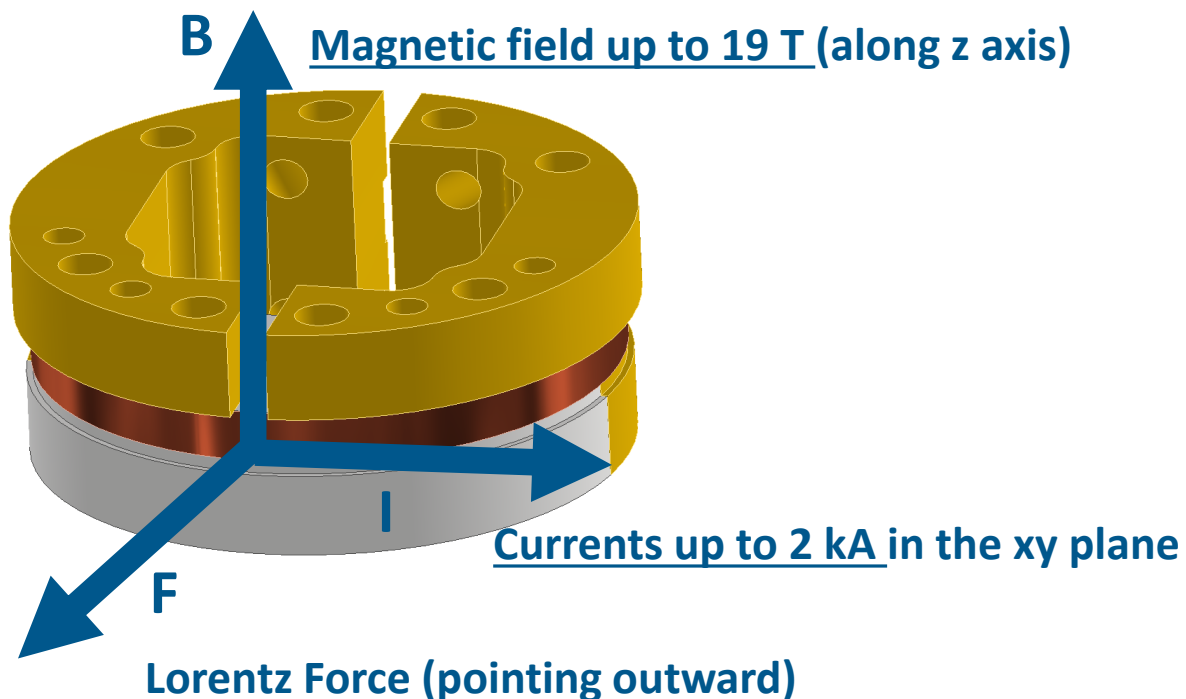
# A short note on the mechanical properties of REBCO tapes



- REBCO tapes are inherently prone to delamination
- Adhesion between layers seems to be process dependent
- A standardized process to determine the properties of the tapes is missing

# Delamination strength measurements under $I \times B$ force

A novel experiment for a direct measurement of the conductor degradation



- The REBCO tape is mounted with the superconducting layer outward
- The Lorentz force is outward and tends to detach the REBCO layer from the substrate
- A transverse tensile stress in the range 1 – 10 MPa is generated

# Conclusions

- Experiments show that the permanent **degradation of  $I_c$  in PIT and RRP wires** is dominated by **residual stresses**. The conclusion is supported by FE simulations
- The results obtained on a **Distributed Tin** wire suggest that **degradation** is driven by the **formation of cracks**. This particular wire has a very low Cu/non Cu ratio
- The study on rolled wires showed that **FE simulations** can become a valuable tool for the development of wire designs tailored to improve the stress tolerance
- The **new experiment** to measure the **delamination strength** of REBCO tapes aims to produce **robust data with a reproducible process**. The results may have many implications also on the manufacturing process



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# Thank you for the attention !

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DE GENÈVE**

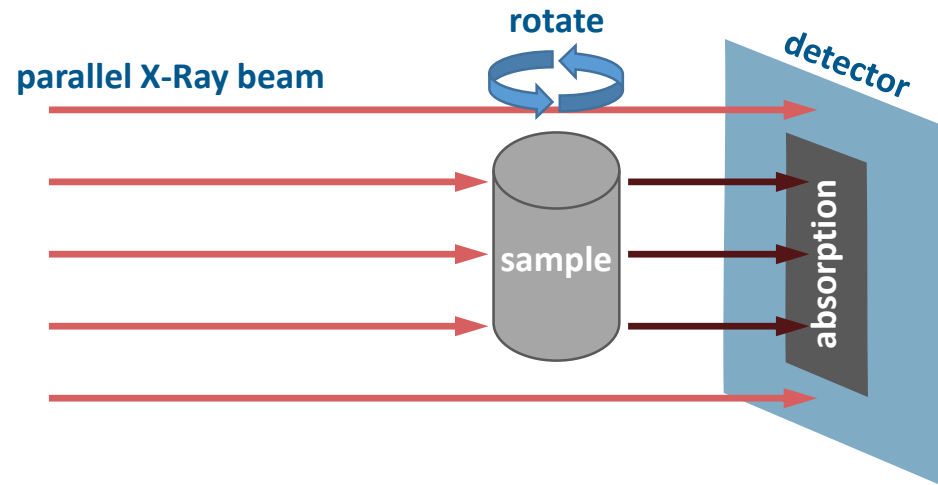
FACULTÉ DES SCIENCES

**Acknowledgments:** This work was done under the auspices of CHART (Swiss Accelerator Research and Technology Collaboration, <https://chart.ch>). Financial support was provided by the European Organization for Nuclear Research (CERN), Memorandum of Understanding for the FCC Study, Addendum FCC-GOV-CC-0176 (KE 4612/ATS).

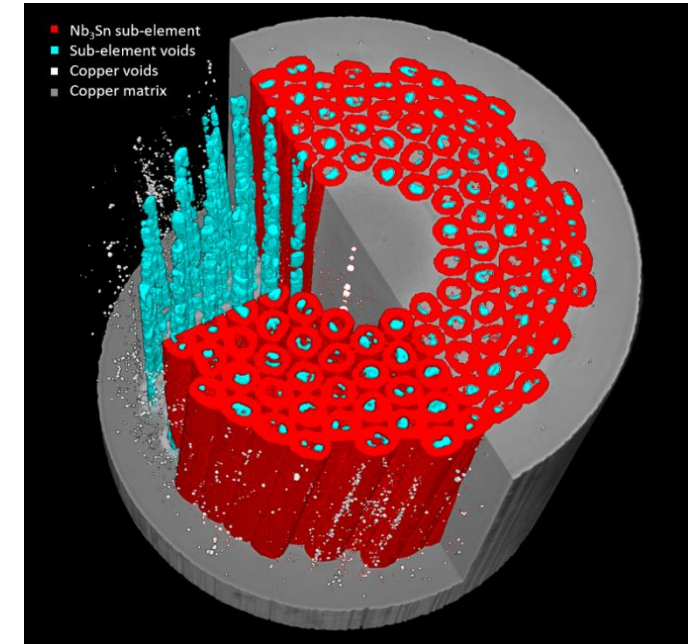
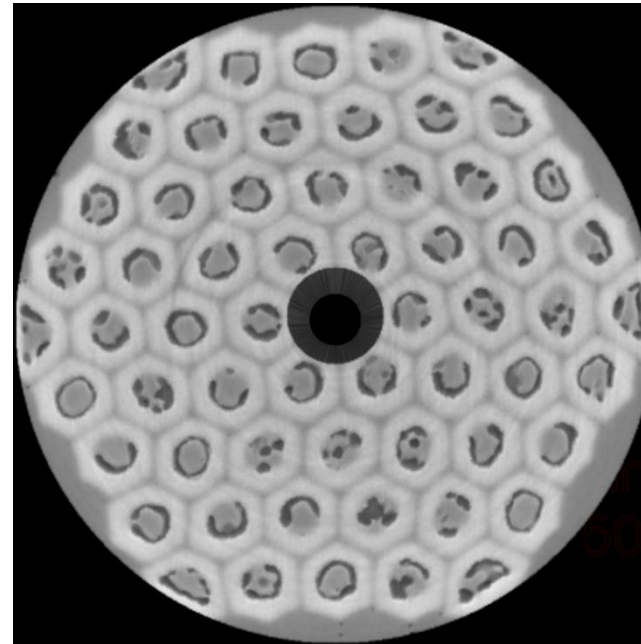


# X-ray tomography and Machine Learning for crack detection

## An independent confirmation of the conclusions



- X-ray photon energy = 80 keV
- 360° rotation of the sample
- 10'000 projections
- 2560 x 2160 pixels
- 0.57  $\mu\text{m}$ /pixel resolution



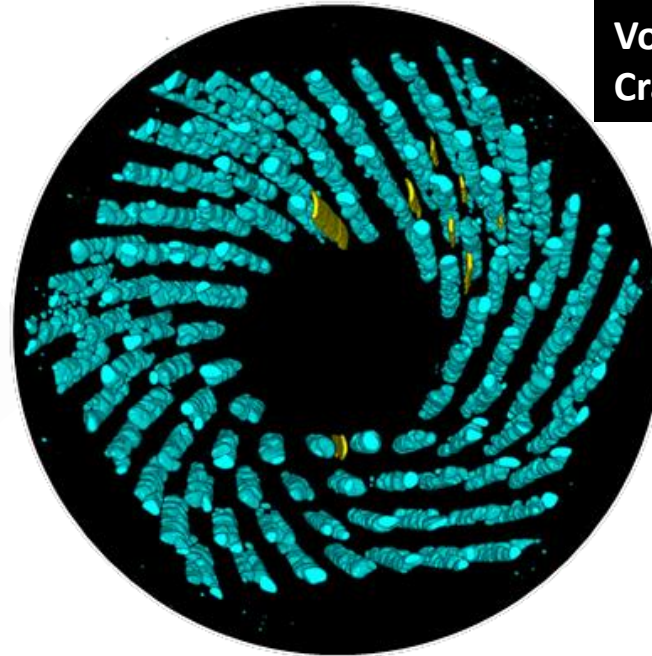
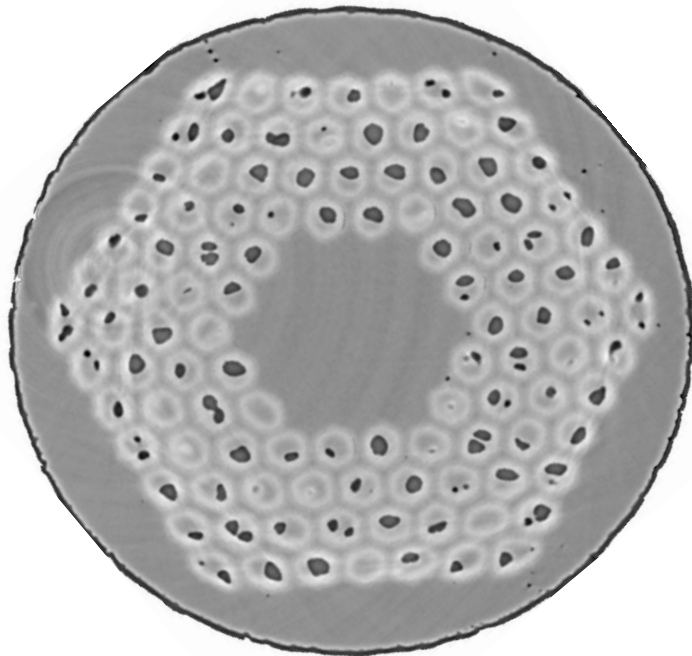
A novel, non-destructive and non-invasive method to investigate the internal structure of high-performance Nb<sub>3</sub>Sn wires combines X-ray microtomography with machine-learning algorithms



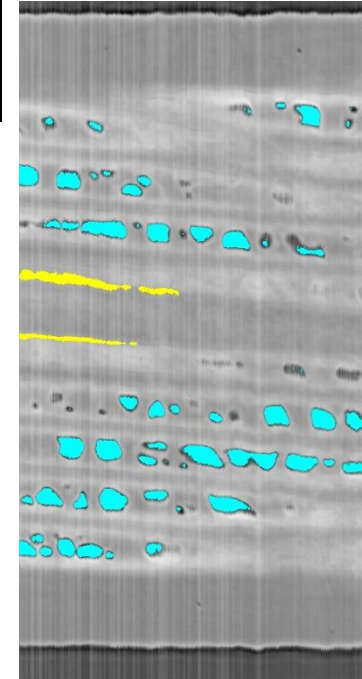
Marta MAJKUT  
Alexander RACK

# X-ray tomography and Neural Networks for crack detection

An independent confirmation of the conclusions



Voids in cyan  
Cracks in yellow



An analysis based on Convolutional Neural Networks was performed on the tomographic scan of the exact same sample used for the  $I_c$  vs  $\sigma$  test, after unload from 240 MPa

**Very few cracks were detected**, none of them interrupting the subelements and responsible for the measured degradation by 15% of  $I_c$



# What has been tested

A comprehensive campaign of **electromechanical tests** to gain knowledge on some very practical aspects

- Probed the **impact of the impregnation scheme on the transverse stress tolerance**

Quantified the effect of the rigidity of the impregnation on the irreversible stress limit of PIT wires

L. Gämperle, *et al.*, Phys. Rev. Research 2 (2020) 013211

DOI: [10.1103/PhysRevResearch.2.013211](https://doi.org/10.1103/PhysRevResearch.2.013211)

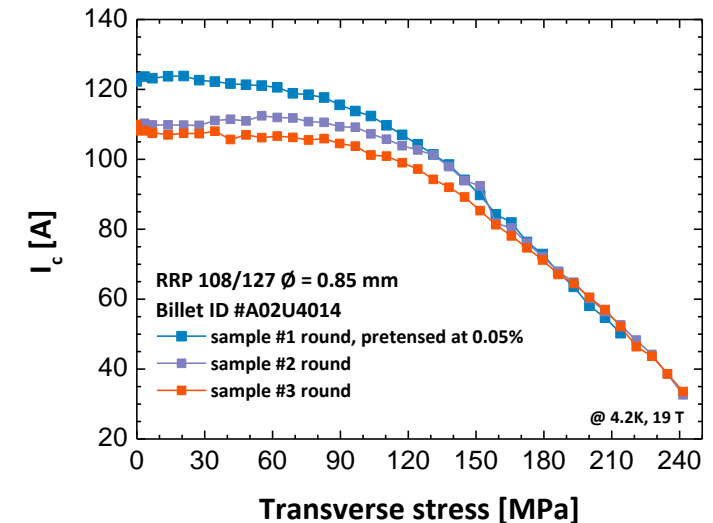
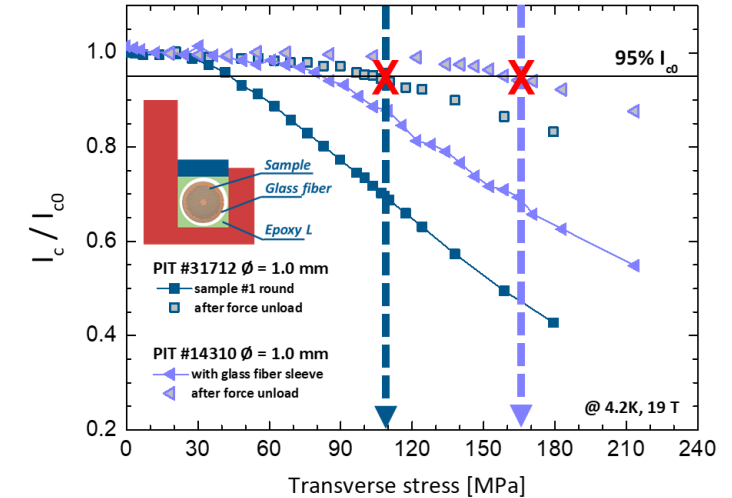
- Assessed the effect of the **longitudinal strain state on the response to transverse stress**

Investigated RRP wires under longitudinal and transverse loads

J. Ferradas Troitino, *et al.*, Supercond. Sci. Technol., 34 (2021) 035008

DOI: [10.1088/1361-6668/abd388](https://doi.org/10.1088/1361-6668/abd388)

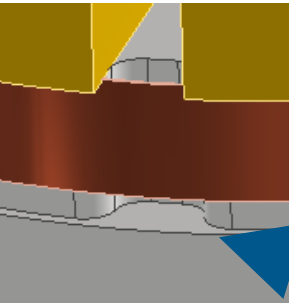
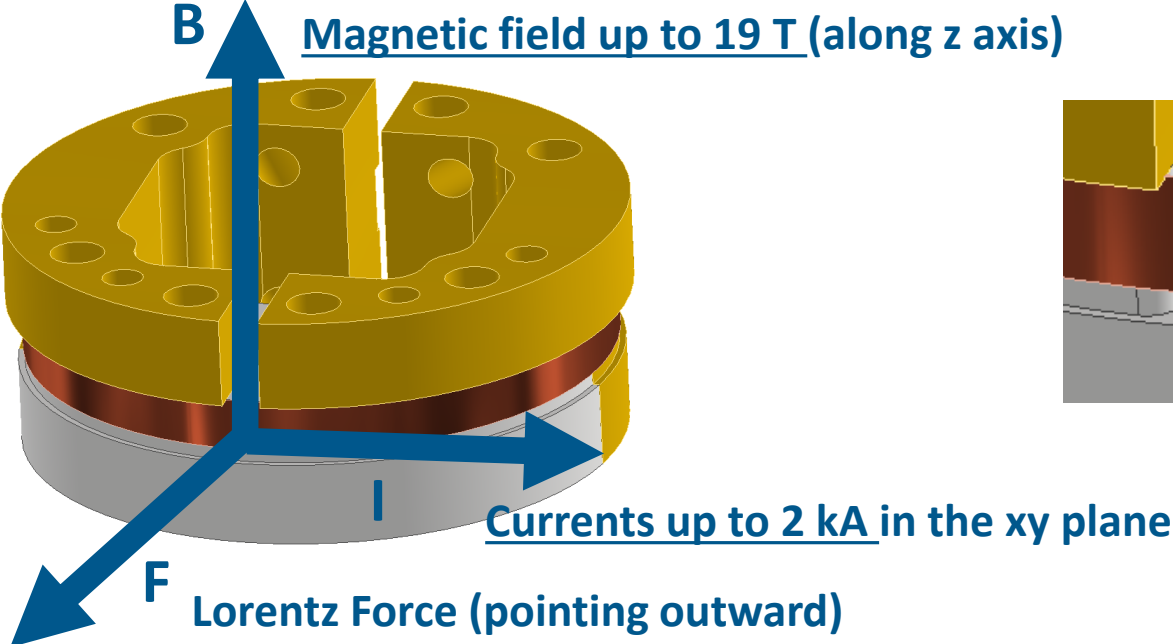
Wire in a glass fiber sleeve + Epoxy L



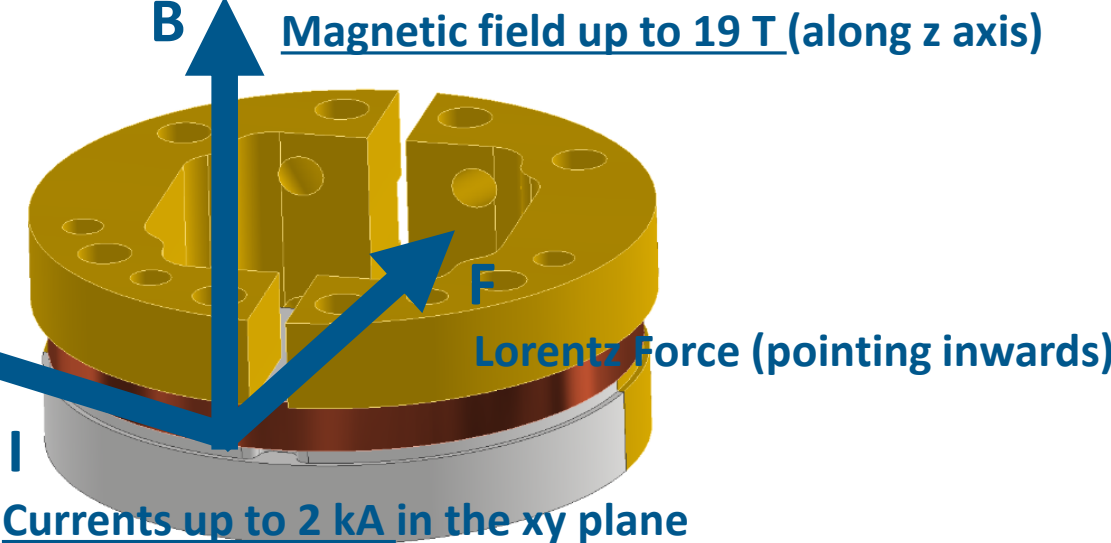
# Delamination strength measurements under $I \times B$ force

A novel experiment for a direct measurement of the conductor degradation

REBCO layer on the outside



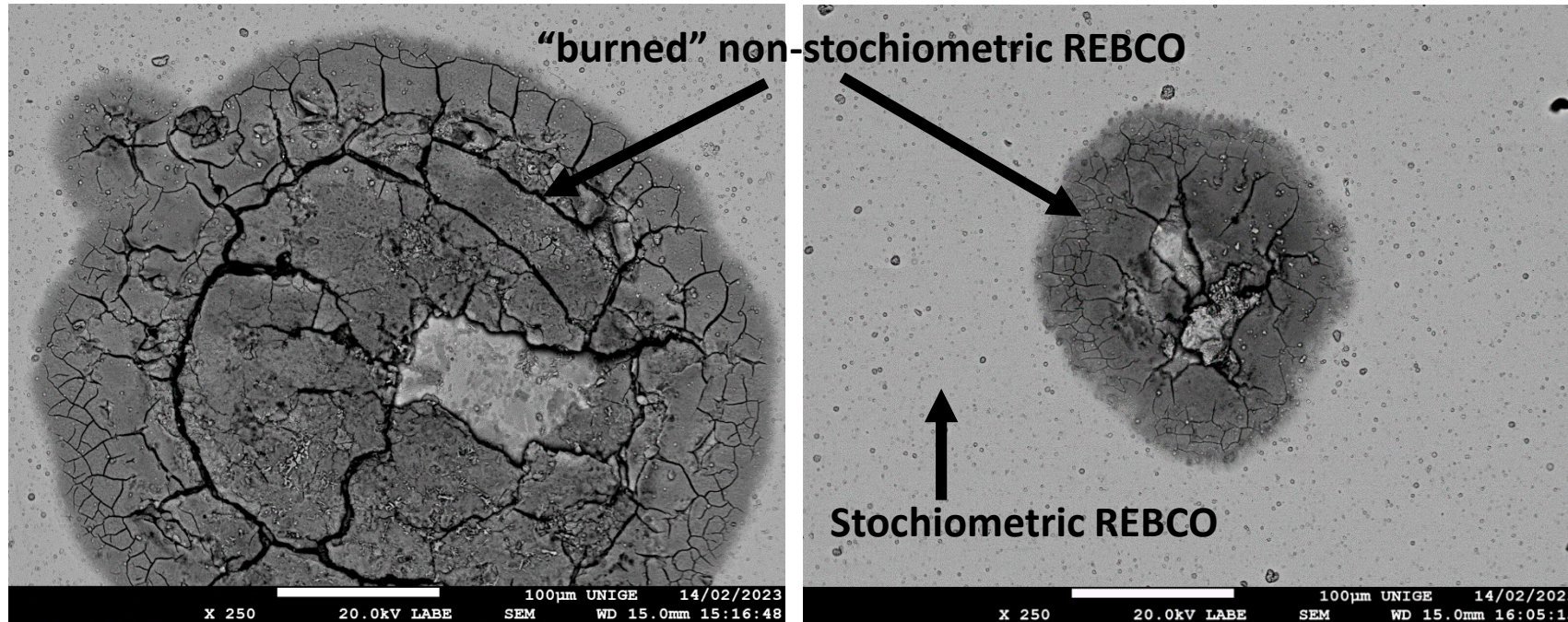
REBCO layer on the inside



# Delamination strength and forensic microscopy

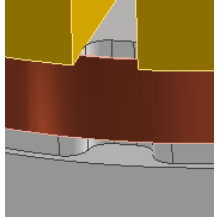
SEM-EDX and 3D Optical microscopy

Delamination → Quench → Local heating → Blistering

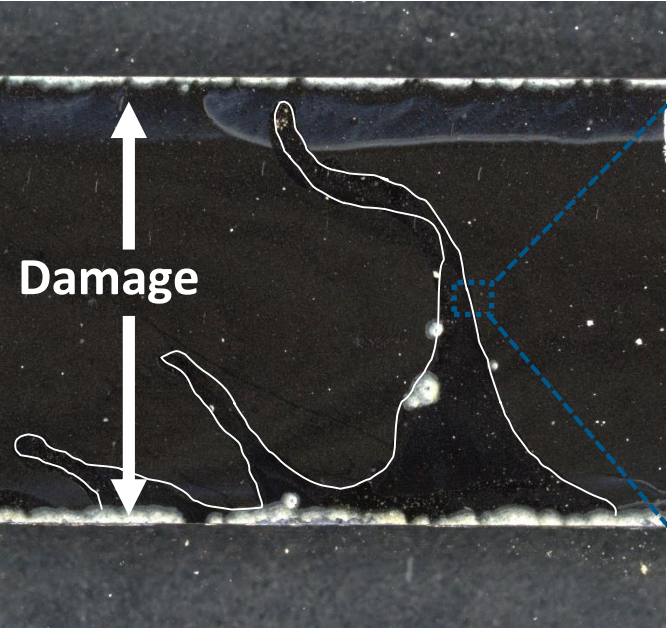


# Delamination strength and forensic microscopy

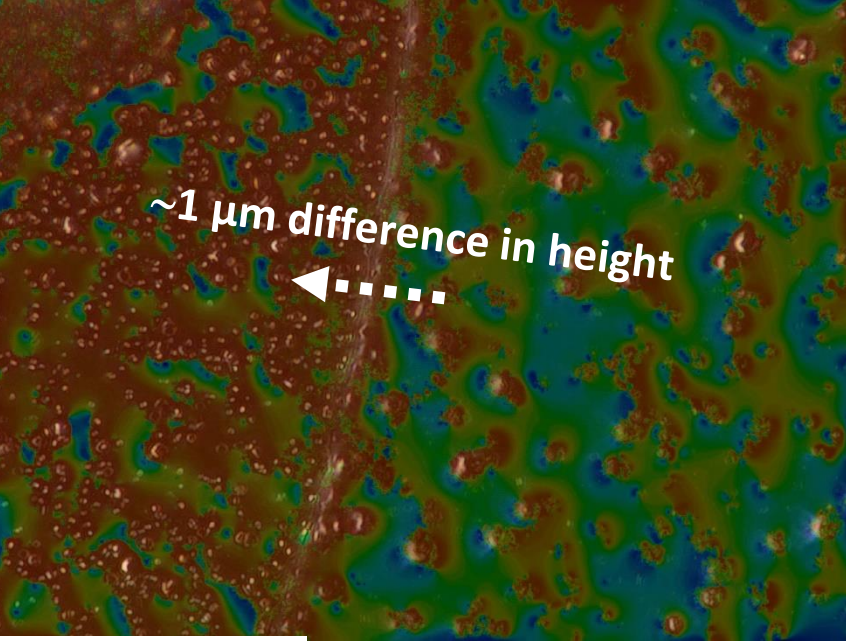
SEM-EDX and 3D Optical microscopy



Evidence of delaminated area where the tape was not supported



3D optical microscope analysis



z-axis color code

97% of  $I_c$  was lost after the delamination experiment

# Summary



Type: FESC SCH04  
Tape ID: 19-0008

Delamination strength ~ 4 MPa  
(45%  $I_c$ )

$\bar{\sigma}_T = 3.9$  MPa     $F_L$  outward  
 $\bar{\sigma}_T = 4.1$  MPa     $F_L$  inward

Longitudinal damage on both  
edges of the CC + Blisters



Type: FESC SCH02  
Tape ID: 23-0010

Delamination strength ~ 4.3 MPa  
(39%  $I_c$ )

$\bar{\sigma}_T = 4.3$  MPa     $F_L$  inward

Longitudinal damage on both  
edges of the CC + Blisters



Type: SCS4050-AP  
Tape ID: M3-1464-7

Delamination strength ~ 5.5 MPa  
(82%  $I_c$ )

$\bar{\sigma}_T = 5.5$  MPa     $F_L$  inward

Longitudinal damage on 1 edge  
of the CC + Blisters

